

Lawrence Berkeley National Laboratory

LBL Publications

Title

Exploring non-adiabaticity to CO reduction reaction through ab initio molecular dynamics simulation

Permalink

<https://escholarship.org/uc/item/8pg9b4zx>

Journal

APL Materials, 8(4)

ISSN

2166-532X

Authors

Zheng, Fan
Wang, Lin-wang

Publication Date

2020-04-01

DOI

10.1063/5.0002318

Supplemental Material

<https://escholarship.org/uc/item/8pg9b4zx#supplemental>

Peer reviewed

Exploring non-adiabaticity to CO reduction reaction through ab initio molecular dynamics simulation

1
Fan Zheng and Lin-wang Wang*

*Joint Center for Artificial Photosynthesis and Materials Sciences Division, Lawrence
Berkeley National Laboratory, Berkeley, California 94720, USA.*

E-mail: lwwang@lbl.gov

2 Abstract

3 Non-adiabatic chemical reaction refers to the electronic excitation during reactions.
4 This effect can not be modeled by the ground-state Born-Oppenheimer molecular dy-
5 namics (BO-MD), where the electronic structure is at the ground-state for every step
6 of ions' movement. Although the non-adiabatic effect has been explored extensively
7 in gas phase reactions, its role to the electrochemical reactions in electrolyte such as
8 water splitting and CO₂ reduction has rarely been explored. On the other hand, elec-
9 trochemical reactions usually involve electron transport, thus, non-adiabatic process
10 can naturally play a significant role. In this work, using one step of CO₂ reduction as
11 an example, we investigate the role of the non-adiabatic effect in the reaction. The
12 reaction barriers are computed by adiabatic BO-MD and non-adiabatic real-time time
13 dependent density functional theory (rt-TDDFT). We find that by including the non-
14 adiabatic effect, rt-TDDFT can increase the reaction barrier up to 6% compared to the
15 BO-MD calculated barrier when solvent model is used to represent the water. Simu-
16 lations with hybrid solvent model using explicit water molecules around the reaction

17 site is also carried out under different overpotentials, similar non-adiabatic effects are
18 found.

19 First-principles methods, such as density function theory (DFT), have been widely used
20 in a variety of electrocatalytical reactions, such as water splitting including oxygen evolu-
21 tion¹⁻⁵ and hydrogen evolution reactions,⁶⁻⁹ CO₂ reduction,¹⁰⁻¹³ and solar fuel cells.¹⁴⁻¹⁶
22 By utilizing state-of-art computational techniques such as computational hydrogen elec-
23 trode model,¹⁷ nudged elastic bands (NEB),¹⁸ and Born-Oppenheimer molecular dynamics
24 (BO-MD),¹² DFT calculations enable a detailed free-energy determination of possible reac-
25 tion paths, including intermediate states, transition states and reaction barriers. Various
26 effects, for example pH,¹⁹ electrode potential,^{10,17} cation induced electric field,^{20,21} and elec-
27 trolyte^{12,14,22,23} have been explored systematically to illustrate the reaction mechanisms and
28 to design high-performance catalysis. Most of these methods, particularly to determine the
29 reaction barriers, are based on the ground-state DFT. By assuming a much slower reac-
30 tion process compared to the time-scale of electron thermalization, the electronic excitations
31 during the related reactions have been ignored, hence, adiabatic Born-Oppenheimer assump-
32 tion is implicitly used in many studies for electrocatalysis. The non-adiabatic (NA) effect
33 results from the electronic evolution with finite time-scale. However, the NA effect to the
34 electrochemical reactions is rarely explored, and we have very limited knowledge about if
35 the non-adiabaticity will affect the electrochemical reactions and to what extend it can con-
36 tribute.

37 Electronic NA effect is defined as the coupling of the electronic ground state to excited
38 eigen-states due to time evolution of the wavefunctions. This effect manifests itself in a
39 chemical reaction when the time-scale of its dynamics is similar to that of the carrier's
40 (electrons and holes) thermalization. As a result, the excited electron (and hole) is not
41 always at its equilibrium ground state. Meanwhile, for non-excited cases with pure charge
42 transport, NA effect can result in charge transfer bottleneck, where carriers need finite time
43 to move from the carrier donor to the acceptor. In comparison, when the reaction is carried
44 out very slowly, the fast relaxation of electronic excitation or fast transfer makes the NA effect
45 unimportant to the reaction. Owing to the ultrafast nature, the NA effect plays a significant

46 role in chemical reactions such as photochemistry,^{24,25} collision,²⁶ atom-stopping,²⁶⁻²⁸ and
47 electron transfer process.^{25,29} Particularly, a great deal of research has been focused on gas-
48 phase catalytic reactions on surfaces to understand the contribution from the NA effect.³⁰⁻³⁵
49 For example, the NA simulation with fewest-switches surface-hopping algorithm has shown
50 the strong NA effect for spin flipping and transition during the O₂ dissociative adsorption on
51 Al and Pd surface, with the estimated rate consistent to the experiments.^{36,37} The reverse
52 process associative desorption of N₂ on Ru(001) further shows the NA effect from an *ab*
53 *initio* simulation indirectly.³⁸ In that work, consistent agreement between the simulation
54 and the experiments can be obtained only after including the NA effect in the calculation. A
55 comprehensive theoretical study has been made to explore the NA effect of H₂/Cu (110) and
56 N₂/W (110). However, their simulation has shown a marginal effect of the non-adiabaticity to
57 diatomic molecules adsorption process.³⁷ Based on these examples, the role of the NA effect
58 seems to depend on specific reaction types. However, for electrochemical reactions under
59 aqueous condition such as heterogenous catalysis, the NA effect has been rarely studied.
60 Electrochemical reactions necessitate the transfer of charge from one place to another, thus
61 it is more likely a NA process. Besides the question of excited state induced by the reaction
62 dynamics, another possibility is the charge transfer bottleneck, which also makes the process
63 NA. One recent work focusing the initial CO₂ adsorption to various metal surfaces has shown
64 very fast electronic hybridization compared to the adsorption,²¹ showing the adiabatic nature
65 of the chemical adsorption process. Different from the initial adsorption process for CO₂
66 reduction, its subsequent steps involve the proton-assisted electron transfer (PAET). This
67 process also exists in water splitting and hydrogen evolution reactions. Some of the fast
68 PAETs can form the transition state within 1 ps or faster,³⁹⁻⁴¹ comparable to the time-scale
69 of the electronic thermalization, indicating the potential role of the NA effect. Nonetheless,
70 the NA effect to the electrochemistry reactions with protons addition is rarely investigated
71 in detail. It leaves many questions unanswered: for example, will the NA effect play a role
72 to the electrochemical reactions such as CO₂ reduction involving the fast proton motion? If

73 this is the case, how much does the non-adiabaticity make change to the reaction barrier?
74 Will the NA effect contribute mostly to the carrier excitation or charge transfer bottleneck?
75 How do the electrolyte and applied electrode potential influence the NA?

