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### Title

Self-heating and electrical performance of carbon nanotube-enhanced cement composites

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<https://escholarship.org/uc/item/83t226np>

### Authors

Lee, Heeyoung

Yu, Wonjun

Loh, Kenneth J

et al.

### Publication Date

2020-07-01

### DOI

10.1016/j.conbuildmat.2020.118838

Peer reviewed

1 **Self-heating and electrical performance of carbon nanotube-enhanced**  
2 **cement composites**

3  
4 **Heeyoung Lee, Ph.D.**

5 Assistant Professor

6 Department of Civil Engineering, Chosun University  
7 Telephone: 82-62-230-7087 Fax: 82-62-608-5216  
8 E-mail: heeyoung0908@chosun.ac.kr  
9

10  
11 **Wonjun Yu**

12 Graduate Student

13 Kyung Hee University  
14 Telephone: 82-31-201-3897 Fax: 82-31-202-8854  
15 E-mail: 1208wonjun@gmail.com  
16

17  
18 **Kenneth J Loh, Ph.D. (Co-corresponding Author)**

19 Professor

20 University of California-San Diego, Department of Structural Engineering  
21 Telephone: 1-858-822-0431 Fax: 1-858-822-0430  
22 E-mail: kenloh@ucsd.edu  
23

24  
25 and  
26

27  
28 **Wonseok Chung, Ph.D., P.E. (Corresponding Author)**

29 Professor

30 Telephone: 82-31-201-2550 Fax: 82-31-202-8854  
31 E-mail: wschung@khu.ac.kr  
32

33  
34 Department of Civil Engineering  
35 College of Engineering  
36 Kyung Hee University  
37 1732 Deokyoung-Daero, Giheung-Gu  
38 Yongin-Si, Gyeonggi-Do 17104  
39

40 Republic of Korea  
41

42 This Paper has been Submitted for  
43 Publication in  
44

45 ***Construction and Building Materials***

46 **Self-heating and electrical performance of carbon nanotube-enhanced cement**  
47 **composites**

48 Heeyoung Lee<sup>a</sup>, Wonjun Yu<sup>b</sup>, Kenneth J. Loh<sup>c\*</sup>, and Wonseok Chung<sup>b\*</sup>

49 <sup>a</sup>Chosun University

50 <sup>b</sup>Kyung Hee University

51 <sup>c</sup>University of California-San Diego

52  
53 \* Corresponding Author: Department of Structural Engineering, University of California-San Diego,  
54 9500 Gilman Drive MC 0085, La Jolla, CA 92093-0085, USA

55 Tel.: 1-858-822-0431

56 Fax: 1-858-822-0430

57 E-mail: kenloh@ucsd.edu

58  
59 \* Corresponding Author: Department of Civil Engineering, College of Engineering, Kyung Hee  
60 University, 1732 Deokyoung-Daero, Giheung-Gu, Yongin-Si, Gyeonggi-Do 17104

61 Tel.: 82-31-201-2550

62 Fax: 82-31-202-8854

63 E-mail: wschung@khu.ac.kr

64  
65 **Abstract**

66 This work hypothesizes that mixing carbon nanotubes with cement improves the thermal and  
67 electrical properties of bulk cement composites. To test this, two different methods of  
68 combining cement and dispersed multi-walled carbon nanotubes (MWCNTs) were considered.  
69 In the first method, cement composites were produced by adding a dispersion of MWCNTs to  
70 cement. In the second, MWCNT-based thin films were spray-coated and combined with  
71 cement to produce cement composites. A third group of specimens was produced using both  
72 MWCNT dispersions and MWCNT film-coated. The experimental parameters considered were  
73 the mixing method, MWCNT concentrations, number of curing days, and voltages applied.  
74 Furthermore, field emission scanning electron microscopy revealed that the MWCNTs were  
75 evenly dispersed within the composites and formed a percolated network. Additionally, X-ray  
76 diffraction analysis confirmed that the products formed during hydration of the composites (i.e.,  
77 C-H and C-S-H) were the same as those generated using ordinary mortar. Upon testing these  
78 mortar-based specimens, it was found that the cement composites formed using a combination  
79 of MWCNT dispersion and MWCNT-based films exhibited the highest heating performance  
80 and lowest electrical resistance. Finally, thermal imaging showed that increased MWCNT  
81 concentrations during specimen casting led to a corresponding increase in their surface  
82 temperature upon voltage application.

83 **Keywords:** cement composite, heating, microstructure, multi-walled carbon nanotube,  
84 nanocomposite, temperature performance

85 **1. Introduction**

86 Various technologies for designing and constructing large-scale concrete structures have  
87 been developed in recent years. Concrete materials are regularly used in various structures  
88 worldwide because of their excellent strength and durability. However, the use of concrete also  
89 leads to various problems related to the construction environment and global warming.  
90 Moreover, concrete structures can experience significant damage in the winter because of the  
91 black ice caused by snow. Thus, multifunctional concrete with improved strength and  
92 durability that can melt snow and ice must be developed to address the above-mentioned  
93 problems related to existing concrete structures.

94 Multifunctional concrete can potentially be synthesized by incorporating nanomaterials in  
95 existing construction materials [1-3]. Nanomaterials, which have particle sizes ranging from  
96 0.1 to 100 nm, are attracting significant attention in various areas because of their high specific  
97 surface area per unit weight and excellent properties [4-7]. Nanomaterials can be mixed with  
98 construction materials based on ordinary cement to improve the mechanical, electrical, and  
99 thermal performances of the materials. Carbon nanotubes (CNTs), whose thermal conductivity  
100 and electrical conductivity are 7.5 and 100 times higher, respectively, than those of copper,  
101 have been mixed successfully with cement-based materials [8-12]. Further, there have been  
102 numerous studies on the mechanical and physical characteristics of cement composites formed  
103 using carbon nanomaterials to improve the strength, electrical conductivity, and heating  
104 performance [13-18].

105 Li and Zhao [13] fabricated specimens of ordinary mortar and its composites with  
106 multiwalled CNTs (MWCNTs) and carbon nanofibers (CNFs) and studied their compressive  
107 strengths. The results showed that the compressive strength of the composite with MWCNTs  
108 at 28 days was 19% higher than that of the composite consisting of ordinary mortar and CNFs.  
109 Further, the load transfer efficiency increased as more MWCNTs were added to the ordinary  
110 mortar because this resulted in a greater number of chemical bonds between the cement  
111 hydrates (C-S-H and calcium hydroxide). Chaipanich et al. [14] analyzed the compressive  
112 strength of cement mortar in which CNTs and fly ash had been mixed. The CNTs were added  
113 in concentrations of 0.5 and 1.0 wt% (cement weight). The results indicated that, for the same  
114 fly ash content, the compressive strength of the mortar increased with an increase in the CNT  
115 concentration because both the internal density and the extent of the CNT networks within the  
116 composite increased. Morsy et al. [15] fabricated mortar composites containing clay with  
117 nanosize particles and MWCNTs and studied their compressive strengths. The clay content  
118 was fixed at 6% of the cement weight and the MWCNT concentration was varied and  
119 composites with dimensions of 50 mm × 50 mm × 50 mm were fabricated. The composite with  
120 0.02 wt% (by cement weight) MWCNTs exhibited a compressive strength 11% higher than  
121 that of ordinary mortar. On the other hand, the composite with 0.1 wt% MWCNT showed a  
122 lower compressive strength. Hamzaoui et al. [16] studied the mechanical properties of mortar  
123 and concrete containing CNTs, measuring the compressive strength of the composites  
124 containing CNTs in different concentrations. The compressive strength was the highest when  
125 CNTs were added in amounts of 0.01 and 0.003% by cement weight. The compressive strength  
126 increased with the addition of the CNTs because they served as bridges between the pores and  
127 cracks. Choi et al. [17] fabricated cement mortar by mixing dispersions of MWCNTs in  
128 distilled water and studied its compressive strength. It was found that the use of distilled water  
129 to form the MWCNT dispersions had little effect on the dispersibility of the CNTs. Kang et al.  
130 [18] studied the effects of the dispersibility of CNTs on the compressive and tensile strengths

131 of cement composites. CNTs were dispersed using surfactants, high-performance plasticizers,  
132 and an ultrasonic treatment. The results showed that the use of both ultrasonic treatment and  
133 high-performance plasticizers improved the compressive and tensile strengths by 10% as  
134 compared to the other dispersion methods.

135 As for studies on the electrical and heating performances of composites formed using  
136 nanomaterials, Nan et al. [19] elucidated the relationship between the CNT content and  
137 improvements in thermal conductivity for composites containing CNTs. Liu et al. [20] studied  
138 the thermal conductivity of nanofluids containing MWCNTs and found that the thermal  
139 conductivity of the nanofluids increased linearly with MWCNT concentration. Li et al. [21]  
140 used CNTs dispersed in sulfuric acid and nitric acid as well as unmixed CNTs to fabricate  
141 composite specimens with dimensions of 40 mm × 160 mm × 40 mm and measured their  
142 electrical resistances. Both the composite containing CNTs dispersed in sulfuric acid and nitric  
143 acid and that containing the unmixed CNTs exhibited significantly reduced electrical  
144 resistances, because the CNTs were uniformly dispersed and formed networks within the  
145 composites. Zhang and Li [22] studied the road deicing performances of MWCNT-containing  
146 cement composites. **The results showed that a composite with a thermal conductivity of 2.83**  
147 **W/m·K was formed when the MWCNTs were mixed in a concentration of 3% by the cement**  
148 **weight.** Thus, it is possible to melt the ice formed on roads using MWCNT–cement composites.  
149 Kim et al. [23] studied the mechanical and electrical properties of cement composites  
150 containing silica fume and CNTs. It was found that the cement composites with low silica fume  
151 and CNT contents exhibited improved mechanical and electrical properties, owing to the  
152 decomposition of the aggregated CNTs into small clusters. Konst and Aza [24] analyzed the  
153 electrical properties and piezoresistive sensitivity of cement composites containing CNTs and  
154 CNFs. **The cement composite formed using both CNTs and CNFs in concentrations of 0.1%**  
155 **by the cement weight.** exhibited the highest piezoresistive sensitivity. Lee et al. [25] fabricated  
156 cement composites using MWCNTs and studied their electrical and thermal properties. The  
157 effects of the mixing method and MWCNT concentration were elucidated. The mixing methods  
158 used were forming a CNT coating on the fine aggregates and mixing the CNTs using a  
159 dispersion. **The MWCNTs were added in concentrations of 0.125 and 0.25 wt% (by cement**  
160 **weight).** The results showed that the composite formed using the former method exhibited  
161 better heating performance. Lee et al. [26] studied the heating performances of CNT-based  
162 cement composites. Different types of CNTs were used in varying concentrations, with the  
163 applied voltage also being a parameter. Specifically, SWCNTs and MWCNTs were used in  
164 combined concentrations of 0.0625 and 0.125 % (cement weight) while the voltages used were  
165 50 and 100 V. **The composite containing the MWCNTs and SWCNTs with a total**  
166 **concentration of 0.125% (by cement weight) exhibited better heating performance than that**  
167 **containing the CNTs in a total concentration of 0.0625 wt%.** Thus, the heating performance  
168 increased with the total CNT concentration. Moreover, when a voltage of 100 V was supplied  
169 to the 0.125 wt% composite, its temperature rose by 70.6 °C. Thus, this composite exhibited  
170 better heating performance than the 0.12 wt% composite.

