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GERMANIUM BLOCKED IMPURITY BAND (BIB) DETECTORS

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Presented at the IR Detector Technology Workshop,  
Mountain View, CA, February 7-9, 1989

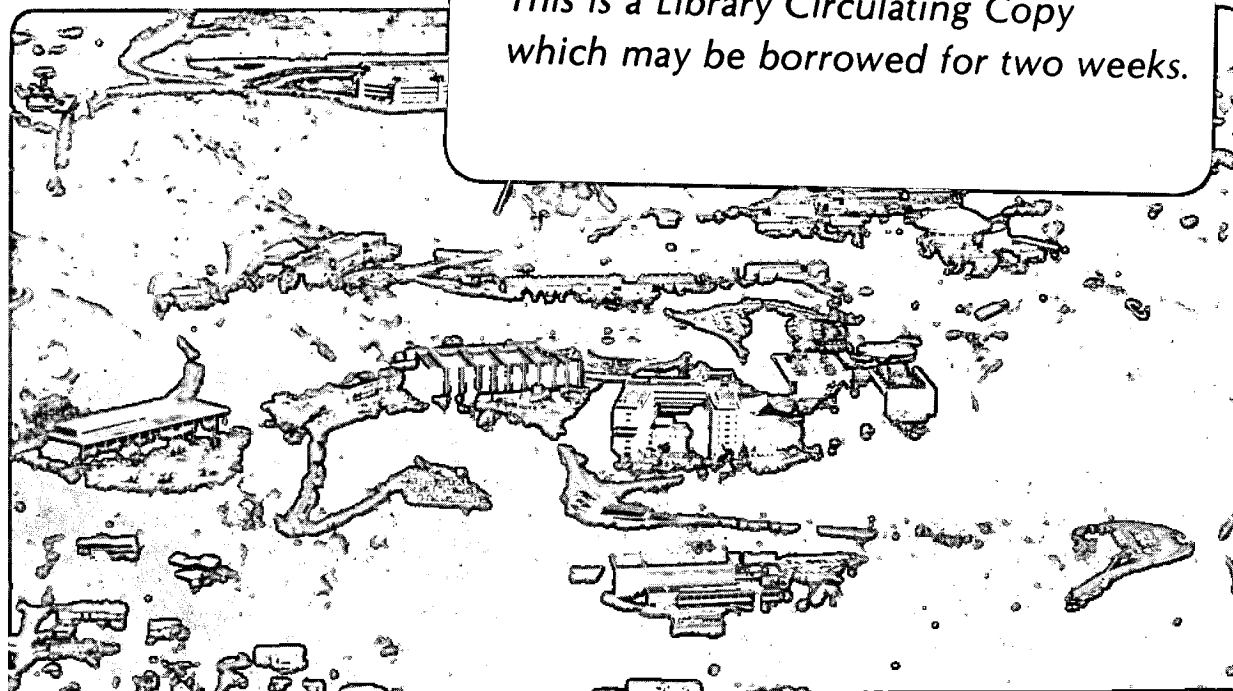
### Germanium Blocked Impurity Band (BIB) Detectors

E.E. Haller, H. Baumann, J. Beeman, W.L. Hansen, P.N. Luke,  
M. Lutz, C.S. Rossington, and I.C. Wu

February 1989

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## **Germanium Blocked Impurity Band (BIB) Detectors**

**E. E. Haller<sup>1,2</sup>, H. Baumann<sup>1,2</sup>, J. Beeman<sup>2</sup>, W. L. Hansen<sup>2</sup>, P. N. Luke<sup>2</sup>,  
M. Lutz<sup>1,2</sup>, C. S. Rossington<sup>1,2,3</sup>, I. C. Wu<sup>1,2</sup>**

**Presented at the IR Detector Technology Workshop,  
February 7-9, 1989, NASA Ames Research Center**

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# **CONTENTS**

## **1 . Introduction**

## **2 . Ge BIB**

## **3 . Ge BIB Detector Development**

### **3.1. Epitaxial Blocking Layer Devices**

#### **3.1.1. Ge epitaxy**

#### **3.1.2. Characterization of epi layers**

#### **3.1.3. Preliminary detector test results**

### **3.2. Ion Implanted BIB Detectors**

## **4 . Conclusions**

## **ACKNOWLEDGMENT**

**This work was supported by NASA Contract W-14,606X under an interagency agreement with the U.S. Department of Energy under Contract No. DE-AC03-76SF00098.**