76 In this work, we seek to understand the role of the NA effects to heterogenous catalyt-
77 ical reactions in aqueous condition. By using one step of CO₂ reduction $\star\text{CO} + \text{H}_3\text{O}^+ +$
78 $e^- \longrightarrow \star\text{COH} + \text{H}_2\text{O}$ (\star : copper surface) on copper [111] surface as an example, we perform
79 the adiabatic (ground-state BO-MD) and non-adiabatic (Erenfest real-time time-dependent
80 DFT (rt-TDDFT)) simulations to model the reaction. In BO-MD, the adiabatic eigen-states
81 are solved by diagonalizing the Hamiltonian at every MD step. The occupation of electrons
82 on each state is based on their eigen-energies. Thus, the electronic structure is always at the
83 ground state for every step. On the contrary, rt-TDDFT evolves the time-dependent wave-
84 function following Schrödinger’s equation, allowing the electronic structure to be excited.
85 Meanwhile, the excitation of the electronic structure may drive the ions movement differently
86 compared to the ground-state electronic structure. With this capability, rt-TDDFT has been
87 widely used to simulate various NA processes such as optical excitations,⁴² proton-assisted
88 chemical reactions,⁴³ and ion sputtering.^{27,44} Different from other TDDFT algorithms where
89 a very small time step ($\Delta t \sim 0.001\text{fs}$) has to be used to evolve the charge density, the imple-
90 mentation we have adopted here uses the adiabatic states ($\phi_j(t)$) as the basis to expand the
91 wavefunction. These adiabatic states are solved from the Hamiltonian at each ion’s step t_n
92 with time-step $\Delta t = t_{n+1} - t_n \leq 0.1\text{fs}$. The time-dependent wavefunction is expanded as:

$$\psi_i(t) = \sum_j C_{i,j}(t) \phi_j(t)$$

93 where adiabatic state $\phi_j(t)$ is solved by diagonalizing the Hamiltonian H at step t : $H(t)\phi_j(t) =$
94 $\epsilon_j(t)\phi_j(t)$. The coefficient $C_{i,j}(t)$ for the wavefunction is evolved from t_n to t_{n+1} non-
95 adiabatically following the Schrödinger’s equation $i\partial\psi(t)/\partial t = H(t)\psi(t)$ using a much smaller

106 time-step. The Hamiltonian in the Schrödinger’s equation is based on the linear interpo-
107 lation from $H(t_n)$ to $H(t_{n+1})$. However, since the adiabatic states (size~100) are used as
108 basis to construct the wavefunction and Hamiltonian instead of plane-waves, the evolution
109 of wavefunction from t_n to t_{n+1} involves only a small size matrix, its cost becomes negligi-
110 ble. This method allows us to evolve wavefunction and ions’ dynamics of a complex system
111 with hundreds of atoms such as the surface chemical reaction presented here. In this work,
112 the reaction is simulated with CO molecule adsorbed on copper [111] surface, and it is at-
113 tacked by H_3O^+ to form COH on copper. We find that the BO-MD and rt-TDDFT with
114 the same initial setups can reveal opposite results: near the reaction barrier, BO-MD allows
115 the reaction to happen, while rt-TDDFT fails to proceed the reaction to form \star COH but
116 return back to \star CO. Such difference clearly demonstrates the role of the non-adiabaticity to
117 electrochemical reactions in aqueous conduction. The reaction barrier change caused by the
118 non-adiabaticity is estimated, which is up to 6% correction compared to the ground-state
119 method calculated barriers. We also explored the reaction with the explicit solvent model
120 and with different electrode potentials, and we find similar NA effects.

111 PWmat^{45,46} package based on the plane-wave pseudopotential DFT is used to perform
112 the total energy calculation, structural optimization, BO-MD, and rt-TDDFT. SG15 pseu-
113 dopotentials⁴⁷ with 50 Ryd plane-wave energy cut-off are used to ensure the convergence
114 of charge density. All the structures are relaxed with the forces below 0.02 eV/Å. In our
115 calculation, we choose four-layer Cu[111] slab surface with 3×3 in x - y direction superlattice.
116 The most widely used DFT exchange-correlation functionals such as PBE and LDA predict
117 CO adsorption on the hallow site of three Cu atoms, contradicting to the experimentally
118 observed top-site (on top of one Cu atom).^{48,49} Instead, we choose revised PBE (revPBE)⁵⁰
119 to reproduce the correct adsorption site. In the dynamical simulations, reaction barrier is
120 sensitive to the length of time step, we find 0.1 fs is enough to obtain accurate reaction
121 barrier (see Supplementary Information (SI)). The rt-TDDFT calculation is also converged
122 with this time step. Both spin-polarized and spin-unpolarized calculations are performed,

123 but these two types of calculation yield quite similar results.

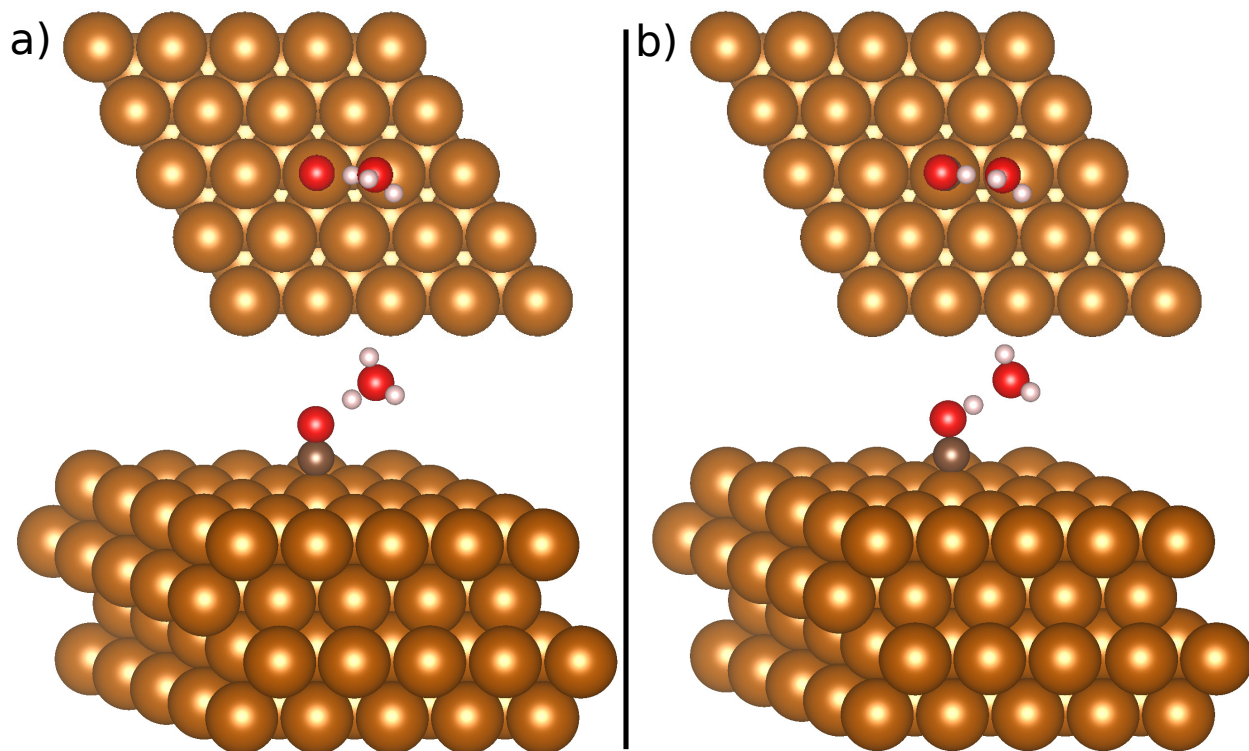


Figure 1: Atomic structures of the a) initial and b) final structures. Top: top view; bottom: side view of the two structures. During the reaction, one proton of hydronium moves from O of H₃O⁺ to O of *CO. Golden: Cu; Red: O; Brown: C; Light violet: H.