171 Finally, considering studies on nanomaterial-coated films, Hone et al. [27] fabricated an  
172 SWCNT film by filtration and desorption through a magnetic field and studied its heating  
173 performance. The SWCNT film exhibited a thermal conductivity of 200 W/m·K, which was  
174 similar to those of graphite and highly crystalline diamond. Haung et al. [28] studied the  
175 thermal conductivity of CNT films exhibiting different arrangements. For the same CNT  
176 concentration, the thermal conductivity was higher when the CNTs were arranged in the same  
177 direction. Kim et al. [29] fabricated CNT films with dimensions of 10 mm × 10 mm by

178 electrostatic spray deposition without using any binders and studied their electrical  
179 conductivity. The CNT films showed good electrical conductivities. In addition, the  
180 capacitance of the films was linearly proportional to their CNT concentration. Pham et al. [30]  
181 analyzed the electrical resistances of films of conductive CNTs and polymers under tensile  
182 strains. The results showed that the electrical resistance increased with an increase in the tensile  
183 strain owing to a decrease in the density of the conductive CNT network and an increase in the  
184 intertube distance. Park et al. [31] studied the heat diffusion performance of films of CNTs and  
185 polymers. The thermal diffusivity of the films was analyzed visually based on thermal images.  
186 Jang and Park [32] fabricated 25 mm × 25 mm films of composites with dispersed CNTs. The  
187 CNT concentration and film thickness were varied and the electrical conductivities and heating  
188 performances of the films were analyzed. The electrical resistance of the composite nanofilms  
189 decreased because the CNT networks became denser with an increase in the CNT concentration.  
190 Further, the electrical conductivity increased as the film thickness was increased. For  
191 temperatures of -5 to 5 °C, the films with low thicknesses exhibited greater temperature  
192 sensitivity.

193 In this study, the heating performances and electrical resistances of cement composites  
194 containing MWCNTs were analyzed. A widely employed method of incorporating MWCNTs  
195 into cement is to disperse the MWCNTs in a solution and then mix the solution with the cement  
196 [33-38]. In this study, however, both an MWCNT-containing solution and MWCNT-coated  
197 films were added to cement, and their effects on the heating performance and electrical  
198 resistance were measured. In particular, the effects of the mixing method used, MWCNT  
199 concentration, number of curing days, and applied voltage were investigated. As stated above,  
200 the mixing methods used included adding a solution containing the dispersed MWCNTs with  
201 ordinary mortar, adding MWCNT-coated films to ordinary mortar, and adding both the  
202 MWCNT solution and the MWCNT-coated films. The MWCNT concentrations used were  
203 0.125, 0.25, and 0.5 wt% (by cement weight), and the mortar samples were cured for 7 and 28  
204 days. The surface temperatures and thermal distributions of the composite samples were  
205 analyzed using thermal images. Finally, the internal microstructures of the MWCNT–cement  
206 composites were analyzed using field emission scanning electron microscopy (FE-SEM) and  
207 X-ray diffraction (XRD) analysis.

208  
209

## 2. Experimental

As stated above, the investigated parameters were the MWCNT mixing method used, MWCNT concentration, number of curing days, and applied voltage. Table 1 lists the values of these parameters. The specimens were initially divided into groups based on the mixing method used. The specimens in Group#1 were fabricated by mixing a solution of dispersed MWCNTs with cement mortar. Those in Group#2 were fabricated by inserting MWCNT-coated films into cement mortar. Finally, those in Group#3 were fabricated by mixing a solution of dispersed MWCNTs with mortar and then inserting MWCNT-coated films into the mortar. The specimens were cured for 7 and 28 days, as these correspond to the compressive strength test criteria for cement mortar. Further, as mentioned previously, the MWCNT concentrations were 0.0, 0.125, 0.25, and 0.5 wt% (cement weight). Finally, direct current (DC) voltages of 10, 20, 30, and 60 V were used. **Group # 2 and Group # 3 did not supply more than 30V because if the voltage is more than 30V, the film may be damaged.**

**The naming scheme used to label the various specimens is as follows.** The first letter represents the mixing method. Here, “S” denotes the method in which a solution of dispersed MWCNTs was mixed in ordinary Portland cement while “F” refers to the method of adding MWCNT-coated films to mortar. Finally, “SF” denotes the method where both the solution of dispersed MWCNTs and the MWCNT-coated films were used. The second letter represents the number of curing days: “7D” means 7 days of curing while “28D” means 28 days of curing. Finally, the numerals “0.0”, “0.125,” “0.25,” and “0.5” represent the total concentrations of the MWCNTs (wt% with respect to the cement weight). **For example, ordinary Portland cement with 7 days of curing is expressed as 'S-7D-0.0'.**

There are no standards available for determining the heating performance and electrical resistance of MWCNT–cement composites. Thus, specimens of the MWCNT–cement composites with dimensions of 50 mm × 50 mm × 50 mm, which are the standard specimen dimensions for testing the compressive strength test of cement mortar as per ASTM C 109, were formed (see Fig. 2) [15, 25, 26, 39]. The sand used was standard sand corresponding to the KSL ISO 679 standard. Additionally, ordinary Portland cement was used as the cement [40-42]. **The used sand is KS L ISO 679 sand. This sand has a specific gravity of 2.6 – 2.67, uniformity of 1.0 – 1.93 and minimum dry unit weight of 13.8 – 14.38kN/m<sup>3</sup>. The used cement in this study is Class 1 Portland cement, as defined in KS L 5201. This cement is identical to ASTM Type I [43]. The MWCNT used in this paper is 99% pure and has an average diameter between 1 – 10nm and a length between 100nm – 1cm. The characteristics of MWCNTs are summarized in Table 2. Table 3 shows the mixing ratios for the various MWCNT–cement composites. The water/cement ratio was kept at 0.5 while the sand/cement ratio was set to 1:2.5 based on the weight.** It should be noted that the MWCNT concentration is relative to the cement weight. The amounts of MWCNT in the cement composite for 0.125 wt%, 0.25 wt%, 0.5 wt% were 0.05 g, 0.1 g, 0.2 g, respectively, Furthermore, the amount of MWCNT present in the composite in which the MWCNT aqueous solution and the MWCNT-coated film were combined is the sum of the amount of MWCNT dispersed in the aqueous solution and the amount coated on the film. **Figure 2 shows the process for fabricating the MWCNT–cement composites. To prepare the polymer films for coating with MWCNTs, first, a MWCNT dispersion was mixed in the copolymer polyacrylic acid for 2 hours under ultrasonication at 22 kHz. To prepare the MWCNT-coated films, polystyrene sulfonate and N-methylpyrrolidone solutions were mixed with this dispersion of MWCNTs [44-45]. Finally, a MWCNT dispersion was evenly coated on the transparent polymer films using an air compressor (Fig. 2(a).) The**

256 amounts of the materials to be mixed were determined based on the mixing ratios. The cement  
257 and sand were subjected to dry mixing for 2 min to ensure homogeneous mixing (Fig. 2(b)).  
258 After the completion of the dry mixing process, plain water or the CNT dispersion was added  
259 to the dry mixture and mixed for 3 min, as shown in Fig. 2(c). **After the mixing process, the**  
260 **mortar formed was compacted a total of 30 times to form specimens by dividing layers.** For  
261 the composites formed using MWCNT-coated polymer films, the films were inserted after the  
262 mortar mixture had been poured (Fig. 2(d)). After the insertion of the films, the specimens were  
263 compacted again (Fig. 2(e)). In the case of composites containing the MWCNT films, Cu  
264 electrodes were attached to the films using silver epoxy. For the specimens formed using the  
265 MWCNT solution, Cu meshes were installed at intervals of 2.5 cm to apply a voltage. A  
266 thermocouple was installed at a depth of 2 cm in each specimen to measure the temperature at  
267 its center (Fig. 2(f)). Each specimen was demolded after being cured for 1 day at room  
268 temperature and then dried in an oven at 45°C [42]. **The purpose of curing is to reduce the**  
269 **influence of moisture on the results of the experiment. The reason for using a 45 ° C oven is**  
270 **that according to KCS (Korean Construction Specification) 14 20 10, the temperature gradient**  
271 **should not exceed 15°C and not exceed 65°C per hour. In addition, a high temperature of 65°C**  
272 **or more may cause damage to the nanoparticles, so a 45°C oven was used.**

273 As shown in Fig. 3(a), the voltage was applied using a DC power supply (EX-200). After  
274 connecting the (+) and (-) electrodes to the Cu meshes/electrodes installed in the  
275 MWCNT–cement composites, a voltage was applied using an insulating rubber plate. For each  
276 sample, the voltage was applied for 60 min. The internal temperature of each composite, which  
277 varied with the applied voltage, was measured by connecting the installed thermocouple to a  
278 data logger (TDS-303). The surface temperatures of the MWCNT–cement composites were  
279 analyzed based on thermal images, which allowed the maximum temperatures to be determined.  
280 To measure the electrical resistances of the MWCNT–cement composites, a digital multimeter  
281 (Keithley 2701) was used, as shown in Fig. 3(b). The (+) and (-) electrodes of the digital  
282 multimeter were connected to the Cu meshes/electrodes installed in the MWCNT–cement  
283 composites. The resistance in response to the applied voltage was measured with the digital  
284 multimeter for 60 min by connecting the multimeter to a computer. **The internal**  
285 **microstructures of the MWCNT–cement composites were analyzed through FE-SEM and**  
286 **XRD analysis. FE-SEM uses an electron beam and an electron lens for microstructural analysis.**  
287 **This involves the magnification of objects based on the secondary and scattered electrons**  
288 **generated when the electron beam collides with the surface of the test specimen. XRD analysis**  
289 **allows for investigation of the structure and characteristics of an object based on the angle of**  
290 **diffraction of X-rays when they collide with the object.**

### 291 292 **3. Results and discussion**

#### 293 **3.1. Heating tests and thermal imaging analysis**

294 The effects of the mixing method, number of curing days, MWCNT concentration, and  
295 applied voltage on the heating performance were investigated. **Figure 4(a) shows the maximum**  
296 **variations in temperature of the composites fabricated using the MWCNT solution after 7 days**  
297 **of curing. When a voltage of 60 V was applied to the specimens fabricated using the MWCNT**  
298 **solution, the temperature rose by 1.0 °C for the 0.0wt% composites, by 1.3 °C for the 0.125wt%**  
299 **composites, by 1.8 °C for the 0.25 wt% composites, and by 19.8 °C for the 0.5 wt% composites.**  
300 **Thus, the temperature increase for the 0.5 wt% composites was 19.8 times higher than that for**