# **1. INTRODUCTION**

- **Extrinsic, photoconductive semiconductor detectors cover the infrared spectrum from a few  $\mu\text{m}$  up to 250  $\mu\text{m}$ .**
- **Photoconductors exhibit high responsivity and low noise equivalent power.**
- **The Si blocked impurity band (BIB) detector invented by M. D. Petroff and M. G. Stapelbroek has a number of advantages over standard bulk photoconductors. These include:**
  - **smaller detection volume leading to a reduction of cosmic ray interference**
  - **extended wavelength response because of dopant wavefunction overlap**
  - **photoconductive gain of unity**

## **2. Ge BIB**

- **The success of Si BIB detectors has been a strong incentive for the development of Ge BIB detectors.**
- **The advantages of Si BIB detectors stated above should, in principle, be realizable for Ge BIB detectors.**
- **If Ge BIB detectors can be made to work out to 250  $\mu\text{m}$  with high responsivity and sufficiently low dark current, they could replace stressed Ge:Ga photoconductors.**
- **Can the dark current be reduced to acceptable levels?**



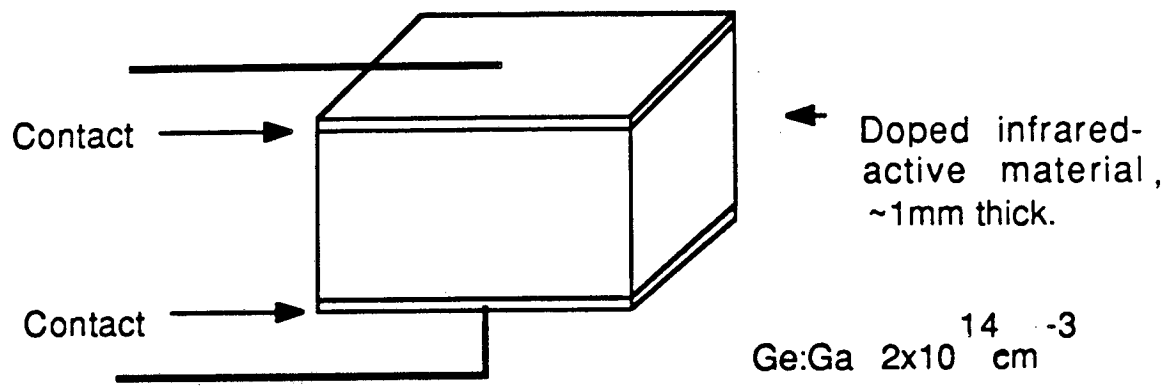


Figure 1(a). Schematic of conventional detector.

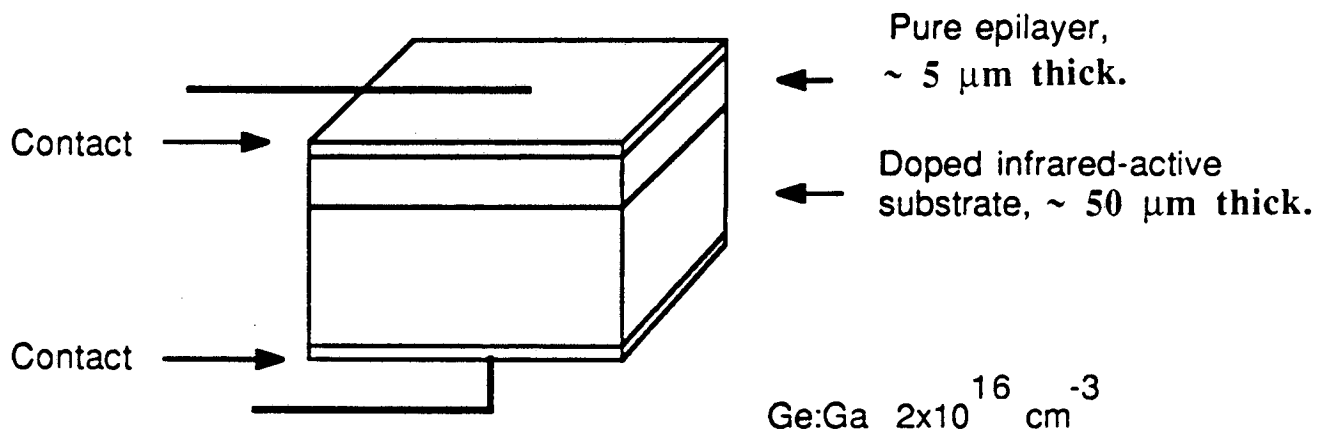


Figure 1(b). Schematic of BIB detector.