124 The solvent has been shown to play a significant role in CO₂ reduction reactions on metal
125 surfaces. Two types of solvent models are commonly used to represent the solvent effect in
126 DFT calculations: implicit solvent model with continuum dielectric response, and explicit
127 solvent model with water molecules in simulation. In our calculation, we have tried both
128 methods to examine the effect of non-adiabaticity. For the implicit solvent model, when the
129 system contains charged ions such as hydronium with strong solvation energy, the solvent
130 model has to be tuned carefully to yield a correct energy for the ions, so that the energetics
131 of the transition from free hydronium to *COH will be correct. Here, the continuum
132 polarizable solvent model is used with specific ion–solvent interaction parameters.^{22,51} The
133 solvent parameters of H and O (belonging to hydronium) are tuned to reproduce the sol-
134 vation energy of the charged ion. However, computing solvation energy of a charged ion

135 in water is a non-trivial task due to water fluctuations. Instead, we borrow the idea of
 136 the computational hydrogen electrode to compute the free energy of the ion with aqueous
 137 condition.⁵² Using hydronium as an example, the reaction $\text{H}_3\text{O}^+(\text{aq}) + e^- \longrightarrow \frac{1}{2}\text{H}_2(\text{g}) +$
 138 $\text{H}_2\text{O}(\text{aq})$ occurs spontaneously at potential $U=0$ V. Thus, the enthalpy of $\text{H}_3\text{O}^+(\text{aq})$ can
 139 be expressed as $H(\text{H}_3\text{O}^+(\text{aq})) = 1/2E(\text{H}_2(\text{g})) + E(\text{H}_2\text{O}(\text{g})) + G_s(\text{H}_2\text{O}) + 4.44$ eV. Here, H
 140 stands for enthalpy, and $G_s(\text{H}_2\text{O})$ is the water solvation energy 0.274 eV obtained from the
 141 experiment,⁵³ and 4.44 eV is the hydrogen electrode potential in terms of vacuum. Note,
 142 the explicit solvent model is used only to describe the enthalpy, instead of free energy of
 143 $\text{H}_3\text{O}^+(\text{aq})$, in agreement to the early work of implicit solvent model development.⁵²⁻⁵⁴ We
 144 tune the solvent parameters of H and O, so that the DFT calculated energy of the hydronium
 145 with implicit solvent model matches $H(\text{H}_3\text{O}^+(\text{aq}))$ obtained with the above formula.

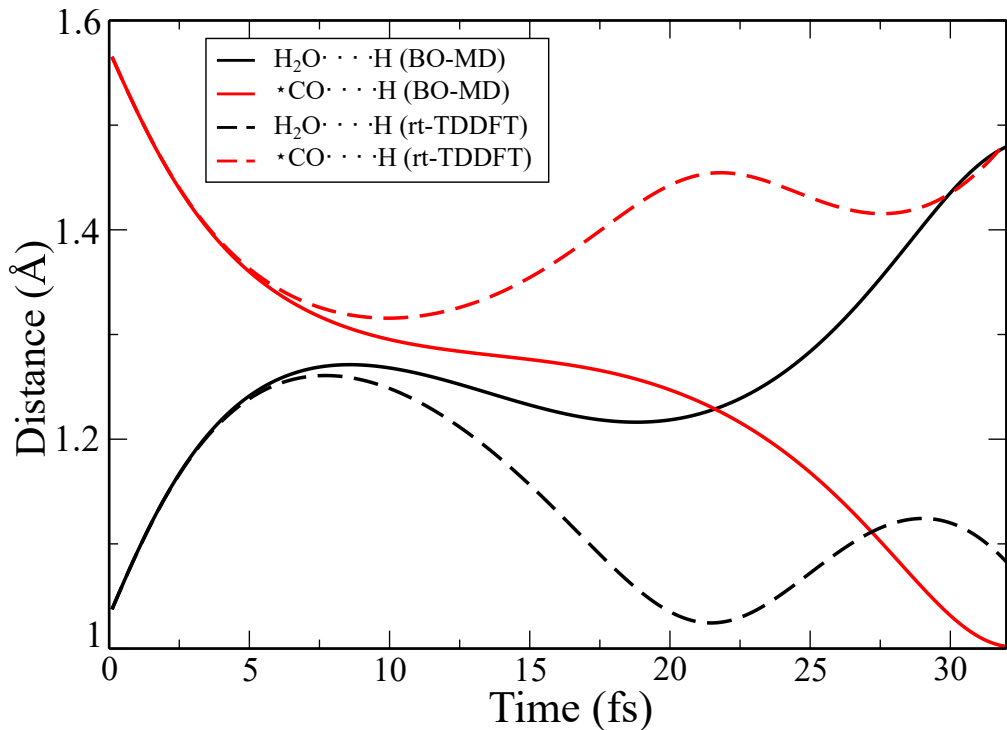


Figure 2: Reaction paths computed by adiabatic BO-MD and NA rt-TDDFT. It records the distance of the proton to oxygen of hydronium and the proton to oxygen of *CO. If the reaction proceeds, the black line and red line switch, indicating the proton transferring from hydronium to *CO. Otherwise, these two lines will return back.

146 Shown in Fig. 1 is the optimized initial and final structures. The initial structure is built

147 with hydronium close to $\star\text{CO}$ with a hydrogen bond, which is a local minimum structure.
148 Such hydrogen bond is optimized yielding a bond length around 1.6 Å. To simulate the
149 reaction with PA-ET using MD, an initial velocity is added to the hydrogen atom of the
150 hydronium close to $\star\text{CO}$, with the direction of velocity pointing to the oxygen of $\star\text{CO}$. By
151 tuning the magnitude of the initial velocity, we can monitor when the proton can transfer
152 from hydronium to $\star\text{CO}$ instead of returning. Such initial kinetic energy of the proton can
153 be treated as the reaction barrier. To find the initial atomic configuration and velocity for
154 this reaction to happen at the exact required kinetic energy, we have adopted a “reversed
155 process” procedure. In this procedure, the nudge elastic band (NEB) calculation is performed
156 first to reveal the reaction path and transition state. Then, by starting from the transition-
157 state structure with a very small initial velocities perturbation toward the initial reaction
158 direction, a short BO-MD is performed. This will yield an initial atomic structure. Starting
159 from this atomic position, with reversed velocity, the BO-MD will drive the system to the
160 transition state due to time inversion symmetry. Thus, a slight increase of the initial velocity
161 can lead to a transition to the final state. On the contrary, a slight reduction (e.g. 0.1%) in
162 velocity will prevent the reaction from happening. Using this way, we can quickly identify the
163 adiabatic reaction barrier using BO-MD. For the reaction $\star\text{CO} + \text{H}_3\text{O}^+ + e^- \longrightarrow \star\text{COH} +$
164 H_2O , the energy difference $\Delta E = E_{\text{final}} - E_{\text{initial}}$ is calculated to be around 0.5 eV. In the
165 experiment, an overpotential is added to overcome ΔE or to make it negative to make the
166 reaction to proceed spontaneous. To mimic the applied overpotential to the electrode, we add
167 two electrons to the system and relax the structures so that the energy difference between
168 the initial and final structures is close to zero. Fig. 2 shows the reaction paths computed
169 with BO-MD and rt-TDDFT. In this figure, both calculations of BO-MD and rt-TDDFT
170 start from the same initial structures and velocities as well as initial electronic structure.
171 The initial kinetic energy of the proton equals the BO-MD reaction barrier to just let the
172 reaction happen. For BO-MD simulation, the proton of hydronium move from H_3O^+ to $\star\text{CO}$
173 quickly at the beginning. Then it starts to slow down from 10 fs to 25 fs. Eventually it

174 bonds to $\star\text{CO}$ after around 30 fs indicated by the exchange of the distances toward CO and
 175 H_2O . We extend the simulation up to 70 fs to make sure the proton will not return back
 176 to water molecule. However, rt-TDDFT reveals a completely different reaction path. The
 177 proton follows almost the same reaction path of BO-MD at the beginning. But it deviates
 178 with the BO-MD's path after around 5 fs, proceeding to the opposite results in the end.
 179 During the simulation, the proton does not move across the reaction barrier, but it returns
 180 back to water molecules re-forming the initial structure. We also perform rt-TDDFT up to
 181 70 fs to confirm that the reaction does not happen during this time.