301 the composites with 0.0 wt% MWCNTs, which was the lowest concentration. Hence, it can be  
302 concluded that the temperature increased with MWCNT concentration. Figure 4(b) shows the  
303 maximum temperature variations at the centers of the composites formed using the MWCNT  
304 films after 7 days of curing. On the other hand, when the voltage applied was 30 V, for the  
305 composites fabricated using the MWCNT films, the temperature increase was 0.7 °C for the  
306 0.125 wt% composites, 9.5 °C for the 0.25 wt% composites, and 37.2 °C for the 0.5 wt%  
307 composites. Thus, the 0.5 wt% composites with the MWCNT films exhibited a temperature  
308 increase that was 7.2 times higher than that for the MWCNT-solution composites with the same  
309 MWCNT concentration at 60 V. Therefore, the composites with the inserted MWCNT films  
310 exhibited relatively higher temperatures even at a lower voltage (30 V). Figure 4(c) shows the  
311 maximum variations in the temperature of the composites fabricated using both the MWCNT  
312 solution and the MWCNT films after 7 days of curing. When a voltage of 30 V was applied to  
313 the composites fabricated using both the MWCNT solution and the MWCNT films, the  
314 temperature increase was 0.8 °C for the 0.125 wt% composites, 10.9 °C for the 0.25 wt%  
315 composites, and 77.5 °C for the 0.5 wt% composites. Thus, for a voltage of 30 V, the 0.5 wt%  
316 composites formed using a combination of the MWCNT solution and the MWCNT films  
317 exhibited a temperature increase 13.9 times higher than that of the composites that had the  
318 same MWCNT concentration but were fabricated using the MWCNT solution and 2.1 times  
319 higher than that of the composites that had the same MWCNT concentration but were  
320 fabricated using MWCNT films. Hence, use of a combination of the MWCNT solution and the  
321 MWCNT films resulted in improved heating performance. This is because the aforementioned  
322 combination generates heat when current flows through the film. Furthermore, the cement  
323 composites consisting of the MWCNT solution have excellent thermal conductivity; therefore,  
324 the heat generated from the MWCNT film is uniformly distributed throughout the  
325 aforementioned combination. For this reason, the heat performance of the composite fabricated  
326 using the MWCNT solution and film combination was the most optimal.

327 Figure 4(d) shows the temperature variations at the centers of the MWCNT–cement  
328 composites formed using the MWCNT solution after 28 days of curing. For an applied voltage  
329 of 60 V, the temperature of the composites fabricated using the MWCNT solution increased  
330 by 0.5 °C for the 0.0 wt% specimen, by 0.6 °C for the 0.125 wt% specimen, by 0.9 °C for the  
331 0.25 wt% specimen, and by 16.8 °C for the 0.5 wt% specimen. The temperature of the 0.125  
332 wt% specimen cured for 28 days was 0.7 °C lower than that of the corresponding 0.125 wt%  
333 specimen cured for 7 days. Similarly, the temperature of the 0.5 wt% specimen cured for 28  
334 days was 3 °C lower than the corresponding specimen cured for 7 days. Figure 4(e) shows the  
335 heating performance graph of the MWCNT–cement composites fabricated by inserting  
336 MWCNT films for different voltages and MWCNT concentrations. When a voltage of 30 V  
337 was applied, the temperature rose by 0.9 °C for the 0.125 wt% specimen, by 3.7 °C for the 0.25  
338 wt% specimen, and by 36.1 °C for the 0.5 wt% specimen. The heating performances of the  
339 specimens cured for 28 days were slightly lower compared to those of the corresponding ones  
340 cured for 7 days. Figure 4(f) shows the heating performance graph of the MWCNT–cement  
341 composites fabricated using both the MWCNT solution and the MWCNT films. For a voltage  
342 of 30 V, the temperature increased by 0.8 °C for the 0.125 wt% specimen, by 5.1 °C for the  
343 0.25 wt% specimen, and by 76.4 °C for the 0.5 wt% specimen. Hence, both the method of using  
344 the MWCNT solution and the MWCNT films and curing for 7 days resulted in excellent  
345 heating performances. Moreover, the specimens in Group#3 also showed slightly lower heating  
346 performances after 28 days of curing. The difference in temperature increase due to the

347 difference in the number of curing days was 0 °C for the 0.125 wt% specimens, 5.8 °C for the  
348 0.25 wt% specimens (this was the largest difference), and 1.1 °C for the 0.5 wt% specimens.  
349 This trend can be attributed to the fact that the hydration reaction did not increase the  
350 temperature significantly, owing to the effects of moisture, when the MWCNT–cement  
351 composites were cured for 7 days [47].

352 Table 4 lists the temperatures of the MWCNT–cement composites according to the various  
353 parameters. The highest temperatures were 77.5 °C for SF-7D-0.5 after 7 days of curing and  
354 76.4 °C for SF-28D-0.5 after 28 days of curing. This is because more CNT networks were  
355 formed in these specimens, as the CNTs were evenly dispersed not only in the MWCNT-coated  
356 films but also within the composites. The temperatures of the composites increased with the  
357 MWCNT concentration. Moreover, as the number of curing days was increased, the increase  
358 in the temperature decreased; this was because of the reduced effect of the moisture present in  
359 the samples. However, the temperature increases after 7 and 28 days of curing were similar.

360 Figure 5 shows the thermal images of the MWCNT–cement composites after 28 days of  
361 curing. As shown in Figure 5, the surface temperature of the cement composite is calculated  
362 by adding the initial internal temperature of the composite and the temperature change. For  
363 example, SF-28D-0.5 has an initial internal temperature of 25.1 °C and a temperature change  
364 of 76.4 °C. Thus, the SF-28D-0.5 surface temperature was calculated to be 101.5 °C. Figures  
365 5 (a) and 5(b) show the thermal images of the composites fabricated using the MWCNT  
366 solution when voltages of 30 and 60 V, respectively, were applied. At 60 V, the surface  
367 temperatures of S-28D-0.125, S-28D-0.25, and S-28D-0.5 were 20.7, 21.1, and 42.4°C,  
368 respectively. Thus, for a voltage of 60 V, the surface temperatures of the MWCNT–cement  
369 composites increased by up to 14.2 °C compared to those for 30 V.

370 For the same applied voltage, the surface temperature increased with the MWCNT  
371 concentration, resulting in clear thermal images. Moreover, in the case of the composites  
372 fabricated using the MWCNT solution, heating occurred primarily between the electrodes  
373 connected to the voltage supply. This was because current has a tendency to flow along the  
374 shortest distance; thus, the heat generated was distributed mainly between the electrodes.  
375 Figure 5(c) shows the thermal images of the composites fabricated by inserting MWCNT films.  
376 When the voltage was 30 V, the highest surface temperature was 60.5 °C and was observed in  
377 the case of F-28D-0.5. However, for the same voltage, the temperatures of the cement  
378 composites fabricated by inserting MWCNT films were 35.1 °C higher than those of the  
379 composites fabricated using the MWCNT solution. Sample OF-28D-0.125 exhibited a  
380 temperature of 21.1 °C, with its thermal image showing the insignificant temperature. Further,  
381 its temperature was 0.4 °C higher than that of S-28D-0.125 and thus not significantly different.  
382 The temperature of F-28D-0.25 was 24.2°C, and its thermal image clearly indicated an increase  
383 in the surface temperature as compared to that of S-28D-0.25. For the composites formed using  
384 the MWCNT films, the temperature change occurred from the center, where the films were  
385 inserted. This was because the MWCNTs were concentrated on the film surfaces. Figure 5(d)  
386 shows the thermal images of the composites fabricated using both the MWCNT solution and  
387 the MWCNT films. The surface temperature of SF-28D-0.5 was 101.5 °C, which was the  
388 highest after 28 days of curing.

389 The samples fabricated using both the solution and the films having MWCNT concentrations  
390 of 0.25 wt% and 0.5 wt% exhibited higher temperatures than those of the composites formed  
391 using only the MWCNT films. The heating performance of the former was better because in  
392 their case, the MWCNTs were evenly dispersed in the films; therefore, the composites, with

393 CNT networks forming throughout the specimens.

394 Figure 6 shows the temperature graphs of the MWCNT–cement composites over time.  
395 Figures 6 (a) and 6(b) show the temperature graphs of the composites fabricated using the  
396 MWCNT solution over time. The samples with 0.125 wt% MWCNTs exhibited relatively  
397 small temperature increases than those with 0.25 and 0.5 wt% MWCNTs. For the composites  
398 fabricated using the MWCNT solution, the temperature gradient increased with the MWCNT  
399 concentration, with the temperatures plateauing once the highest temperatures had been  
400 reached. Moreover, the temperature gradient after 7 days of curing was higher than that after  
401 28 days. Thus, it can be concluded that the temperature gradient was relatively higher because  
402 the current flowed well in the composites owing to the effects of the moisture present within  
403 them. Figures 6 (c) and 6(d) show the temperature increase graphs over time of the composites  
404 fabricated using the MWCNT films and cured for 7 and 28 days, respectively. The temperature  
405 gradients and temperatures varied depending on the mixing method used. Except for F-0.5-  
406 30V at 7 and 28 days, the temperatures remained constant over time as with the method of  
407 mixing solution. For F-0.5-30V, the temperature increased initially, reaching the maximum  
408 value, and then decreased over time. Figures 6(e) and 6(f) show the temperature graphs over  
409 time for the composites fabricated using both the MWCNT solution and the MWCNT films.  
410 The temperature gradients were relatively higher than those of the specimens formed using the  
411 MWCNT solution as well as those formed using the MWCNT films. This suggests that the  
412 temperatures and thermal diffusivities increased in the former case because the MWCNTs  
413 dispersed in the solution and films formed CNT networks together. For SF-0.5-30V as well,  
414 the temperature first increased, reaching the maximum value, and then decreased. As with the  
415 ordinary mortar, when a voltage of 30 V or higher was applied to the composites formed using  
416 the MWCNT films, the inserted films were damaged, as shown in Figure 7. As a result, the  
417 networks of the dispersed MWCNTs were destroyed, resulting in a decrease in the heating  
418 performance. Therefore, care should be taken when applying high voltages (30V or higher), as  
419 the MWCNT films can be damaged in such cases.