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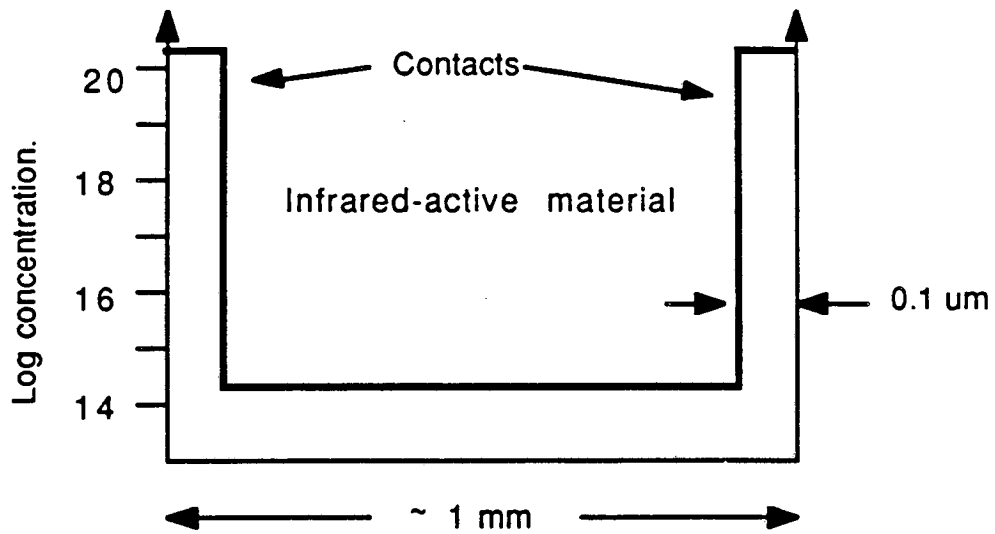


Figure 2(a). Doping levels in a conventional Ge detector.

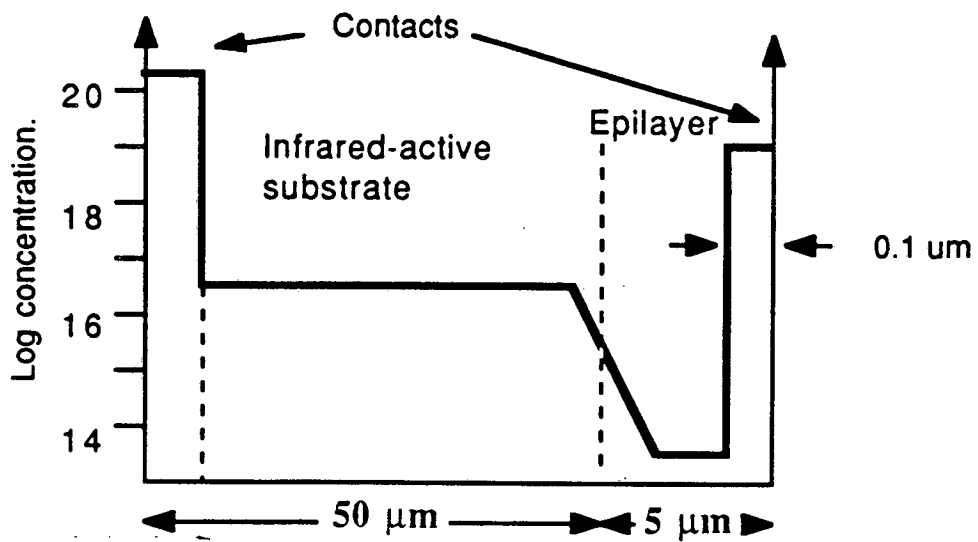
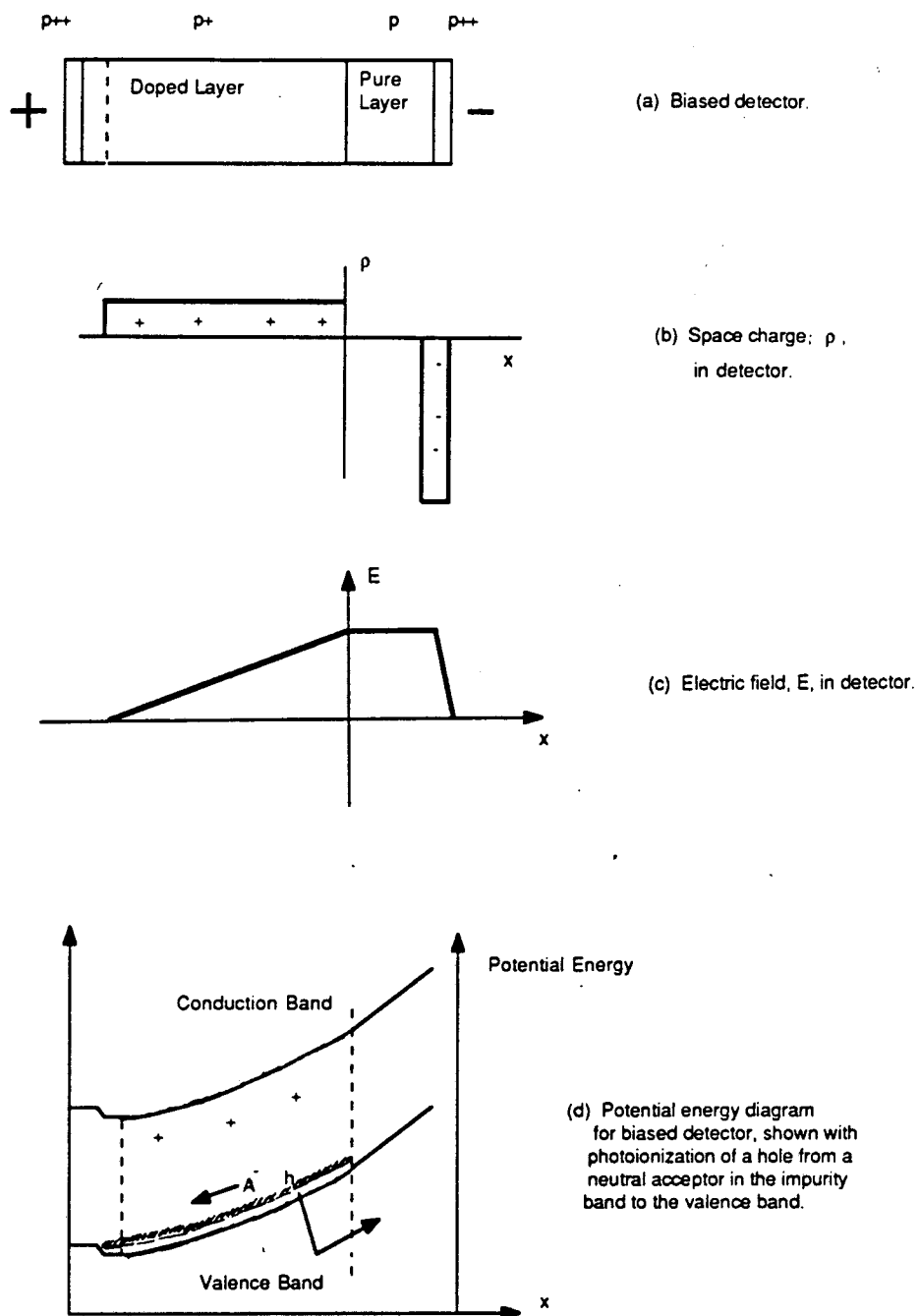