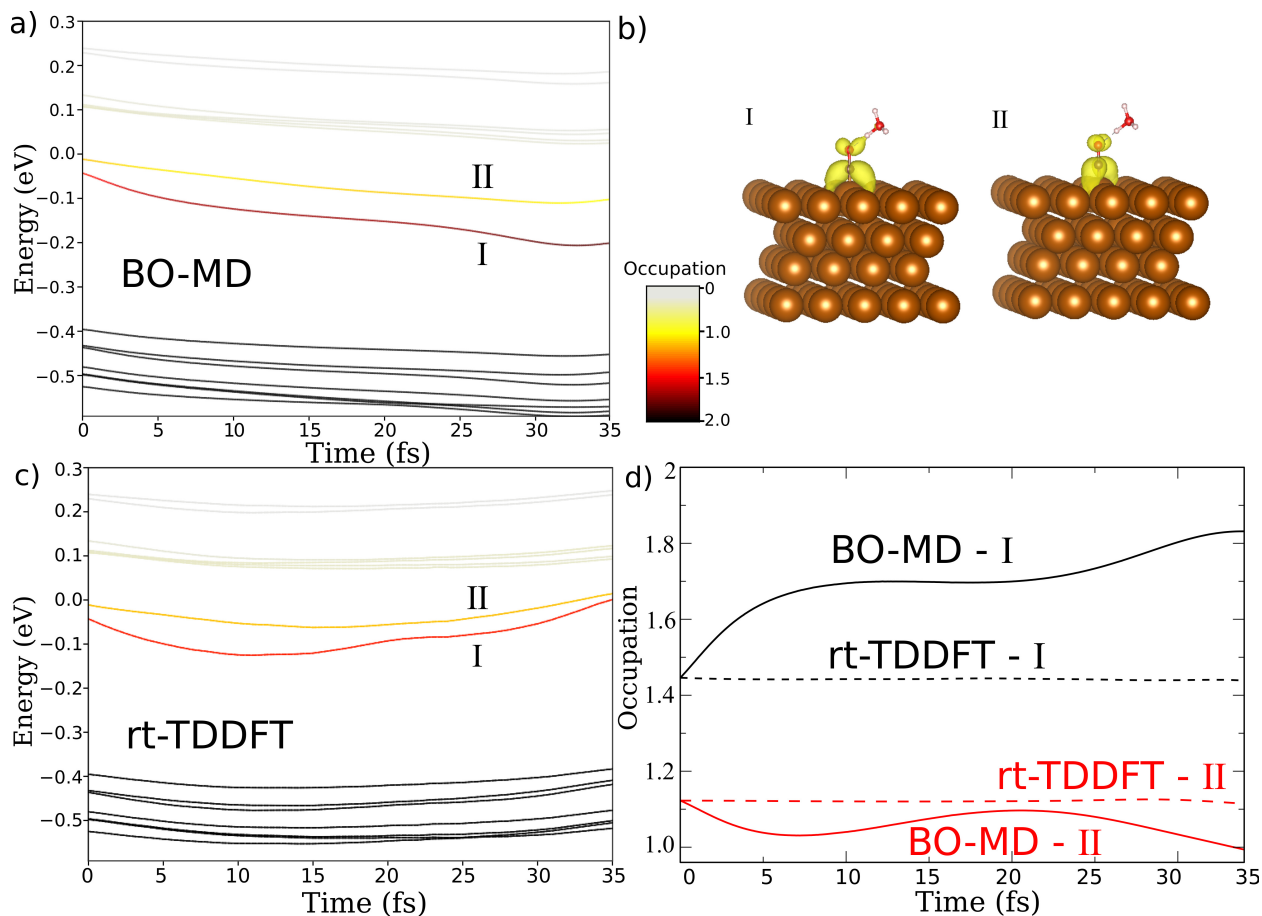


Figure 3: a) Eigen energies and occupations of the states near Fermi energy, extracted from BO-MD simulation. b) I: Charge density of the state at time $t=0$ fs with eigen energy around -0.01 eV. Its initial occupation is 1.1. II: Charge density for the state with eigen-energy around -0.04 eV at $t=0$ fs. It initial occupation is around 1.45. c) Eigen-energies and occupations of the states near Fermi energy, extracted from rt-TDDFT simulation with the same initial structure and velocity to BO-MD. d) Occupations of the adiabatic state (I and II) as a function of time for BO-MD and rt-TDDFT simulations.

182 To unveil the underlying reason for the dramatic difference, Fig. 3a shows the eigen-
183 energies for the states near the Fermi level in BO-MD simulation. The colors indicate the
184 occupation of the states during the reaction. Near the Fermi level, there are two eigen-
185 states with wavefunctions mostly on adsorbed CO on copper as shown in Fig. 3b. At time
186 $t = 0$ fs, these two states are almost degenerate except that they are splitted owing to
187 the weak hybridization with H_3O^+ . During the reaction (Fig. 3a), most of the states have
188 relatively small changes, except the state hybridized with hydronium near Fermi level. When
189 the proton is moving close to $\star\text{CO}$, the energy of state I becomes lower, indicating the
190 hybridization developed between the proton and $\star\text{CO}$. More importantly, we also track the
191 change of the occupation of this state as shown in Fig. 3d. Initially, at $t = 0$ and room
192 temperature, the state I is 72% occupied, while the state II is 55% occupied. As the reaction
193 goes, the occupation of state I rises until it is fully occupied. On the other hand, the
194 occupation of state II slightly reduces. The total occupation of 2.55 increases to about 3.0
195 (non-spin case). Thus, there are around 0.45 electrons increase on these two levels. Such
196 0.45 electrons increase indicates that the previously empty proton is occupied by electrons.
197 Enough charge occupation on H manifests the bond formation between H and CO. The
198 major part of the 0.45 electron transfer is provided from Cu slab. Such electron transfer
199 from Cu can be verified by a direct charge measurement before and after the reaction. With
200 a horizontal plane ($x-y$ plane) with its z -value in the middle between the top-layer Cu and
201 C atoms, the total electrons above this plane is found to increase by 0.35 after the reaction.
202 This is also consistent with the results reported in Ref. 21. It is interesting that this charge
203 is not 1. Under the computational hydrogen electrode (CHE) approximation, this charge
204 transfer should be 1.

205 The above picture is dramatically different in rt-TDDFT simulation. As shown in Fig. 3c,
206 at the beginning of the reaction, the state I and II change in similar way as in the BO-MD.
207 But after 15 fs, they become different. More dramatically, the occupations of state I and II
208 almost do not change during the simulation time. The occupation on the adiabatic state I

209 and II are calculated as: $f_j(t) = \sum_i |\langle \phi_j(t) | \psi_i(t) \rangle|^2 O(i) = \sum_i |C_{i,j}(t)|^2 O(i)$, where $O(i)$ is the
 210 occupation of the time evolution wavefunction $\psi_i(t)$ which does not change under rt-TDDFT.
 211 The charge on H is controlled by both the hybridization strength of the adiabatic CO-H state
 212 and the occupation for this state. If starting the simulation from same initial structure and
 213 velocities to BO-MD, the relative constant $f_j(t)$ for state I and II by rt-TDDFT leads to
 214 the situation that less charge is transferred to H from Cu, which suppresses the proton's
 215 motion towards CO and reduces the bond strength of CO-H bond eventually. As a result,
 216 there is no formation of CO-H bond (due to the lack of electrons), and the system bounces
 217 back to H_3O^+ as shown in Fig. 2. The lack of charge transfer is also verified by the direct
 218 charge measurement above the horizontal plane as discussed above. The change of charge
 219 from Cu is less than 0.35 compared to BO-MD (Fig. 4b). This example clearly shows how
 220 the non-adiabaticity plays a role in electrochemical reactions. Although this is only for one
 221 step, the observation is general since most of the reduction and oxidation reactions involves
 222 fast protonation or deprotonation.