420

### 421 3.2. Electrical resistance measurements

422 Figure 8(a) shows the resistance graphs of the MWCNT–cement composites formed using  
423 the MWCNT solution after 7 days of curing. In the case of the composites fabricated using the  
424 MWCNT solution, the resistance of the 0.0 wt% specimen was 972 k $\Omega$ , 0.125 wt% specimen  
425 was 52.8 k $\Omega$  and that of the 0.25 wt% specimen was 20.72 k $\Omega$ . Thus, the resistance of the latter  
426 was 46.9 times lower than 0.0 wt% specimen. Further, the resistance of the 0.5 wt% specimen  
427 was 0.9 k $\Omega$ , which was 59 times lower than that of the 0.125 wt% specimen (Fig. 8(a)). These  
428 results indicate that the electrical resistance reduced as the MWCNT concentration was  
429 increased, because the CNTs formed denser networks as their content was increased[46].  
430 Figure 8(b) shows the resistance graphs of the MWCNT–cement composites fabricated using  
431 the MWCNT films after 7 days of curing. In the case of the composites formed using the  
432 MWCNT films, the resistance was 88.4 k $\Omega$  for the 0.125 wt% specimen, 0.74 k $\Omega$  for the 0.25  
433 wt% specimen, and 0.075 k $\Omega$  for the 0.55 wt% specimen (Fig. 10(b)). Further, the resistances  
434 of the 0.25 wt% and 0.5 wt% specimens were 28% and 12% lower, respectively, compared to  
435 those of the composites fabricated using the MWCNT solution—the 0.125 wt% specimen was  
436 an exception here. Figure 8(c) shows the electrical resistances of the MWCNT–cement  
437 composites fabricated using both the MWCNT solution and the MWCNT films. The electrical  
438 resistance was 82.9 k $\Omega$  for the 0.125 wt% specimen, 0.71 k $\Omega$  for the 0.25 wt% specimen, and

439 0.05 k $\Omega$  for the 0.5 wt% specimen. These results were similar to those for the corresponding  
440 specimens fabricated using the MWCNT films. This suggests that the differences in electrical  
441 resistance of the corresponding specimens from the two groups were insignificant because the  
442 electrodes of the MWCNT films were integrated with the films. The electrical resistance  
443 experiments measured the resistance caused by a fine current supply. Since the CNT-films used  
444 in Group # 2 and Group # 3 were integrated with the copper electrode, the difference in  
445 resistance due to small currents accurately measures the resistance of CNTs dispersed in the  
446 film. Therefore, the differences in resistance of CNT-films coated with the same concentration  
447 are considered to be insignificant.

448 Figure 8(d) shows the resistance graphs for the composites produced using the MWCNT  
449 solution after 28 days of curing. The resistance of the 0.0 wt% specimen was 980 k $\Omega$ , which  
450 was the largest resistance increase for the composites formed using the MWCNT solution after  
451 28 days of curing. The resistance of the 0.125 wt% specimen was 62.1 k $\Omega$ , which was 9.3 k $\Omega$   
452 higher compared to that after 7 days of curing. The resistance of the 0.25 wt% specimen  
453 increased by 5.08 k $\Omega$  to 25.8 k $\Omega$  while the resistance of the 0.5 wt% specimen increased by  
454 0.08 k $\Omega$  to 0.98 k $\Omega$ . With an increase in the curing duration, the electrical resistance of the  
455 specimens increased but their temperature upon the application of a voltage decreased slightly.  
456 Figure 8(e) shows the resistance graphs for the composites produced using the MWCNT films  
457 after 28 days of curing. The resistance was 98.9 k $\Omega$  for the 0.125 wt% specimen, 3.2 k $\Omega$  for  
458 the 0.25 wt% specimen, and 0.1 k $\Omega$  for the 0.5 wt% specimen. For the 0.5 wt% specimen, the  
459 resistance was higher by 0.025 k $\Omega$  compared to that after 7 days of curing. Figure 8(f) shows  
460 the resistance graphs of the composites fabricated using both the MWCNT solution and the  
461 MWCNT films after 28 days of curing. The resistance of the 0.125 wt% specimen was 98.5  
462 k $\Omega$  and higher by 0.4 k $\Omega$  than that of OF-28D-0.125. The resistance of the 0.25 wt% specimen  
463 was 2.91 k $\Omega$  and higher by 0.29 k $\Omega$  than that of F-28D-0.25. Finally, the resistance of the 0.5  
464 wt% specimen was 0.067 k $\Omega$ . These resistance values were lower than those of the  
465 corresponding composites fabricated using the MWCNT films after 7 days of curing.

466 Table 5 lists the electrical resistances of the various MWCNT–cement composites. Of the  
467 various composites formed using the MWCNT solution, S-7D-0.5 exhibited the lowest  
468 resistance at 0.9 k $\Omega$ . As for the composites formed using the MWCNT films, OF-7D-0.5  
469 showed a resistance of 0.075 k $\Omega$ , which was 12 times lower than that of S-7D-0.5. Finally, in  
470 the case of the composites fabricated using both the MWCNT solution and the MWCNT films,  
471 SF-7D-0.5 exhibited a resistance of 0.05 k $\Omega$ , which was the lowest of all the composites. The  
472 resistance decreased as the MWCNT concentration was increased because the presence of a  
473 greater number of conductive MWCNTs resulted in denser CNT networks. Further, as the  
474 number of curing days was increased, the resistance increased slightly because the hydration  
475 process of the cement composites occurred to completion. However, the differences in  
476 resistance corresponding to the different curing durations were insignificant[38].

### 478 3.3. Microstructural analysis

479 Figure 9 shows the XRD patterns of the composites formed using the MWCNT solution.  
480 Peaks were observed between  $2\theta$  values of  $26^\circ$  and  $29^\circ$  for the pure mortar sample as well as  
481 the 0.125, 0.25, and 0.5 wt% samples. This suggested that portlandite (C-H), a hydrate  
482 generated during the hydration reaction, was formed in the case of every sample. The peak at  
483  $38^\circ$ , which was observed for every specimen, is attributable to calcium silicate hydrates (C-S-  
484 H), which significantly affect the strength. These results indicated that in each  
485 MWCNT–cement composite, the same hydration process occurred and that the same hydration

486 products were generated as those in ordinary mortar.

487 Figure 10 shows FE-SEM images of the MWCNT–cement composites. The red crosses in  
488 the images represent MWCNTs. Figure 10(a) shows a microstructural image of the composite  
489 formed using the 0.5 wt% MWCNT solution. The image confirmed that MWCNTs were evenly  
490 dispersed in the composite and that CNT networks were formed in it, as the cement hydrates  
491 and MWCNTs were connected. The formation of these CNT networks appears to have  
492 improved the heating performance and electrical conductivity of the composites. Figure 10(b)  
493 shows a microstructural image of a film damaged by a voltage of 30 V. No CNT networks can  
494 be seen in this film, as the MWCNTs connected on the film surface became disconnected after  
495 the application of the voltage. This was the reason for the poorer heating performance and  
496 higher electrical resistance of the composites with the MWCNT-coated films. Figure 10(c)  
497 shows a microstructural image of an undamaged MWCNT-coated film. A number of red  
498 crosses can be seen distributed on the surface of the film. This indicated that the MWCNTs did  
499 not aggregate and remained concentrated on the surface of the film, forming CNT networks.

500

501



502 **4. Conclusions**

503 In this study, MWCNT–cement composites were fabricated using MWCNTs, which were  
504 mixed in cement using three different methods. The heating performances and electrical  
505 resistances of the fabricated composites were also investigated. Based on the experimental  
506 results, the performances of the MWCNT–cement composites were analyzed, and the  
507 following conclusions were drawn:

508  
509 1. With respect to the methods used for incorporating MWCNTs, the composites formed by  
510 inserting MWCNT-coated films in a mortar exhibited better heating performance than those  
511 formed by the mixing of a MWCNT-dispersed solution in the cement. The highest temperature  
512 increase at 30 V was 37.2 °C and was seen in the case of F-7D-0.5, while that at 60 V was 19.8  
513 °C and was seen in the case of S-7D-0.5. The heating performance of the composites formed  
514 using the MWCNT films was 1.9 times higher than that of the composites formed using the  
515 MWCNT solution. This result suggested that the increase in temperature per unit area was  
516 higher in the former case because the MWCNTs were concentrated on the film surfaces.

517 2. In the case of the composites fabricated using both the MWCNT solution and the  
518 MWCNT films, the highest temperature increase was 77.5 °C and was observed in the case of  
519 SF-7D-0.5. This sample also exhibited the lowest electrical resistance at 0.05 kΩ. Thus, use of  
520 both the MWCNT solution and the MWCNT films is the most suitable method for improving  
521 the heating performance and electrical conductivity. This is because more CNT networks were  
522 formed in these specimens, as the CNTs were evenly dispersed not only in the MWCNT-coated  
523 films but also within the composites themselves.

524 3. The higher the MWCNT concentration was, the higher the heating performance and the  
525 lower the electrical resistance became. This is because the presence of a greater number of  
526 conductive MWCNTs meant that more CNT networks were formed in the cement composites.

527 4. As the number of curing days was increased, the heating performance decreased slightly  
528 and so did the electrical resistance. After 7 days of curing, the heating performance increased  
529 relatively and so did the electrical resistance owing to the effects of the moisture present  
530 within the composites. **Future research will be conducted to study the heating and resistive  
531 performance of long-term curing over 28 days of curing of MWCNT-cement composites.**

532 5. The XRD analysis results confirmed that portlandite (C-H) and calcium silicate hydrates  
533 (C-S-H) were also formed in the MWCNT–cement composites in the same manner as in  
534 ordinary mortar during the hydration process. Thus, it can be concluded that the same hydration  
535 processes also occurred in cement composites containing MWCNTs. The FE-SEM results  
536 confirmed the formation of CNT networks on the surfaces of the MWCNT films, which  
537 improved the heating performance and reduced the electrical resistance. In the case of the  
538 MWCNT film damaged by the application of a high voltage, the heating performance  
539 decreased and the electrical resistance increased owing to breakage of the CNT networks.

540  
541

542 **Acknowledgements**

543 This study is a basic research project conducted with support from the National Research Foundation of Korea  
544 through government funds (Ministry of Science, ICT and Future Planning, South Korea). The project numbers are  
545 2017R1A2B4010467, 2017R1C1B1006732. Professor K. Loh was supported by the U.S. Federal Aviation  
546 Administration (FAA) under cooperative agreement no. 13-G-017, which is a collaborative project with Dr. R.  
547 Wu of UC Davis and Professor J. Lynch of the University of Michigan. Mr. Sumit Gupta (UC San Diego) is  
548 acknowledged for his help in the preparation of the MWCNT-based thin films used in this study.