Figure 2 (b). Doping levels in a Ge BIB detector.

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**Fig. 3. Schematics of space charge, electric field and potential energy for a reverse biased p-type BIB detector.**

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### **3 . Ge BIB DETECTOR DEVELOPMENT**

#### **3.1. Epitaxial Blocking Layer Devices**

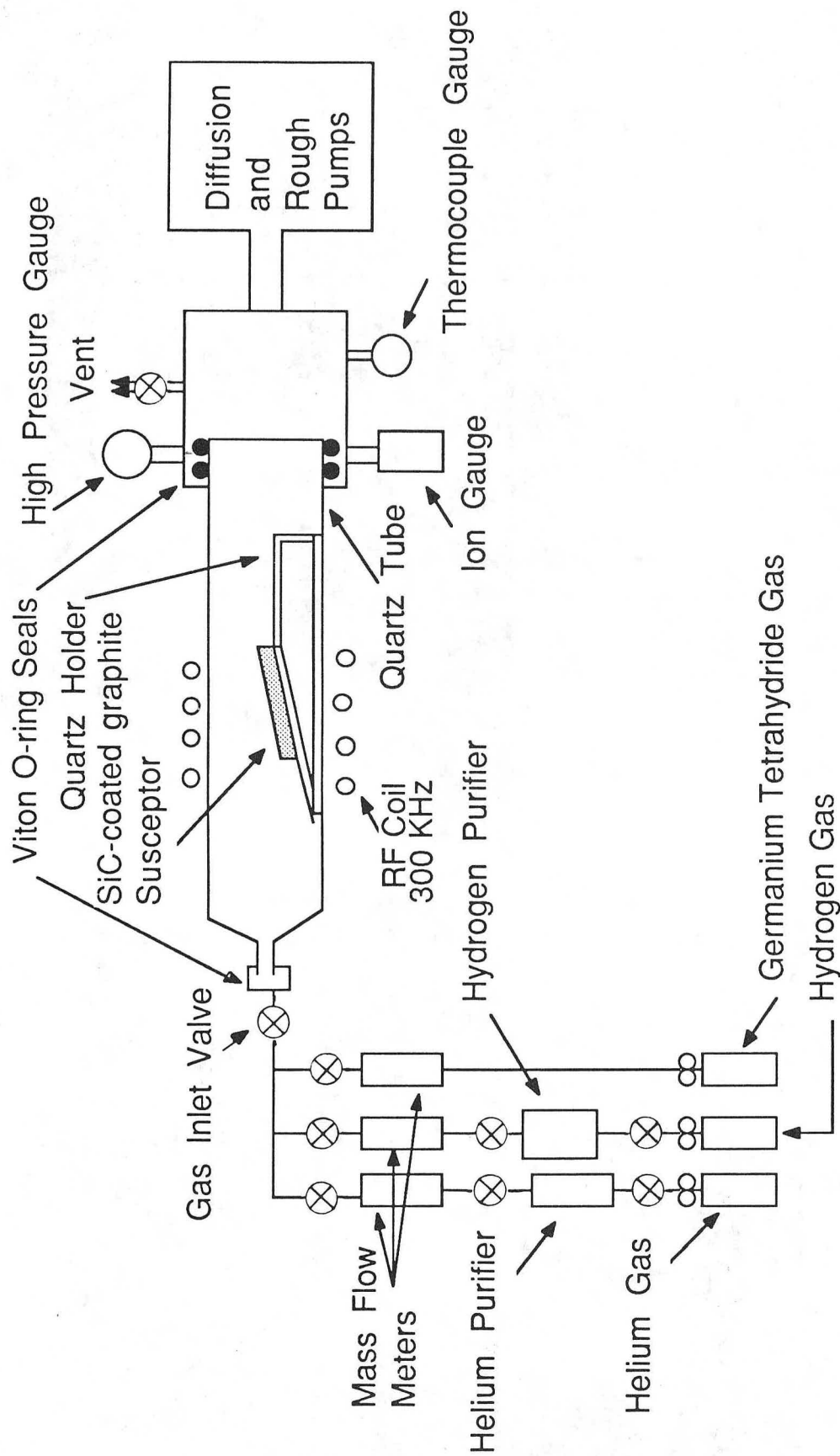
##### **3.1.1. Ge epitaxy**

- Whereas Si epitaxy techniques have been developed to a very high degree of perfection, Ge epitaxy has been attempted only on a few occasions.
- Ge chemistry is very different from Si chemistry.
- Ultra-pure Ge compounds [ $\text{Ge}(\text{CH}_3)_4$ ,  $\text{Ge}(\text{C}_2\text{H}_5)_4$ ] are being developed for III-V semiconductor technology. They may be useful to Ge epitaxy.