223 For the rt-TDDFT simulation, the microscopic mechanism for the reaction becomes quite
 224 different from that of BO-MD. In order to induce the reaction, a higher initial velocity shall
 225 be provided. In this case, we find that at least 12 meV additional initial kinetic energy
 226 must be supplied, corresponding to 6.1% reaction barrier underestimation by BO-MD and
 227 other ground-state calculation methods. It is interesting to investigate how the reaction can
 228 happen if the occupations of adiabatic states tend to be constant. Shown in Fig. 4a compares
 229 two simulations (BO-MD and rt-TDDFT) with both giving rise to the reactions by just
 230 overcoming the barrier (thus rt-TDDFT has higher initial velocities than BO-MD). Similar
 231 to the above rt-TDDFT case which has the same velocities to BO-MD, the occupations of
 232 the state I and II in this rt-TDDFT simulation are mostly unchanged starting from $t = 0$ fs.
 233 However, for a given time during the reaction, the proton is closer to CO than that of BO-MD,
 234 owing to its higher initial velocity in TDDFT. Although the occupations of the adiabatic state
 235 I and II are constant, the adiabatic CO-H hybridization is stronger in rt-TDDFT because

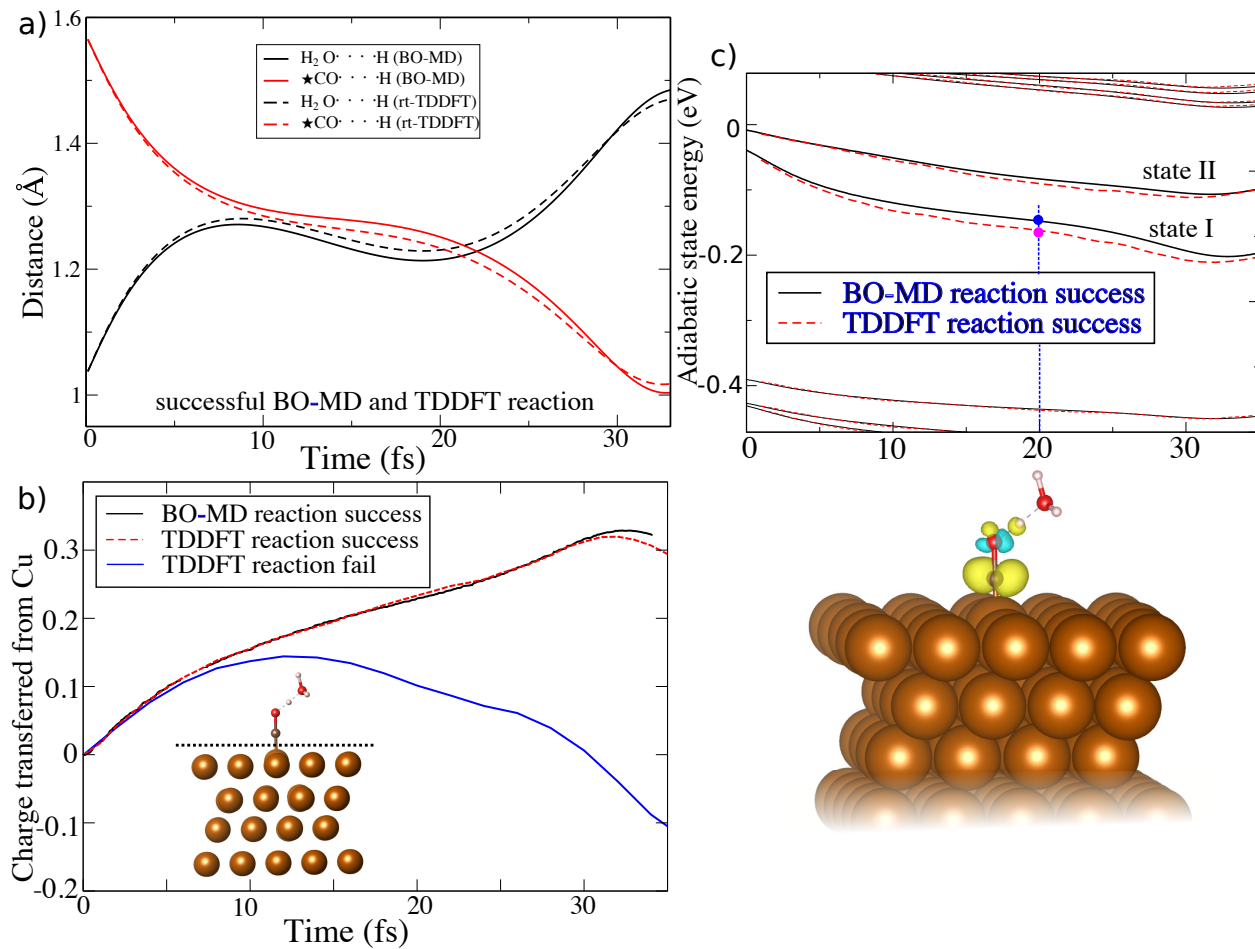


Figure 4: a) Reaction paths for BO-MD and rt-TDDFT. Both simulations have proton bonding to $\star\text{CO}$ to make the reaction successful (thus rt-TDDFT has a higher initial velocity than BO-MD). b) Measured change of total charge counted above the plane. This horizontal plane has its z -value in the middle between C and top Cu layer. Here, simulation-“TDDFT reaction fail” has the same initial velocity to “BO-MD reaction success”, while “TDDFT reaction success” has higher initial velocity than “BO-MD reaction success”. c) Eigen-energy of the adiabatic states for “BO-MD reaction success” and “TDDFT reaction success”. The bottom isosurface is the state I charge density difference of BO-MD and rt-TDDFT at $t = 20$ fs (charge density at “Red” dot minus “Blue” dot). Yellow color in the isosurface indicates positive; blue indicates negative.

236 of the closer distance between H and CO. This can be shown in Fig. 4c, where the charge
237 density difference of the adiabatic state I from TDDFT and BO-MD at 20 fs is plotted as an
238 example. It shows the electron gain near the proton for the adiabatic state I in rt-TDDFT.
239 Such stronger hybridization between CO and H compensates for the invariant occupation
240 in rt-TDDFT, transferring enough charge to the proton to form the CO-H bond and finish
241 the reaction. Meanwhile, the change of charge above Cu substrate is measured during this
242 rt-TDDFT simulation (Fig. 4b). Compared to BO-MD, rt-TDDFT (reaction success) shows
243 quite similar change of the charge out of Cu, and rt-TDDFT (reaction success) does not show
244 a slower charge transfer. Thus, we believe the charge transfer bottleneck is a less dominant
245 consequence of NA effect.

246 Finally, we examine the situation with spin-polarization. After turning on the spin, the
247 reaction path shows negligible difference compared to Fig. 2. We do note that, if the k -point
248 is not sufficient, in some cases, for the BO-MD simulation, after the proton exchange, the
249 system can become spin-polarized. We expect this could be a real case if CO is sitting in
250 small Cu cluster instead of bulk Cu (see SI Fig. 4). This spin-polarization however, will never
251 be developed in rt-TDDFT, since such spin flip is impossible without spin-orbit coupling.
252 Even with spin-orbit, the time of the reaction discussed here will not be enough to make
253 such spin flip.

254 As afore discussed, the implicit solvent model reproduces the energetic of the solvation
255 effect to ions. However, it does has its disadvantages,⁵⁵ primarily as an averaged contin-
256 uum media, it lacks the atomistic bonding information. More importantly, for dynamical
257 simulations, implicit solvent has instant dielectric screening response. But in reality, the
258 surrounding water will not have enough time to rotate itself and re-arrange the structure
259 following the fast proton transfer movement. Meanwhile, the surrounding water molecules
260 could form hydrogen bonds with hydronium or even with \star CO to change the energy lev-
261 els. To overcome this challenge, we utilize a hybrid solvent model by sampling an explicit
262 water molecules layer around the reaction site. Implicit solvent model is still used outside