549 **References**

- 550 [1] L. Raki, J. Beaudoin, R. Alizadeh, J. Makar, T. Sato, Cement and concrete nanoscience  
551 and nanotechnology, *Mater.* 3 (2010) 918–942.
- 552 [2] F. Sanchez, K. Sobolev, Nanotechnology in concrete—a review, *Constr. Build. Mater.*  
553 24 (2010) 2060–2071.
- 554 [3] M.J. Hanus, A.T. Harris, Nanotechnology innovations for the construction industry,  
555 *Prog. Materi. Sci.* 58 (2013) 1056–1102.
- 556 [4] Z. Fan, J. Chen, M. Wang, K. Cui, H. Zhou, Y. Kuang, Preparation and characterization  
557 of manganese oxide/CNT composites as supercapacitive materials, *Diam. Relat. Materi.*  
558 15 (2006) 1478–1483.
- 559 [5] M.T. Kim, K.Y. Rhee, J.H. Lee, D. Hui, A.K. Lau, Property enhancement of a carbon  
560 fiber/epoxy composite by using carbon nanotubes, *Compos. Part B: Eng.* 42 (2011)  
561 1257–1261
- 562 [6] S. Chuah, Z. Pan, J.G. Sanjayan, C.M. Wang, W.H. Duan, Nano reinforced cement and  
563 concrete composites and new perspective from graphene oxide, *Constr. Building Mater.*  
564 73 (2014) 113–124
- 565 [7] Lee, S. H., Kang, D., Oh, I. K. (2017). Multilayered graphene-carbon nanotube-iron  
566 oxide three-dimensional heterostructure for flexible electromagnetic interference  
567 shielding film. *Carbon*, 111, 248-257.
- 568 [8] Y. Saez de Ibarra, J.J. Gaitero, E. Erkizia, I. Campillo, Atomic force microscopy and  
569 nanoindentation of cement pastes with nanotube dispersions. *Phys. Stat. Sol. (a)* 203  
570 (2006) 1076–1081.
- 571 [9] B. Han, X. Yu, E. Kwon, A self-sensing carbon nanotube/cement composite for traffic  
572 monitoring, *Nanotechnol.* 20 (2009) 445501.
- 573 [10] R. Siddique, A. Mehta, Effect of carbon nanotubes on properties of cement mortars,  
574 *Constr. Build. Mater.* 50 (2014) 116–129.
- 575 [11] M.S. Konsta-Gdoutos, Z.S. Metaxa, S.P. Shah, Highly dispersed carbon nanotube  
576 reinforced cement based materials, *Cement Concr. Res.* 40 (2010) 1052–1059.
- 577 [12] E. García-Macías, A. D'Alessandro, R. Castro-Triguero, D. Pérez-Mira, F. Ubertini,  
578 Micromechanics modeling of the electrical conductivity of carbon nanotube cement-  
579 matrix composites, *Compos. Part B: Eng.* 108 (2017) 451–469.
- 580 [13] G.Y. Li, P.M. Wang, X. Zhao, Mechanical behavior and microstructure of cement  
581 composites incorporating surface-treated multi-walled carbon nanotubes, *Carbon* 43  
582 (2005) 1239–1245.
- 583 [14] A. Chaipanich, T. Nochaiya, W. Wongkeo, P. Torkittikul, Compressive strength and  
584 microstructure of carbon nanotubes–fly ash cement composites, *Mater. Sci. Eng. A*,  
585 527 (2010) 1063–1067.
- 586 [15] M.S. Morsy, S.H. Alsayed, M. Aqel, Hybrid effect of carbon nanotube and nano-clay  
587 on physico-mechanical properties of cement mortar, *Constr. Build. Mater.* 25 (2011)  
588 145–149.
- 589 [16] R. Hamzaoui, A. Bennabi, S. Guessasma, R. Khelifa, N. Leklou, Optimal carbon  
590 nanotubes concentration incorporated in mortar and concrete, In *Adv. Materi. Res.* (Vol.  
591 587, pp. 107-110). Trans Tech Publications.
- 592 [17] H. Choi, D. Kang, G.S. Seo, W. Chung, Effect of some parameters on the compressive  
593 strength of MWCNT-cement composites, *Adv. Mater. Sci. Eng.* (2015)



- 594 [18] S.J. Ha, S.T. Kang, J.H. Lee, Strength of CNT cement composites with different types  
595 of surfactants and doses, *J. Kor. Inst. Struct. Maint. Insp.* 19 (2015) 99–107.
- 596 [19] C.W. Nan, G. Liu, Y. Lin, M. Li, Interface effect on thermal conductivity of carbon  
597 nanotube composites, *Appl. Phys. Lett.* 85 (2004) 3549–3551.
- 598 [20] M.S. Liu, M.C.C. Lin, I.T. Huang, C.C. Wang, Enhancement of thermal conductivity  
599 with carbon nanotube for nanofluids, *Int. Commun. Heat Mass Transf.* 32 (2005)1202–  
600 1210.
- 601 [21] G.Y. Li, P.M. Wang, X. Zhao, Pressure-sensitive properties and microstructure of  
602 carbon nanotube reinforced cement composites, *Cement Concr. Compos.* 29 (2007)  
603 377–382.
- 604 [22] Q. Zhang, H. Li, Experimental investigation on the ice/snow melting performance of  
605 CNFP & MWCNT/cement-based deicing system, In *Proceedings of the 6th*  
606 *International Workshop on Advanced Smart Materials and Smart Structures*  
607 *Technology, Dalian, China, 2011*, pp. 25–26.
- 608 [23] H.K. Kim, I.W. Nam, H.K. Lee, Enhanced effect of carbon nanotube on mechanical  
609 and electrical properties of cement composites by incorporation of silica fume, *Compos.*  
610 *Struct.* 107 (2014) 60–69.
- 611 [24] M.S. Konsta-Gdoutos, C.A. Aza, Self sensing carbon nanotube (CNT) and nanofiber  
612 (CNF) cementitious composites for real time damage assessment in smart structures,  
613 *Cement Concr. Compos.* 53 (2014) 162–169.
- 614 [25] H. Lee, Y.M. Song, K.J. Loh, W. Chung, Thermal response characterization and  
615 comparison of carbon nanotube-enhanced cementitious composites, *Compos. Struct.*  
616 202 (2018) 1042–1050.
- 617 [26] H. Lee, D. Kang, J. Kim, K. Choi, W. Chung, Void detection of cementitious grout  
618 composite using single-walled and multi-walled carbon nanotubes, *Cement Concr.*  
619 *Compos.* 95 (2019) 237–246.
- 620 [27] J. Hone, M.C. Llaguno, N.M. Nemes, A.T. Johnson, J.E. Fischer, D.A. Walters, R.E.  
621 Smalley, Electrical and thermal transport properties of magnetically aligned single wall  
622 carbon nanotube films, *Appl. Phys. Lett.* 77 (2000) 666–668.
- 623 [28] H. Huang, C.H. Liu, Y. Wu, S. Fan, Aligned carbon nanotube composite films for  
624 thermal management, *Adv. Mater.* 17 (2005) 1652–1656.
- 625 [29] J.H. Kim, K.W. Nam, S.B. Ma, K.B. Kim, Fabrication and electrochemical properties  
626 of carbon nanotube film electrodes, *Carbon*, 44 (2006) 1963–1968.
- 627 [30] G.T. Pham, Y.B. Park, Z. Liang, C. Zhang, B. Wang, Processing and modeling of  
628 conductive thermoplastic/carbon nanotube films for strain sensing. *Compos. Part B:*  
629 *Eng.* 39 (2008) 209–216.
- 630 [31] J. Park, A. Lee, Y. Yim, E. Han, Electrical and thermal properties of PEDOT: PSS  
631 films doped with carbon nanotubes, *Synth. Metals* 161 (2011) 523–527.
- 632 [32] S.H. Jang, Y.L. Park, Carbon nanotube-reinforced smart composites for sensing  
633 freezing temperature and deicing by self-heating, *Nanomater. Nanotechnol.* 8 (2018)  
634 1847980418776473.
- 635 [33] A. Sobolkina, V. Mechtcherine, V. Khavrus, D. Maier, M. Mende, M. Ritschel, A.  
636 Leonhardt, Dispersion of carbon nanotubes and its influence on the mechanical  
637 properties of the cement matrix, *Cement Concr. Compos.* 34 (2012) 1104–1113.
- 638 [34] A. Al-Dahawi, O. Öztürk, F. Emami, G. Yıldırım, M. Şahmaran, Effect of mixing  
639 methods on the electrical properties of cementitious composites incorporating different  
640 carbon-based materials, *Construc. Build. Mater.* 104 (2016)160–168.

- 641 [35] Al-Dahawi, A., Yıldırım, G., Öztürk, O., Şahmaran, M. (2017). Assessment of self-  
642 sensing capability of Engineered Cementitious Composites within the elastic and  
643 plastic ranges of cyclic flexural loading. *Construction and Building Materials*, 145, 1-  
644 10.
- 645 [36] Yıldırım, G., Sarwary, M. H., Al-Dahawi, A., Öztürk, O., Anıl, Ö., Şahmaran, M.  
646 (2018). Piezoresistive behavior of CF-and CNT-based reinforced concrete beams  
647 subjected to static flexural loading: shear failure investigation. *Construction and*  
648 *Building Materials*, 168, 266-279.
- 649 [37] Sarwary, M. H., Yıldırım, G., Al-Dahawi, A., Anıl, Ö., Khiavi, K. A., Toklu, K.,  
650 Şahmaran M. (2019). Self-sensing of flexural damage in large-scale steel-reinforced  
651 mortar beams. *ACI Materials Journal*, 116(4), 209-221.
- 652 [38] Yıldırım, G., Öztürk, O., Al-Dahawi, A., Ulu, A. A., Şahmaran, M. (2020). Self-  
653 sensing capability of Engineered Cementitious Composites: Effects of aging and  
654 loading conditions. *Construction and Building Materials*, 231, 117132.
- 655 [39] ASTM C109, Standard test method for compressive strength of hydraulic cement  
656 mortars (Using 2-in, or 50-mm cube specimens), ASTM International (2016).
- 657 [40] KSL ISO 679, Methods of testing cements-determination of strength, Korean  
658 Standards Association (2011).
- 659 [41] KS L 5201, Portland cement, Korean Standards Association (2016).
- 660 [42] KCS 14 20 10, Concrete, Korean Construction Specification (2016).
- 661 [43] ASTM C150, Standard specification for Portland cement, ASTM International (2012).
- 662 [44] J.G. Gonzalez, S. Gupta, K.J. Loh, Multifunctional cement composites enhanced with  
663 carbon nanotube thin film interfaces, In *Proceedings of the IEEE* 104.8, 2016, pp. 1547-  
664 1560
- 665 [45] S. Gupta, J.G. Gonzalez, K.J. Loh, Self-sensing concrete enabled by nano-engineered  
666 cement-aggregate interfaces, *Struct. Health Monit.* 16.3 (2017) 309–323.
- 667 [46] Al-Dahawi, A., Sarwary, M. H., Öztürk, O., Yıldırım, G., Akın, A., Şahmaran, M.,  
668 Lachemi, M. (2016). Electrical percolation threshold of cementitious composites  
669 possessing self-sensing functionality incorporating different carbon-based materials.  
670 *Smart Materials and Structures*, 25(10), 105005.
- 671 [47] Han, B., Yu, X., & Ou, J. (2010). Effect of water content on the piezoresistivity of  
672 MWNT/cement composites. *Journal of Materials Science*, 45(14), 3714-3719.
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677 **Figure Captions**