## **Substrate choice and preparation**

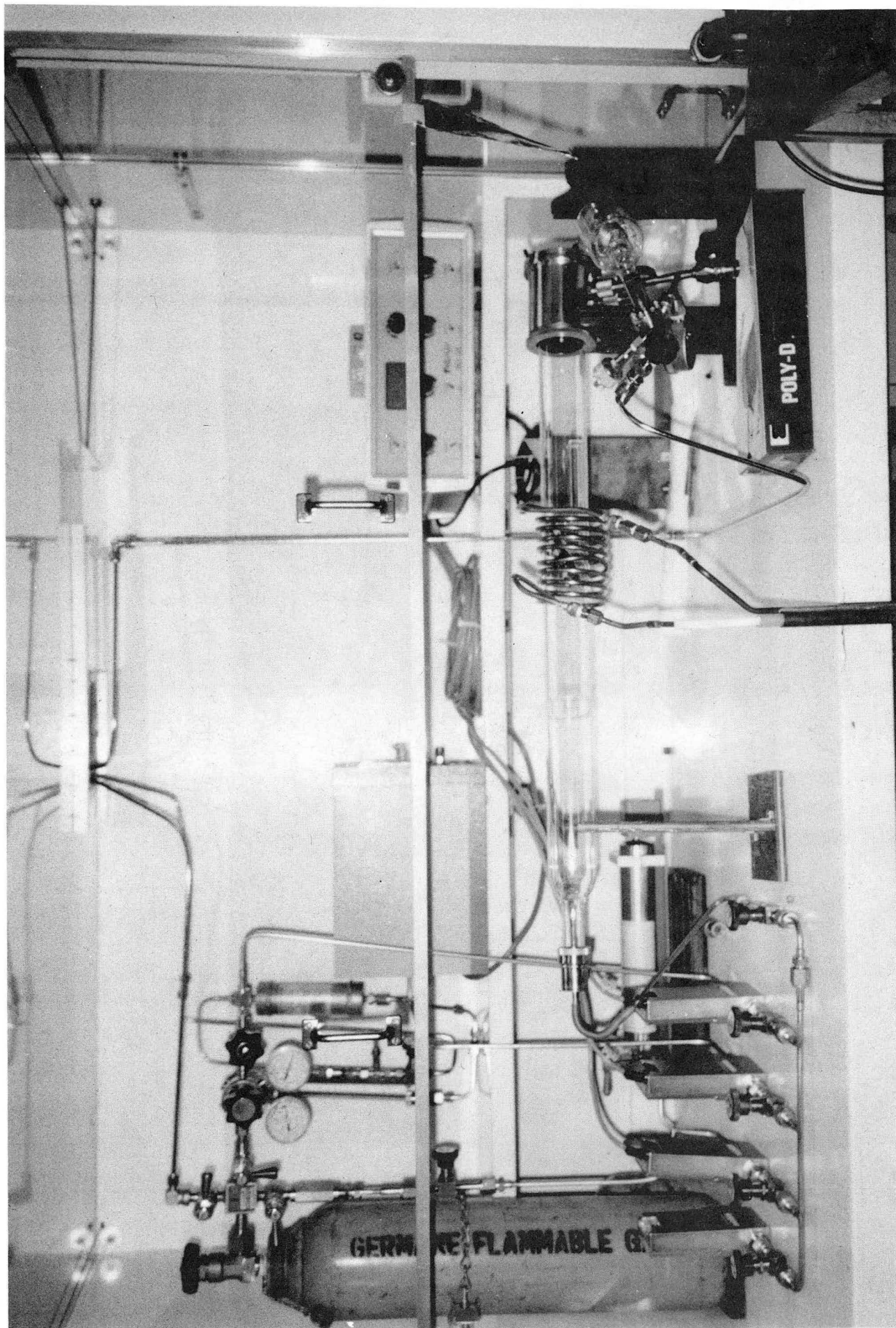
- We have used a number of different crystals with various crystallographic orientations in the development of Ge epitaxy:
  - n-type wafers ( $\sim 10^{11} \text{ cm}^{-3}$ ) are used for the electrical characterization of the epitaxial layers which are typically p-type because of residual copper contamination (junction isolation).
  - p-type wafers ( $\sim 10^{15} \text{ cm}^{-3}$ ) are used for I-V comparison tests with conventional photoconductors.
  - p-type wafers ( $\sim 2 \times 10^{16} \text{ cm}^{-3}$ , low compensation) are used for Ge BIB detectors.

