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Authors

Pool, V. L.
Klem, M. T.
Arenholz, E.
et al.

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Site Determination of Mn Doping in Protein Encapsulated γ -Fe₂O₃ Nanoparticles

V. L. Pool^{1, 4}; *M. T. Klem*^{2, 4}; *E. Arenholz*⁵; *C. Jolley*^{2, 4}; *T. Douglas*^{2, 4}; *M. Young*^{3, 4}; *Y. U. Idzerda*^{1, 4}

1. Physics, Montana State University, Bozeman, MT, USA.
2. Chemistry and Biochemistry, Montana State University, Bozeman, MT, USA.
3. Plant Sciences and Pathology, Montana State University, Bozeman, MT, USA.
4. Center for Bio-inspired Nanomaterials, Montana State University, Bozeman, MT, USA.
5. Advanced Light Source, LBNL, Berkeley, CA, USA.

Separately controlling the properties of iron oxide nanoparticles without modifying their size is important for actualizing many of the exciting applications of protein encapsulated magnetic nanoparticles. In addition, determining the effect that the nanoparticle size has on material properties is fundamentally interesting. One important way to make these modifications is through doping with other magnetic transition metals.

In this study, Mn has been doped (0-33%) into 6.7 nm, γ -Fe₂O₃ nanoparticles grown inside the horse-spleen ferritin (HSF) protein and compared to similarly protein encapsulated pure γ -Fe₂O₃ and Mn-oxide nanoparticles to quantify the variation in magnetic behavior and to determine the Mn doping site [1]. By using soft-X-ray absorption spectroscopy (XAS), soft-X-ray magnetic circular dichroism (XMCD), hard X-ray scattering position distribution functions (PDFs), and frequency dependent Alternating Current Magnetic Susceptibility (ACMS), we have ascertained that the Mn dopant is substituting preferentially as Mn²⁺ and prefers the octahedral site in the γ -phase Fe₂O₃ spinel structure. The measured Mn L₂₃ XAS spectra (shown in Figure 1) are compared to measured reference powders and molecular-orbital calculations supporting this conclusion of the Mn dopant substitution site [2]. We find that the Mn L₂₃ XAS multiplet structure for the nanoparticles is simpler than for our bulk standards, complicating this identification but suggesting that the nanoparticle lattices are relaxed from the distortions present in the bulk. These Mn XAS spectra are nearly identical to those for similarly prepared Mn-oxide nanoparticles synthesized in HFN. For these nanoparticles, hard X-ray scattering position distribution functions (PDFs) identify that these Mn-oxide particles have formed into the spinel structure.

A simple picture of the effect of doping into the various Fe-sites of the γ -Fe₂O₃ spinel structure suggests that there will be a significantly different magnetic moment per iron lattice position depending on the site favored by the dopant atom. By performing Fe L₂₃ edge XMCD on the particles, it is possible to obtain the magnetic moment per iron lattice position and provide insight into the site occupation of the remaining Fe atoms. (It should be noted that straight-forward magnetometry measurements cannot be properly mass normalized due to the presence of empty protein shells.) This result (shown as dots in Figure 2) displays a rapid decline in the average Fe moment, more rapid than a simple substitution model would predict (dashed line). Moreover, both the Fe L₃ XMCD lineshape (INSET of Figure 2) and the Mn L₃ XAS lineshape show that as the Mn

concentration increases, the occupation by Fe of the Fe^{3+} octahedral site decreases while that of Mn in the octahedral site increases [3]. Finally, from a Neel-Arrhenius plot of the frequency dependent ACMS measurements we conclude that these particles are non-interacting and that the anisotropy energy for each particle is linearly reduced with increasing Mn concentration. For 33% Mn doping, the anisotropy energy is nearly one half that of pure $\gamma\text{-Fe}_2\text{O}_3$.

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- [1] T. Douglas and M. Young, Nature 393, 152 (1998).
 [2] J.-S. Kang, et al, Phys. Rev. B 77, 035121 (2008).
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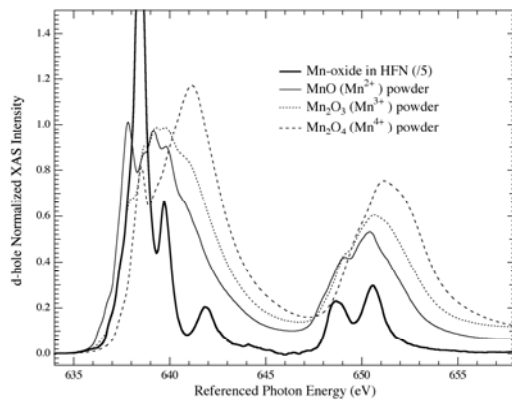


FIGURE 1: Mn L_{23} XAS spectra for Mn doped into 6.7 nm diameter $\gamma\text{-Fe}_2\text{O}_3$ nanoparticles (heavy line) and for Mn oxide reference powders (for comparison).

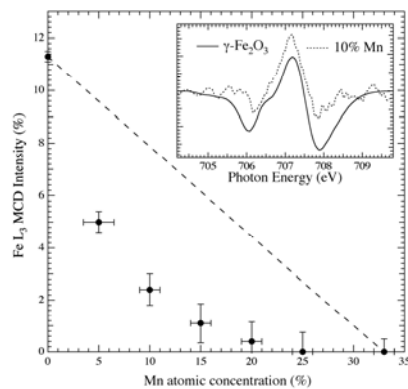


FIGURE 2: Fe L_3 XMCD intensity as a function of Mn concentration (exp. – circles, model – dashed line). INSET: Normalized XMCD spectra for 0% and 10% Mn doping.