678 Figure 1 Schematics of various specimens.

679 Figure 2. Sample fabrication process.

680 Figure 3. Test setup for measuring electrical resistance and heating performance.

681 **Figure 4. Variations in temperature upon application of voltage.**

682 Figure 5. Thermal images of specimens after curing for 28 days.

683 Figure 6. Temperature–time curves.

684 Figure 7. Damaged MWCNT-covered film.

685 **Figure 8. Electrical resistance with various parameters.**

686 Figure 9. XRD patterns of various samples.

687 Figure 10. FE-SEM images of various samples.

688

689 **Table Captions**

690 Table 1. Parameters used for temperature measurements

691 **Table 2. Properties of MWCNTs**

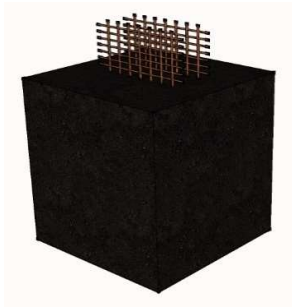
692 Table 3. Mixing ratios of various specimens

693 Table 4. Results of heating tests

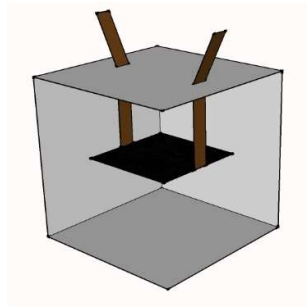
694 Table 5. Results of electrical resistance measurements

695

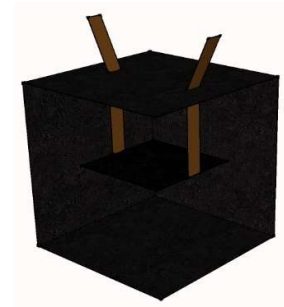
696



(a) MWCNT solution



(b) MWCNT films



(c) MWCNT solution and  
MWCNT films

697

**Fig. 1. Schematics of various specimens.**



(a) Dispersion of MWCNTs



(b) Mixing of cement and sand



(c) Mixing of MWCNT dispersion (3 min)



(d) Insertion of MWCNT films

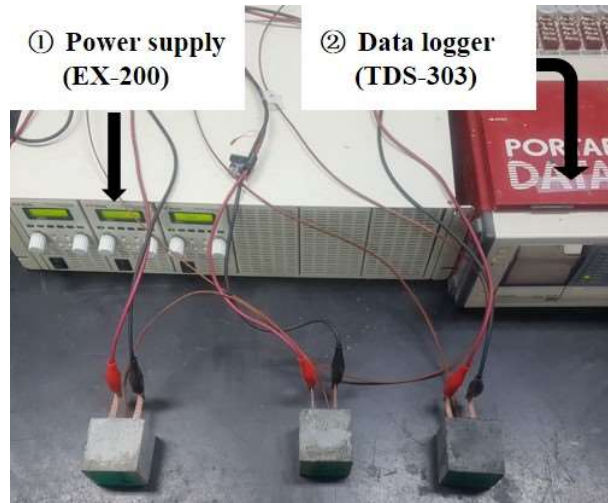


(e) Compaction

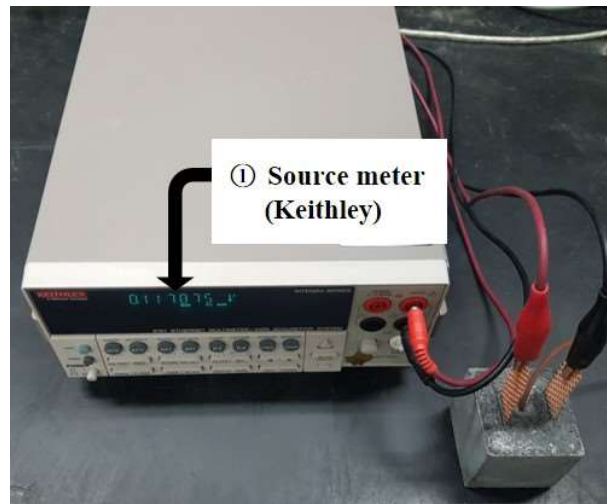


(f) Insertion of thermocouple

**Fig. 2. Sample fabrication process.**



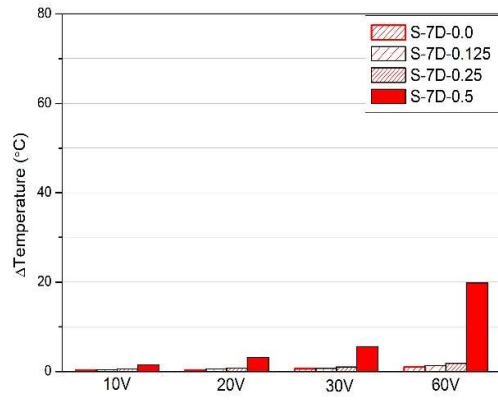
(a) Measurement of temperature upon application of voltage



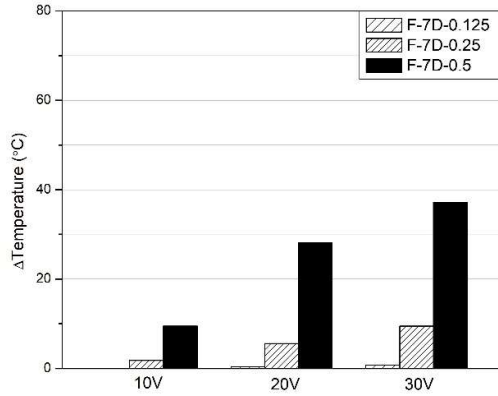
(b) Measurement of electrical resistance upon application of voltage

699

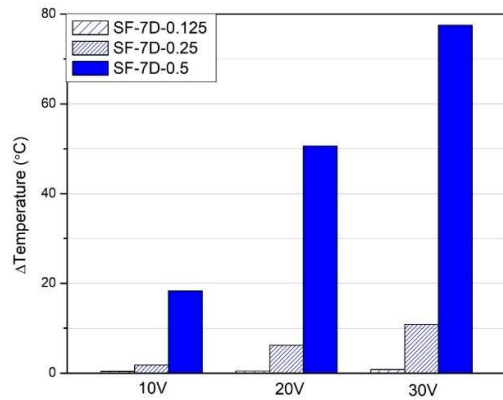
**Fig. 3. Test setup for measuring electrical resistance and heating performance.**



(a) MWCNT solution of 7 curing days

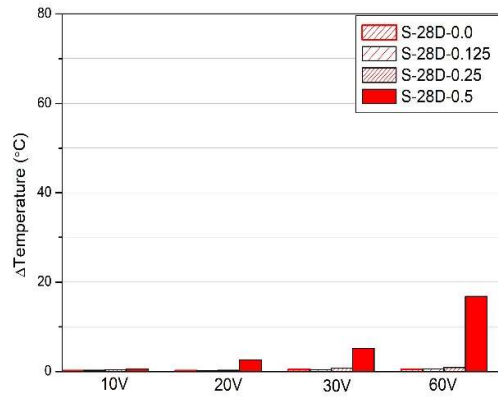


(b) MWCNT films of 7 curing days

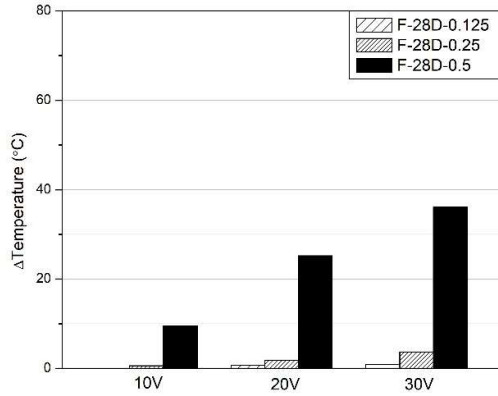


(c) MWCNT solution and MWCNT films of 7 curing days

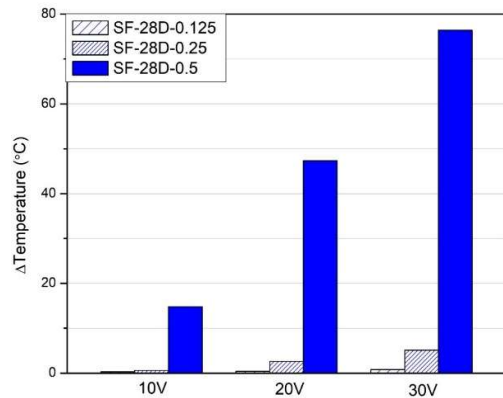
**Fig. 4. Variations in temperature upon application of voltage (continue).**