- **Wafer polishing process:**
  - **mechanical planar lapping with alumina slurry.**
  - **mechano-chemical polishing with syton containing  $\text{H}_2\text{O}_2$ .**
  - **brief etch in  $\text{HNO}_3\text{:HF}$  (3:1) followed by soak in HF (1% in  $\text{H}_2\text{O}$ ) to remove oxides.**
- **Epitaxy:**
  - **first experiments with atmospheric pressure vapor phase epitaxy (VPE). Disadvantage: high substrate temperature,  $\text{H}_2$  diluted feed gas (contamination, diffusion of dopants into the blocking layer).**
  - **current experiments are performed with low pressure VPE. Advantage: low substrate temperature.**



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Fig. 4. Schematic of horizontal VPE apparatus. Quartz tube is 5.7 cm O.D. x 75 cm long.



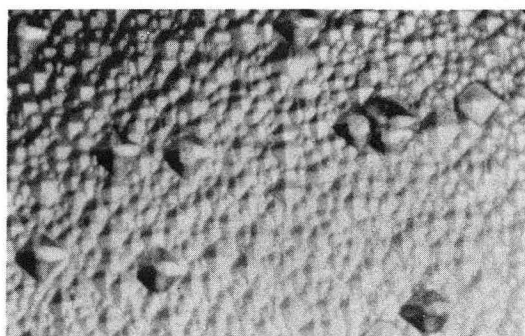
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Fig. 5. Photograph of the horizontal VPE chamber.



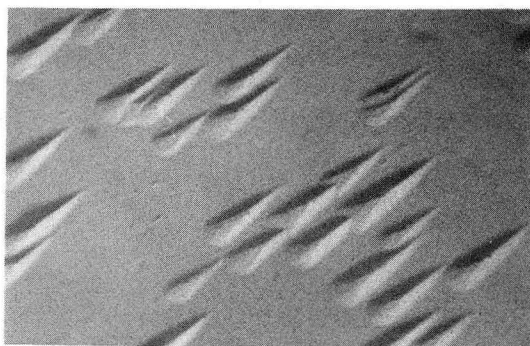
### **3.1.2. Characterization of epi layers**

- **Optical micrographs**
- **Variable temperature Hall effect and resistivity**
- **Rutherford backscattering (channeling) spectrometry (RBS)**
- **Secondary ion spectrometry (SIMS)**
- **Spreading resistance measurements**