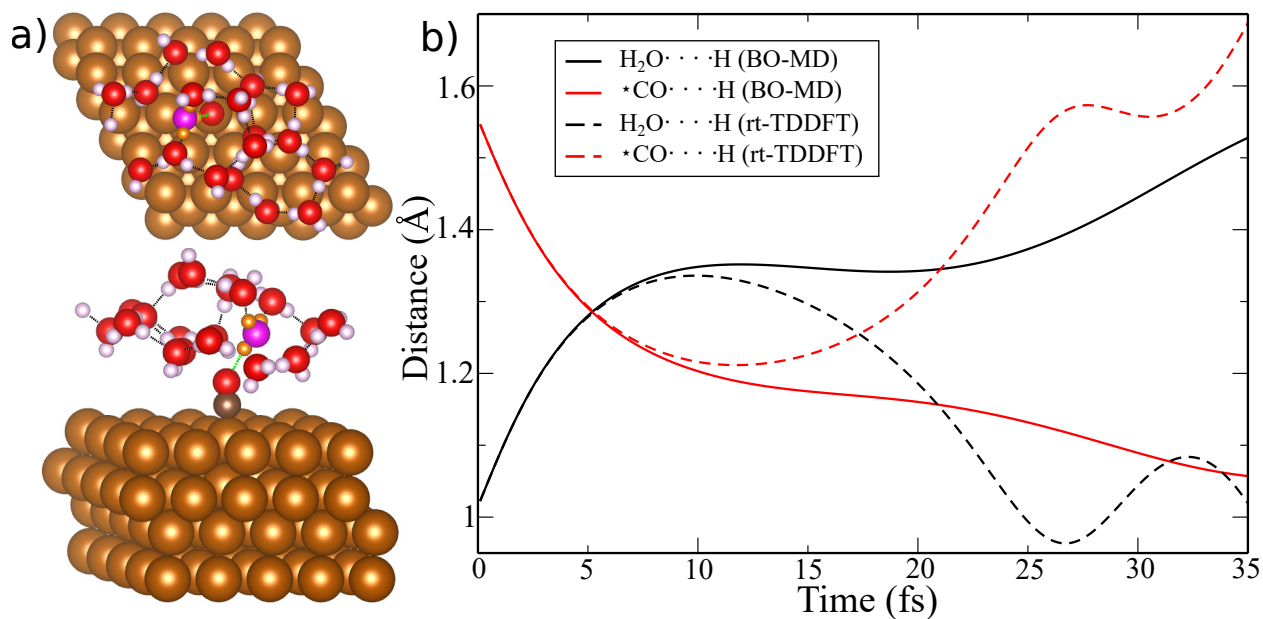


Figure 5: a) Structure of Cu/CO/H₃O⁺ with additional 14 H₂O molecules around reaction site. Top: Top view, Bottom: side view. Black and green dashed lines are hydrogen bonds between water molecules, hydrogen bonds between hydronium and *CO, respectively. Hydronium are highlighted with different colors (Violet: oxygen, Orange: Hydrogen) for clarity. b) Reaction paths simulated by BO-MD and rt-TDDFT with same initial structures and velocities. Similar to Fig. 2, it records the distance of the proton to oxygen of hydronium and the proton to oxygen of *CO.

263 the explicit solvent model layer. Obtaining the structure for the water molecules is not
264 trivial. Here, the in-house code based on the genetic algorithm is used to find the global
265 energy-minimum (see SI). Genetic-algorithm structure searching is analogous to the evolu-
266 tionary process in the biology. For a population consisting of finite number of structures,
267 the structures with lower free energies are more likely to be selected to combine into the
268 child generation, similar to the nature selection. By iterating such selection process from
269 the parent- to child-generation, it is possible to find out the global minimum given enough
270 number of generations. In this case, we add another 14 water molecules around $\star\text{CO}$ and
271 hydronium. $\star\text{CO}$ and hydronium are fixed during the evolutionary iterations (see SI). Shown
272 in Fig. 5a is the final structure obtained. To make the free energies of the initial and final
273 structures to be the same, 4 additional electrons are added to the system. Following the
274 same procedure for the implicit solvent model case, we perform ground-state NEB to find
275 the reaction path, reverse and tune the velocities to get BO-MD reaction barrier. Here, the
276 BO-MD or NEB calculated adiabatic barrier is higher than those with implicit solvent. This
277 is because at the transition state where the proton is in the middle of CO and H_2O , there
278 is a strong solvent polarization energy towards the relatively isolated proton in the implicit
279 model case. Such polarization energy does not exist in the explicit water molecule due to
280 the lack of response of the surrounding water molecules. rt-TDDFT is carried out with the
281 same initial condition that is used for BO-MD. However, the NA effect becomes important
282 near the reaction barrier similar to the implicit solvent case: rt-TDDFT and BO-MD yield
283 opposite results for the reaction as shown in Fig. 5b. The electronic structure's evolution
284 by BO-MD is illustrated in SI Fig.5, including their occupations. As the reaction goes, the
285 eigen-energy of the state is lowered indicating the development of the hybridization between
286 the proton and $\star\text{CO}$. Meanwhile, BO-MD predicts the increased occupation of this state.
287 But rt-TDDFT illustrates a constant value for the occupations (shown in SI Fig.5c), although
288 the energies of the adiabatic states is lowered owing to the hybridization. Eventually, the
289 reaction does not happen and it returns back to the initial structure.

290 Here, the number of added electrons is to mimic the applied overpotential. Meantime, we
 291 perform the calculation with 3 additional electrons. In this case, reducing one electron shifts
 292 up the relative free energy of the final $\star\text{COH}$ structure by around 0.2 eV. The calculated
 293 ground-state barrier is increased from 0.36 eV to 0.41 eV. Using BO-MD and rt-TDDFT,
 294 the reaction paths and the evolutions of the electronic structure including occupations are
 295 shown in SI SFig.6. These results indicate clearly that the NA effect still plays a role when
 296 the applied potential is altered. We also perform both spin-polarized and spin-unpolarized
 297 calculations. These two types of calculation give almost the same reaction path and eigen-
 298 energy/occupation change during the reaction, i.e. the system is always non-magnetic during
 299 the reaction. Table. 1 lists the reaction barrier calculated by the adiabatic methods and rt-
 300 TDDFT involving the non-adiabaticity. From Table. 1, we see that although the three cases
 301 (implicit solvent, explicit solvent model and different overpotential) have rather different
 302 barrier, the barrier increase due to NA effect are all similar around 10 meV.

Table 1: Reaction barriers calculated by NEB, BO-MD and rt-TDDFT. Here, NEB and BO-MD are only ground-state calculations. rt-TDDFT involves the NA effect beyond the ground-state approximation. Last column is the percentage change of the barrier by the NA effect.

Reaction Barrier E	E_{NEB} (eV)	$E_{\text{BO-MD}}$ (eV)	$E_{\text{rt-TDDFT}}$ (eV)	$E_{\text{rt-TDDFT}} - E_{\text{BO-MD}}$ (meV)
Implicit solvent (add $2e^-$)	0.080	0.196	0.208	12
Hybrid solvent (add $4e^-$)	0.360	0.288	0.299	11
Hybrid solvent (add $3e^-$)	0.411	0.363	0.373	10

303 To summarize, using one step of CO_2 reduction on copper [111] surface ($\star\text{CO} + \text{H}_3\text{O}^+ +$
 304 $e^- \longrightarrow \star\text{COH} + \text{H}_2\text{O}$) as an example, we investigate how the NA effect is involved to
 305 influence the reaction. We believe this is one of the first few works to directly illuminate
 306 the NA effect in electrochemical reaction with the electrolytes. In this reaction, the proton
 307 of hydronium is attacking $\star\text{CO}$ to form $\star\text{COH}$. By tuning initial velocity of the proton and
 308 monitoring the reaction using ground-state BO-MD, we can identify the adiabatic reaction
 309 barrier to be the initial kinetic energy of the proton, which just let the reaction to finish.
 310 However, by using the same initial kinetic energy and structure, although BO-MD can finish

311 the reaction, rt-TDDFT simulation involving the NA effect disallow the reaction to finish
312 but return the proton back to hydronium. A higher kinetic energy must be supplied to drive
313 the proton move over the barrier to form the final structure. Additional electrons are added
314 to the system to mimic the applied overpotential to the electrode. Both implicit continuum
315 solvent and explicit water solvent are used to simulate the same reaction. However, the NA
316 effect still remains in all the case. Our calculation demonstrates that involving the NA effect
317 increases the reaction barrier by 10 meV for all the models and electrode potentials we have
318 tested.

319 Acknowledgement

320 This material is based on the work performed by the Joint Center for Artificial Photosyn-
321 thesis, a DOE Energy Innovation Hub, supported through the Office of Science of the U.S.
322 Department of Energy under Award number DE-SC0004993. We use the resource of National
323 Energy Research Scientific Computing center (NERSC) located in Lawrence Berkeley Na-
324 tional Laboratory and the computational resource of the Oak Ridge Leadership Computing
325 Facility at the Oak Ridge National Laboratory under the Innovative and Novel Computa-
326 tional Impact on Theory and Experiment project.