(d) MWCNT solution of 28 curing days

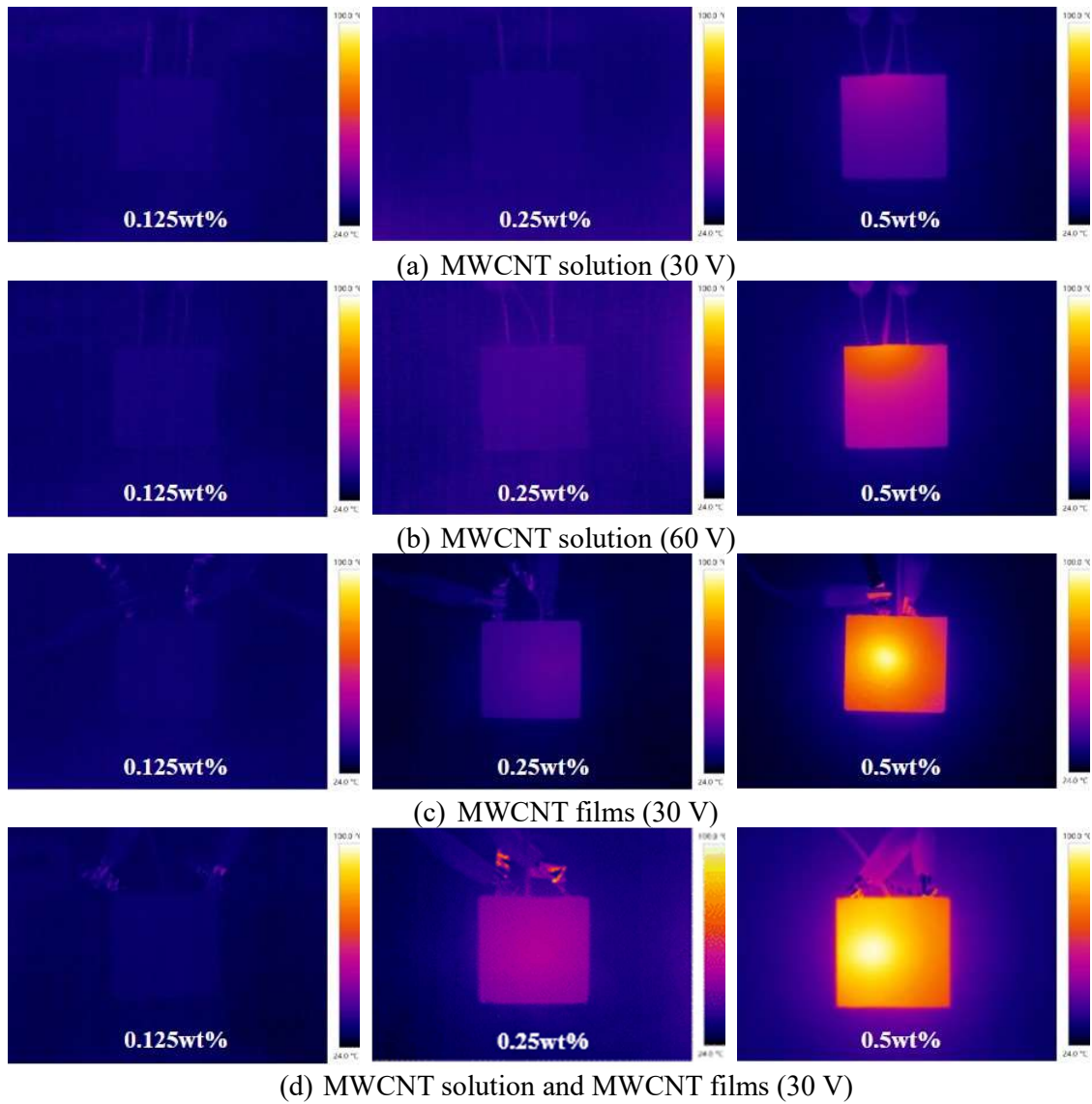


(e) MWCNT films of 28 curing days



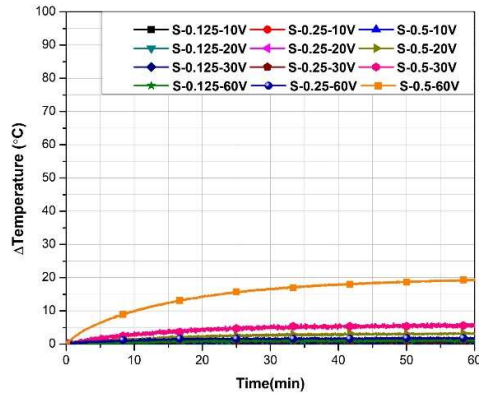
(f) MWCNT solution and MWCNT films of 28 curing days  
**Fig. 4. Variations in temperature upon application of voltage.**



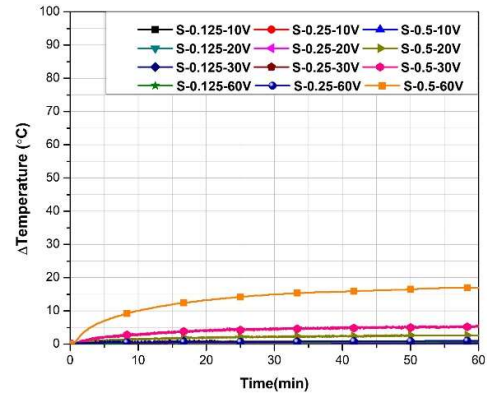


(d) MWCNT solution and MWCNT films (30 V)  
**Fig. 5. Thermal images of specimens after curing for 28 days.**

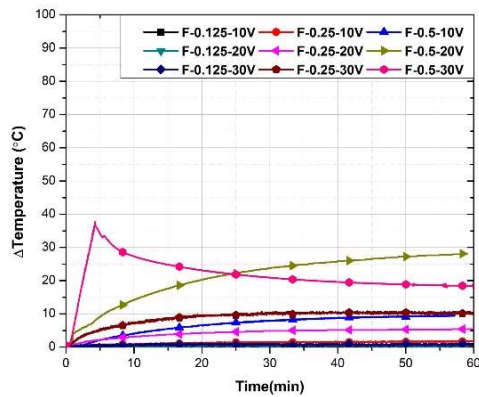
702



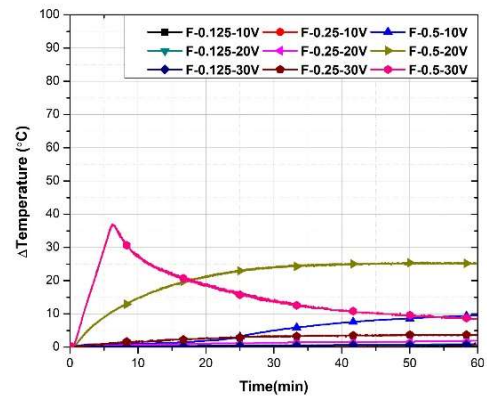
(a) MWCNT solution (curing for 7 days)



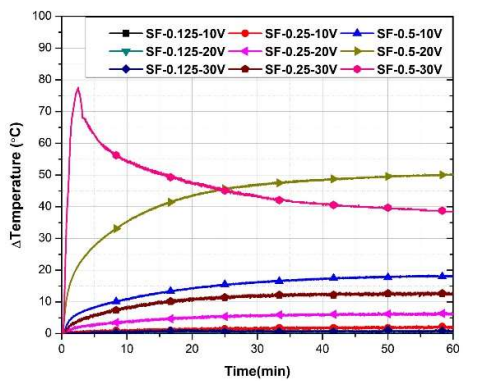
(b) MWCNT solution (curing for 28 days)



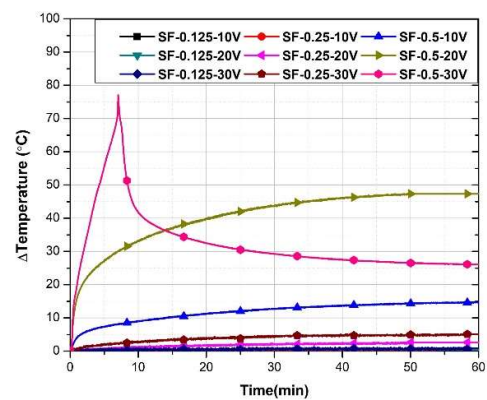
(c) MWCNT films (curing for 7 days)



(d) MWCNT films (curing for 28 days)

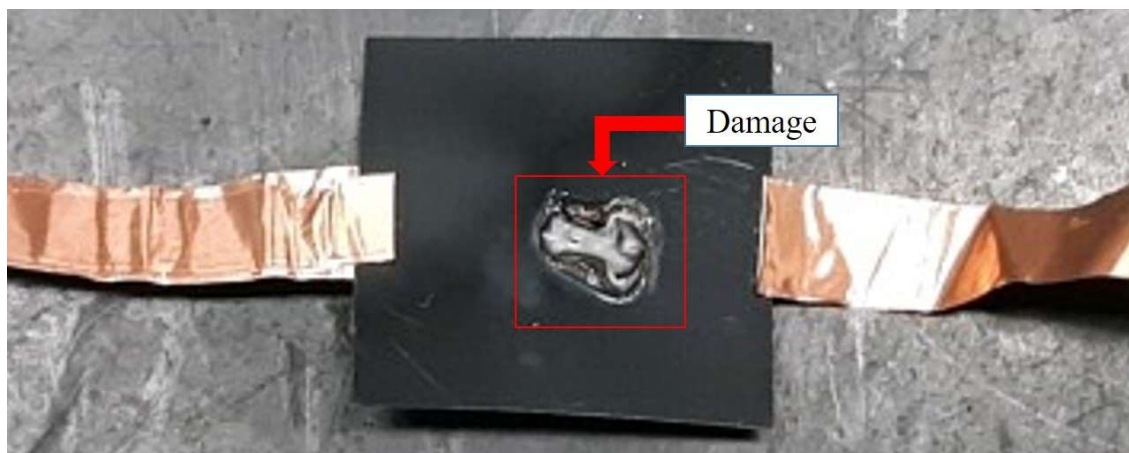


(e) MWCNT solution and MWCNT films (curing for 7 days)



(f) MWCNT solution and MWCNT films (curing for 28 days)

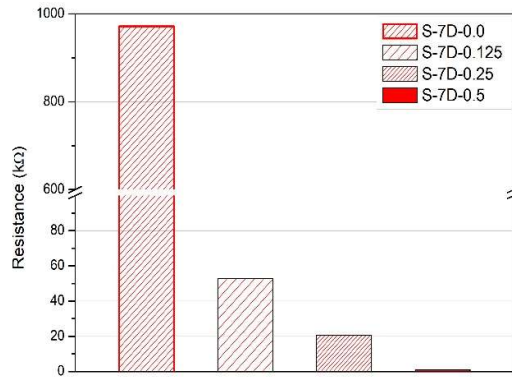
**Fig. 6. Temperature–time curves.**



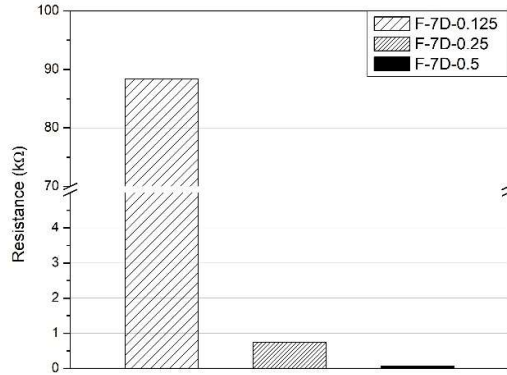
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**Fig. 7. Damaged MWCNT-covered film.**



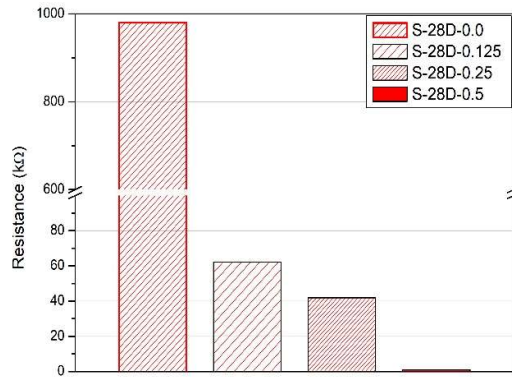
(a) MWCNT solution of 7 curing days



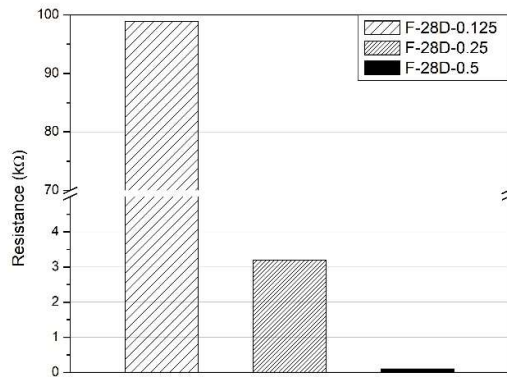
(b) MWCNT films of 7 curing days



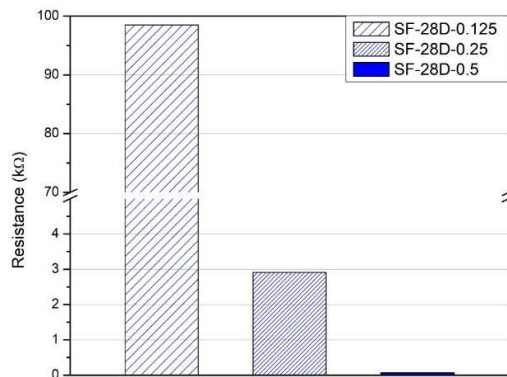
(c) MWCNT solution and MWCNT films of 7 curing days  
**Fig. 8. Electrical resistance for the parameter (continue).**