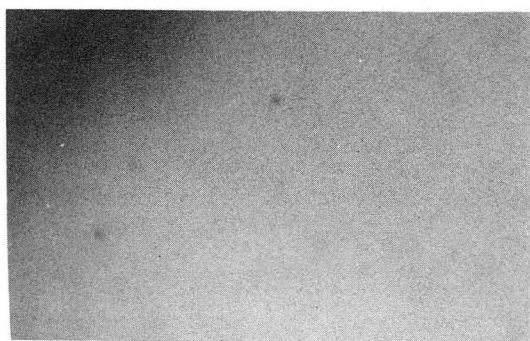
0.03 mm

(100)  $N_D - N_A = 1 \times 10^{14} / \text{cm}^3$   
 580 °C; 5 sccm  $\text{GeH}_4$ ,  
 with  $\text{H}_2$  reduction step,  
 polycrystalline deposition



0.03 mm

(113)  $N_D - N_A = 2 \times 10^{14} / \text{cm}^3$   
 580 °C; 5 sccm  $\text{GeH}_4$ ,  
 with  $\text{H}_2$  reduction step,  
 no growth (etching)



0.03 mm

(113)  $N_D - N_A = 5 \times 10^{11} / \text{cm}^3$   
 550 °C; 10 sccm  $\text{GeH}_4$ ,  
 no  $\text{H}_2$  reduction step,  
 single crystal deposition

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Fig. 6. Optical micrographs of Ge epi layers.

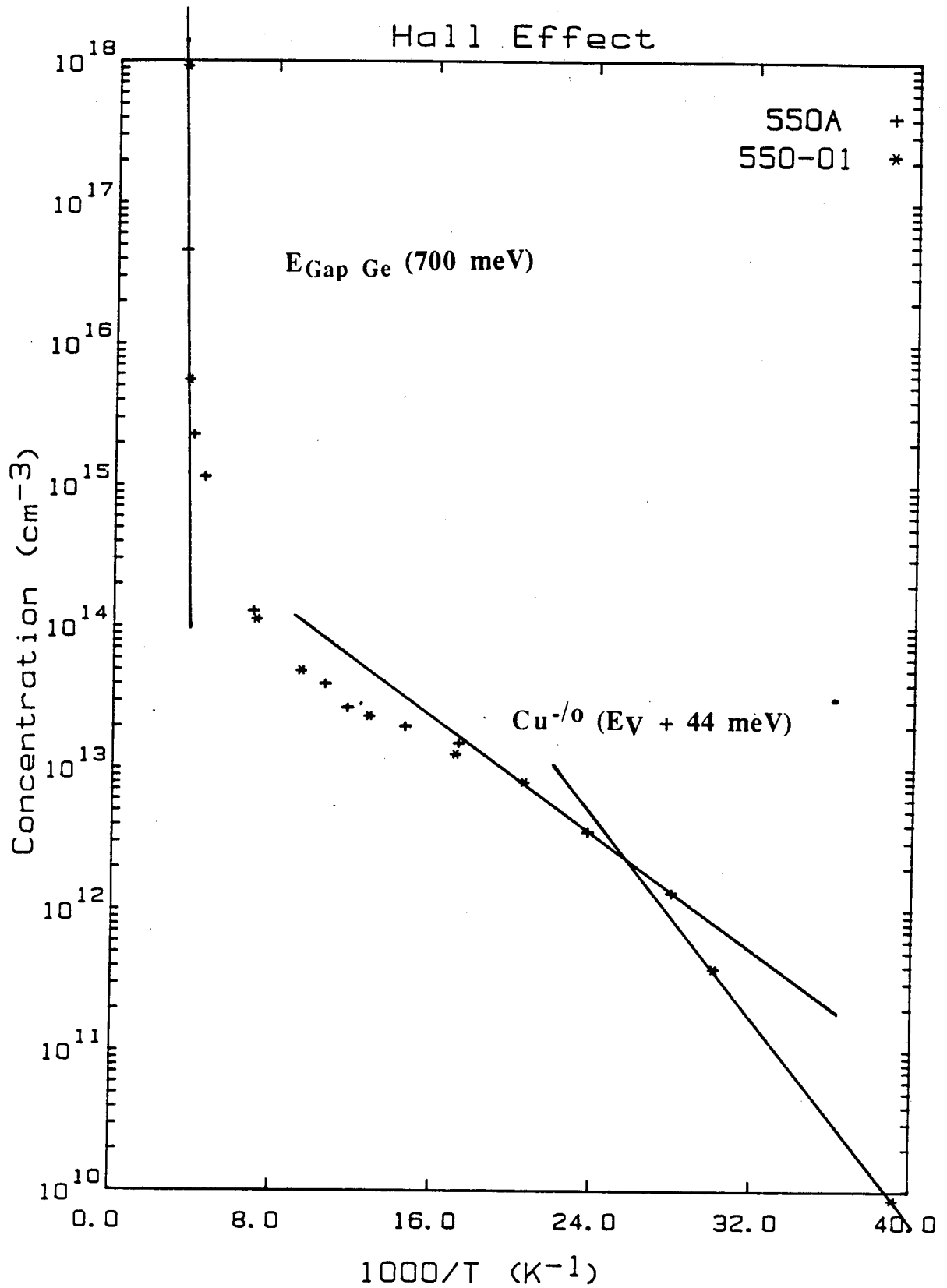


Fig. 7. Variable temperature Hall effect measurements of a Ge epilayer on an n-type [113] substrate. The hole freeze-out curves indicate a light copper contamination. The two curves (+, \*) are measurements of the same sample and demonstrate reproducibility.