327 References

- 328 (1) Valdés, Á.; Qu, Z.-W.; Kroes, G.-J.; Rossmeisl, J.; Nørskov, J. K. Oxidation and Photo-
329 Oxidation of Water on TiO₂ Surface. *The Journal of Physical Chemistry C* **2008**, *112*,
330 9872–9879.
- 331 (2) Pham, T. A.; Ping, Y.; Galli, G. Modelling Heterogeneous Interfaces for Solar Water
332 Splitting. *Nature Materials* **2017**, *16*, 401–408.
- 333 (3) Pham, H. H.; Cheng, M.-J.; Frei, H.; Wang, L.-W. Surface Proton Hopping and Fast-

- 334 Kinetics Pathway of Water Oxidation on Co_3O_4 (001) Surface. *ACS Catalysis* **2016**,
335 *6*, 5610–5617.
- 336 (4) Wu, Y.; Lazic, P.; Hautier, G.; Persson, K.; Ceder, G. First Principles High Throughput
337 Screening of Oxynitrides for Water-Splitting Photocatalysts. *Energy & Environmental*
338 *Science* **2013**, *6*, 157–168.
- 339 (5) Zhou, Y.; Gao, G.; Li, Y.; Chu, W.; Wang, L.-W. Transition-Metal Single Atoms in
340 Nitrogen-Doped Graphenes as Efficient Active Centers for Water Splitting: A Theoret-
341 ical Study. *Physical Chemistry Chemical Physics* **2019**, *21*, 3024–3032.
- 342 (6) Greeley, J.; Jaramillo, T. F.; Bonde, J.; Chorkendorff, I.; Nørskov, J. K. Computational
343 High-Throughput Screening of Electrocatalytic Materials for Hydrogen Evolution. *Nature*
344 *Materials* **2006**, *5*, 909–913.
- 345 (7) Chhetri, M.; Maitra, S.; Chakraborty, H.; V. Waghmare, U.; R. Rao, C. N. Superior
346 Performance of Borocarbonitrides, $\text{B}_x\text{C}_y\text{N}_z$, as Stable, Low-Cost Metal-Free
347 Electrocatalysts for the Hydrogen Evolution Reaction. *Energy & Environmental Science*
348 **2016**, *9*, 95–101.
- 349 (8) Tang, Q.; Jiang, D.-e. Mechanism of Hydrogen Evolution Reaction on 1T-MoS₂ from
350 First Principles. *ACS Catalysis* **2016**, *6*, 4953–4961.
- 351 (9) Gao, G.; O’Mullane, A. P.; Du, A. 2D MXenes: A New Family of Promising Catalysts
352 for the Hydrogen Evolution Reaction. *ACS Catalysis* **2017**, *7*, 494–500.
- 353 (10) A. Peterson, A.; Abild-Pedersen, F.; Studt, F.; Rossmeisl, J.; K. Nørskov, J. How
354 Copper Catalyzes the Electroreduction of Carbon Dioxide into Hydrocarbon Fuels.
355 *Energy & Environmental Science* **2010**, *3*, 1311–1315.
- 356 (11) Montoya, J. H.; Peterson, A. A.; Nørskov, J. K. Insights into C-C Coupling in CO₂
357 Electroreduction on Copper Electrodes. *ChemCatChem* **2013**, *5*, 737–742.

- 358 (12) Cheng, T.; Xiao, H.; Goddard, W. A. Free-Energy Barriers and Reaction Mechanisms
359 for the Electrochemical Reduction of CO on the Cu(100) Surface, Including Multiple
360 Layers of Explicit Solvent at pH 0. *The Journal of Physical Chemistry Letters* **2015**,
361 *6*, 4767–4773.
- 362 (13) Cheng, T.; Xiao, H.; Goddard, W. A. Reaction Mechanisms for the Electrochemical
363 Reduction of CO₂ to CO and Formate on the Cu(100) Surface at 298 K from Quantum
364 Mechanics Free Energy Calculations with Explicit Water. *Journal of the American*
365 *Chemical Society* **2016**, *138*, 13802–13805.
- 366 (14) Ping, Y.; Goddard, W. A.; Galli, G. A. Energetics and Solvation Effects at the Pho-
367 toanode/Catalyst Interface: Ohmic Contact versus Schottky Barrier. *Journal of the*
368 *American Chemical Society* **2015**, *137*, 5264–5267.
- 369 (15) Scheuermann, A. G.; Lawrence, J. P.; Kemp, K. W.; Ito, T.; Walsh, A.; Chidsey, C.
370 E. D.; Hurley, P. K.; McIntyre, P. C. Design Principles for Maximizing Photovoltage
371 in Metal-Oxide-Protected Water-Splitting Photoanodes. *Nature Materials* **2016**, *15*,
372 99–105.
- 373 (16) Yan, Q.; Yu, J.; Suram, S. K.; Zhou, L.; Shinde, A.; Newhouse, P. F.; Chen, W.;
374 Li, G.; Persson, K. A.; Gregoire, J. M.; Neaton, J. B. Solar Fuels Photoanode Materials
375 Discovery by Integrating High-Throughput Theory and Experiment. *Proceedings of the*
376 *National Academy of Sciences* **2017**, *114*, 3040–3043.
- 377 (17) Nørskov, J. K.; Rossmeisl, J.; Logadottir, A.; Lindqvist, L.; Kitchin, J. R.; Bligaard, T.;
378 Jónsson, H. Origin of the Overpotential for Oxygen Reduction at a Fuel-Cell Cathode.
379 *The Journal of Physical Chemistry B* **2004**, *108*, 17886–17892.
- 380 (18) Berne, B. J.; Ciccotti, G.; Coker, D. F. [No Title Found]. Classical and Quantum
381 Dynamics in Condensed Phase Simulations. LERICI, Villa Marigola, 1998.

- 382 (19) Rossmeisl, J.; Chan, K.; Ahmed, R.; Tripković, V.; E. Björketun, M. pH in Atomic
383 Scale Simulations of Electrochemical Interfaces. *Physical Chemistry Chemical Physics*
384 **2013**, *15*, 10321–10325.
- 385 (20) Ringe, S.; Clark, E. L.; Resasco, J.; Walton, A.; Seger, B.; Bell, A. T.; Chan, K. Under-
386 standing Cation Effects in Electrochemical CO₂ Reduction. *Energy & Environmental*
387 *Science* **2019**, *12*, 3001–3014.
- 388 (21) Bajdich, M.; Fields, M.; Chen, L. D.; Sandberg, R. B.; Chan, K.; Nørskov, J. K.
389 Electron Transfer to CO₂ during Adsorption at the Metal — Solution Interface.
- 390 (22) Andreussi, O.; Dabo, I.; Marzari, N. Revised Self-Consistent Continuum Solvation in
391 Electronic-Structure Calculations. *The Journal of Chemical Physics* **2012**, *136*, 064102.
- 392 (23) Mathew, K.; Sundararaman, R.; Letchworth-Weaver, K.; Arias, T. A.; Hennig, R. G.
393 Implicit Solvation Model for Density-Functional Study of Nanocrystal Surfaces and
394 Reaction Pathways. *The Journal of Chemical Physics* **2014**, *140*, 084106.
- 395 (24) Tritsaris, G. A.; Vinichenko, D.; Kolesov, G.; Friend, C. M.; Kaxiras, E. Dynamics of
396 the Photogenerated Hole at the Rutile TiO₂(110)/Water Interface: A Nonadiabatic
397 Simulation Study. *The Journal of Physical Chemistry C* **2014**, *118*, 27393–27401.
- 398 (25) Hammes-Schiffer, S. Theory of Proton-Coupled Electron Transfer in Energy Conversion
399 Processes. *Accounts of chemical research* **2009**, *42*, 1881–1889.
- 400 (26) Wang, Z.; Li, S.-S.; Wang, L.-W. Efficient Real-Time Time-Dependent Density Func-
401 tional Theory Method and Its Application to a Collision of an Ion with a 2D Material.
402 *Physical Review Letters* **2015**, *114*.
- 403 (27) Bi, G.; Kang, J.; Wang, L.-W. High Velocity Proton Collision with Liquid Lithium:
404 A Time Dependent Density Functional Theory Study. *Physical Chemistry Chemical*
405 *Physics* **2017**, *19*, 9053–9058.