(d) MWCNT solution of 28 curing days

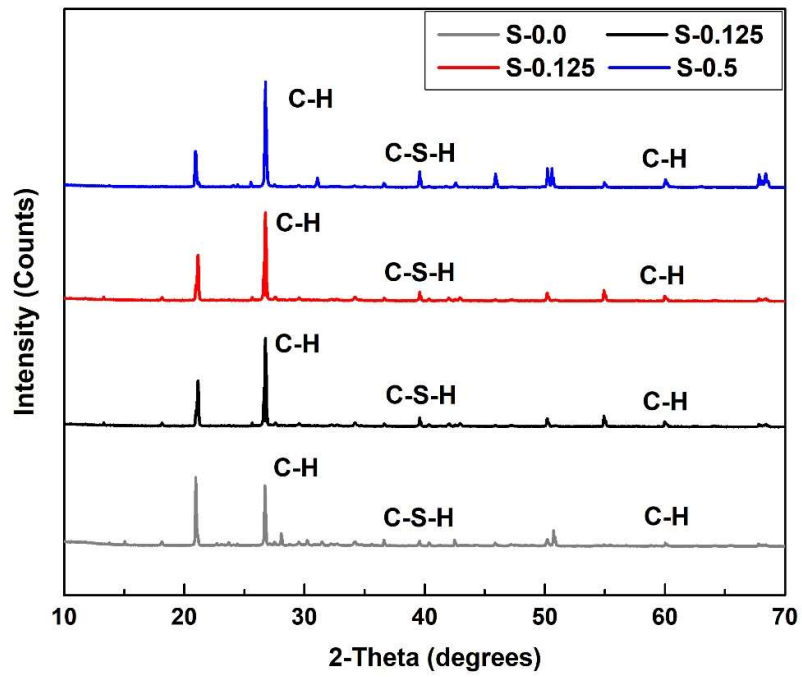


(e) MWCNT films of 28 curing days



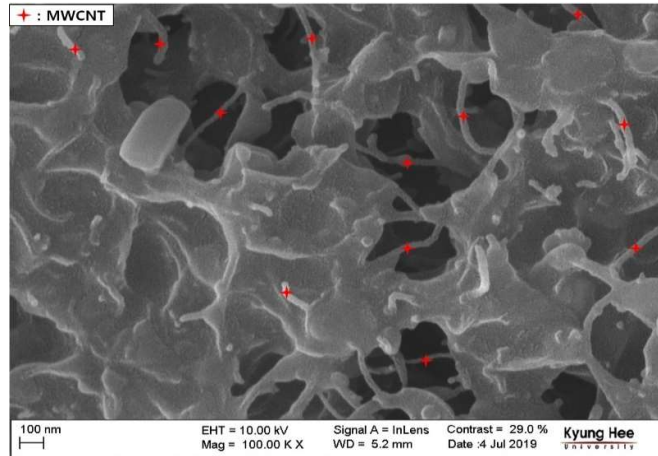
(f) MWCNT solution and MWCNT films of 28 curing days

**Fig. 8. Electrical resistance for the parameter.**

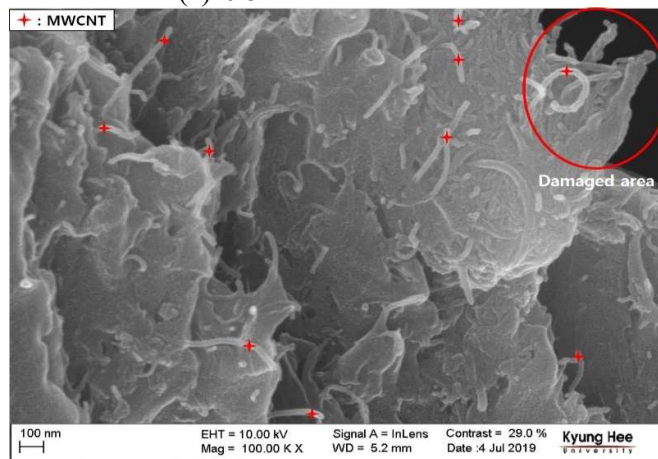


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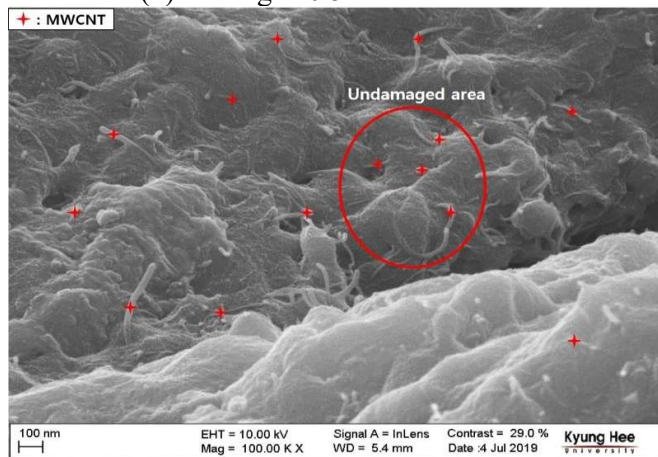
Fig. 9. XRD patterns of various samples.



(a) 0.5 wt% MWCNT solution



(b) Damaged 0.5 wt% MWCNT film



(c) Undamaged 0.5 wt% MWCNT film

**Fig. 10. FE-SEM images of various samples.**

710 **Table 1. Parameters used for temperature measurements**

Mixing Method	Specimen name	No. of curing days	MWCNT concentration(%)	Voltage (V)
MWCNT solution	S-7D-0.0	7	0.0	10/20/30/60
	S-7D-0.125		0.125	10/20/30/60
	S-7D-0.25		0.25	10/20/30/60
	S-7D-0.5		0.5	10/20/30/60
	S-28D-0.0	28	0.0	10/20/30/60
	S-28D-0.125		0.125	10/20/30/60
	S-28D-0.25		0.25	10/20/30/60
	S-28D-0.5		0.5	10/20/30/60
MWCNT films	F-7D-0.125	7	0.125	10/20/30
	F-7D-0.25		0.25	10/20/30
	F-7D-0.5		0.5	10/20/30
	F-28D-0.125	28	0.125	10/20/30
	F-28D-0.25		0.25	10/20/30
	F-28D-0.5		0.5	10/20/30
MWCNT solution and MWCNT films	SF-7D-0.125	7	0.125	10/20/30
	SF-7D-0.25		0.25	10/20/30
	SF-7D-0.5		0.5	10/20/30
	SF-28D-0.125	28	0.125	10/20/30
	SF-28D-0.25		0.25	10/20/30
	SF-28D-0.5		0.5	10/20/30

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713 **Table 2. Properties of MWCNT**

<b>Structure</b>	Diameter : 1nm~10nm length : 100nm~1cm
<b>Modulus of Elasticity</b>	0.27~0.95 TPa
<b>Tensile strength</b>	11~63 GPa
<b>Electrical conductivity</b>	$0.17\sim 2 \times 10^5$ S/cm
<b>Thermal conductivity</b>	3,000 W/mK
<b>Purity</b>	99 %
<b>Density</b>	1.33 g/cm

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716 **Table 3. Mixing ratios of various specimens**

Name	CNTs (g)		Water/cement ratio	Water (g)	Cement (g)	Sand (g)
	MWCNT solution	MWCNT films				
S-0.0	-	-	0.5	40	80	200
S-0.125	0.05	-		40	80	200
S-0.25	0.1	-		40	80	200
S-0.5	0.2	-		40	80	200
F-0.125	-	0.05		40	80	200
F-0.25	-	0.1		40	80	200
F-0.5	-	0.2		40	80	200
SF-0.125	0.025	0.025		40	80	200
SF-0.25	0.05	0.05		40	80	200
SF-0.5	0.1	0.1		40	80	200

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719 **Table 4. Results of heating tests**

Name	No. of curing days	MWCNT concentration (%)	$\Delta$ Temperature ( $^{\circ}$ C)			
			10 V	20 V	30 V	60 V
S-7D-0.0	7	0.0	0.2	0.3	0.6	1.0
S-7D-0.125		0.125	0.4	0.6	0.7	1.3
S-7D-0.25		0.25	0.6	0.8	1.0	1.8
S-7D-0.5		0.5	1.5	3.2	5.6	19.8
S-28D-0.0	28	0.0	0.1	0.2	0.3	0.5
S-28D-0.125		0.125	0.1	0.2	0.4	0.6
S-28D-0.25		0.25	0.15	0.3	0.75	0.9
S-28D-0.5		0.5	0.6	2.6	5.2	16.8
F-7D-0.125	7	0.125	0.3	0.4	0.7	-
F-7D-0.25		0.25	1.8	5.6	9.5	-
F-7D-0.5		0.5	9.5	25.2	37.2	-
F-28D-0.125	28	0.125	0.4	0.7	0.9	-
F-28D-0.25		0.25	0.6	1.9	3.7	-
F-28D-0.5		0.5	9.5	28.1	36.1	-
SF-7D-0.125	7	0.125	0.2	0.5	0.8	-
SF-7D-0.25		0.25	1.8	6.2	10.9	-
SF-7D-0.5		0.5	18.3	50.6	77.5	-
SF-28D-0.125	28	0.125	0.1	0.4	0.8	-
SF-28D-0.25		0.25	0.6	2.6	5.1	-
SF-28D-0.5		0.5	14.8	47.3	76.4	-

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721 **Table 5. Results of electrical resistance measurements**

<b>Name</b>	<b>No. of curing days</b>	<b>MWCNT concentration (%)</b>	<b>Resistance (k<math>\Omega</math>)</b>
S-7D-0.0	7	0.0	972
S-7D-0.125		0.125	52.8
S-7D-0.25		0.25	20.72
S-7D-0.5		0.5	0.9
S-28D-0.0	28	0.0	980
S-28D-0.125		0.125	62.1
S-28D-0.25		0.25	25.8
S-28D-0.5		0.5	0.98
F-7D-0.125	7	0.125	88.4
F-7D-0.25		0.25	0.74
F-7D-0.5		0.5	0.075
F-28D-0.125	28	0.125	98.9
F-28D-0.25		0.25	3.2
F-28D-0.5		0.5	0.1
SF-7D-0.125	7	0.125	82.9
SF-7D-0.25		0.25	0.71
SF-7D-0.5		0.5	0.05
SF-28D-0.125	28	0.125	98.5
SF-28D-0.25		0.25	2.91
SF-28D-0.5		0.5	0.067

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724 **Declarations of conflict of interest:** There are no conflicts of interest to be declared.