- 406 (28) Yost, D. C.; Yao, Y.; Kanai, Y. Examining Real-Time Time-Dependent Density Func-
407 tional Theory Nonequilibrium Simulations for the Calculation of Electronic Stopping
408 Power. *Physical Review B* **2017**, *96*.
- 409 (29) Wang, D.; Liu, Z.-P.; Yang, W.-M. Proton-Promoted Electron Transfer in Photocatal-
410 ysis: Key Step for Photocatalytic Hydrogen Evolution on Metal/Titania Composites.
411 *ACS Catalysis* **2017**, *7*, 2744–2752.
- 412 (30) Alducin, M.; Díez Muiño, R.; Juaristi, J. Non-Adiabatic Effects in Elementary Reaction
413 Processes at Metal Surfaces. *Progress in Surface Science* **2017**, *92*, 317–340.
- 414 (31) Kroes, G.-J.; Juaristi, J. I.; Alducin, M. Vibrational Excitation of H₂ Scattering from
415 Cu(111): Effects of Surface Temperature and of Allowing Energy Exchange with the
416 Surface. *The Journal of Physical Chemistry C* **2017**, *121*, 13617–13633.
- 417 (32) Jiang, B.; Alducin, M.; Guo, H. Electron–Hole Pair Effects in Polyatomic Dissociative
418 Chemisorption: Water on Ni(111). *The Journal of Physical Chemistry Letters* **2016**,
419 *7*, 327–331.
- 420 (33) Luo, X.; Jiang, B.; Juaristi, J. I.; Alducin, M.; Guo, H. Electron-Hole Pair Effects
421 in Methane Dissociative Chemisorption on Ni(111). *The Journal of Chemical Physics*
422 **2016**, *145*, 044704.
- 423 (34) Füchsel, G.; del Cueto, M.; Díaz, C.; Kroes, G.-J. Enigmatic HCl + Au(111) Reaction:
424 A Puzzle for Theory and Experiment. *The Journal of Physical Chemistry C* **2016**, *120*,
425 25760–25779.
- 426 (35) Goikoetxea, I.; Juaristi, J. I.; Alducin, M.; Muiño, R. D. Dissipative Effects in the
427 Dynamics of N₂ on Tungsten Surfaces. *Journal of Physics: Condensed Matter* **2009**,
428 *21*, 264007.

- 429 (36) Carbogno, C.; Behler, J.; Reuter, K.; Groß, A. Signatures of Nonadiabatic
430 O_2 Dissociation at Al(111): First-Principles Fewest-Switches Study.
431 *Physical Review B* **2010**, *81*, 035410.
- 432 (37) Juaristi, J. I.; Alducin, M.; Muiño, R. D.; Busnengo, H. F.; Salin, A. Role of Electron-
433 Hole Pair Excitations in the Dissociative Adsorption of Diatomic Molecules on Metal
434 Surfaces. *Physical Review Letters* **2008**, *100*, 116102.
- 435 (38) Diekhöner, L.; Hornekær, L.; Mortensen, H.; Jensen, E.; Baurichter, A.;
436 Petrunin, V. V.; Luntz, A. C. Indirect Evidence for Strong Nonadiabatic Coupling in
437 N_2 Associative Desorption from and Dissociative Adsorption on Ru(0001). *The Journal*
438 *of Chemical Physics* **2002**, *117*, 5018–5030.
- 439 (39) Fischer, S. A.; Duncan, W. R.; Prezhdo, O. V. Ab Initio Nonadiabatic Molecular Dy-
440 namics of Wet-Electrons on the TiO_2 Surface. *Journal of the American Chemical So-*
441 *ciety* **2009**, *131*, 15483–15491.
- 442 (40) Petek, H.; Zhao, J. Ultrafast Interfacial Proton-Coupled Electron Transfer. *Chemical*
443 *Reviews* **2010**, *110*, 7082–7099.
- 444 (41) Oscar, B. G.; Liu, W.; Rozanov, N. D.; Fang, C. Ultrafast Intermolecular Proton Trans-
445 fer to a Proton Scavenger in an Organic Solvent. *Physical Chemistry Chemical Physics*
446 **2016**, *18*, 26151–26160.
- 447 (42) Turro, N. J. *Modern Molecular Photochemistry*; University Science Books, 1991.
- 448 (43) Hammes-Schiffer, S. Theoretical Perspectives on Proton-Coupled Electron Transfer Re-
449 actions. *Accounts of Chemical Research* **2001**, *34*, 273–281.
- 450 (44) Miyamoto, Y.; Zhang, H. Electronic Excitation in an Ar^{7+} Ion
451 Traversing a Graphene Sheet: Molecular Dynamics Simulations. *Physical Review B*
452 **2008**, *77*, 161402.

- 453 (45) Jia, W.; Cao, Z.; Wang, L.; Fu, J.; Chi, X.; Gao, W.; Wang, L.-W. The Analysis of
454 a Plane Wave Pseudopotential Density Functional Theory Code on a GPU Machine.
455 *Computer Physics Communications* **2013**, *184*, 9–18.
- 456 (46) Jia, W.; Fu, J.; Cao, Z.; Wang, L.; Chi, X.; Gao, W.; Wang, L.-W. Fast Plane Wave
457 Density Functional Theory Molecular Dynamics Calculations on Multi-GPU Machines.
458 *Journal of Computational Physics* **2013**, *251*, 102–115.
- 459 (47) Hamann, D. R. Optimized Norm-Conserving Vanderbilt Pseudopotentials. *Physical Re-*
460 *view B* **2013**, *88*, 085117.
- 461 (48) Kresse, G.; Gil, A.; Sautet, P. Significance of Single-Electron Energies for the Descrip-
462 tion of CO on Pt(111). *Physical Review B* **2003**, *68*.
- 463 (49) Mason, S. E.; Grinberg, I.; Rappe, A. M. First-Principles Extrapolation Method for
464 Accurate CO Adsorption Energies on Metal Surfaces. *Physical Review B* **2004**, *69*.
- 465 (50) Hammer, B.; Hansen, L. B.; Nørskov, J. K. Improved Adsorption Energetics within
466 Density-Functional Theory Using Revised Perdew-Burke-Ernzerhof Functionals. *Phys-*
467 *ical Review B* **1999**, *59*, 7413–7421.
- 468 (51) Mathew, K.; Sundararaman, R.; Letchworth-Weaver, K.; Arias, T. A.; Hennig, R. G.
469 Implicit Solvation Model for Density-Functional Study of Nanocrystal Surfaces and
470 Reaction Pathways. *The Journal of Chemical Physics* **2014**, *140*, 084106.
- 471 (52) Bryantsev, V. S.; Diallo, M. S.; Goddard III, W. A. Calculation of Solvation Free
472 Energies of Charged Solutes Using Mixed Cluster/Continuum Models. *The Journal of*
473 *Physical Chemistry B* **2008**, *112*, 9709–9719.
- 474 (53) Zhan, C.-G.; Dixon, D. A. First-Principles Determination of the Absolute Hydration
475 Free Energy of the Hydroxide Ion. *The Journal of Physical Chemistry A* **2002**, *106*,
476 9737–9744.

- 477 (54) Palascak, M. W.; Shields, G. C. Accurate Experimental Values for the Free Energies of
478 Hydration of H⁺, OH⁻, and H₃O⁺. *The Journal of Physical Chemistry A* **2004**, *108*,
479 3692–3694.
- 480 (55) Gauthier, J. A.; Ringe, S.; Dickens, C. F.; Garza, A. J.; Bell, A. T.; Head-Gordon, M.;
481 Nørskov, J. K.; Chan, K. Challenges in Modeling Electrochemical Reaction Energetics
482 with Polarizable Continuum Models. *ACS Catalysis* **2019**, *9*, 920–931.