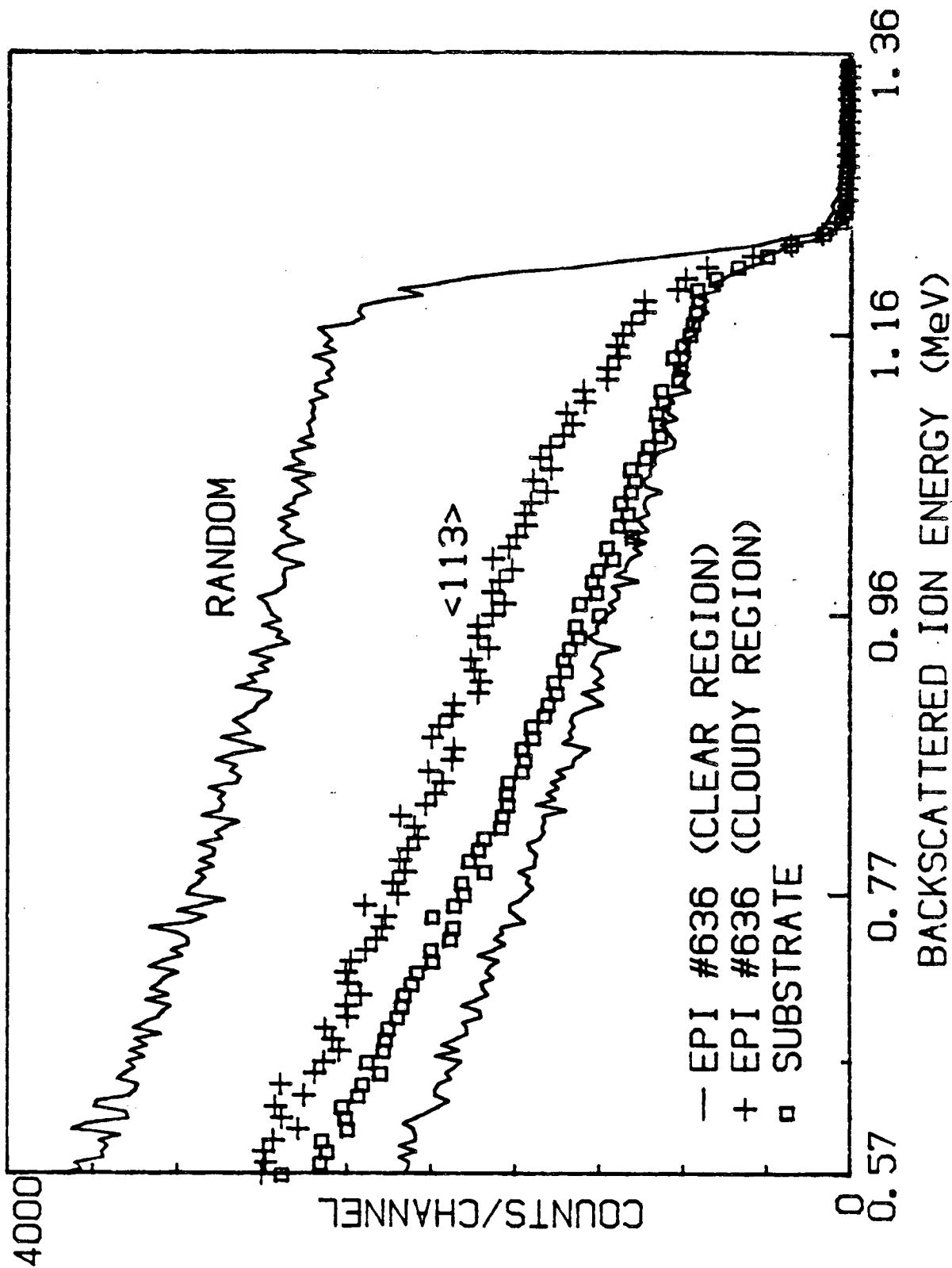


Fig. 8. RBS channeling spectra of a Ge epi film (#636). The "cloudy" region shows significant dechanneling indicating a high defect concentration.

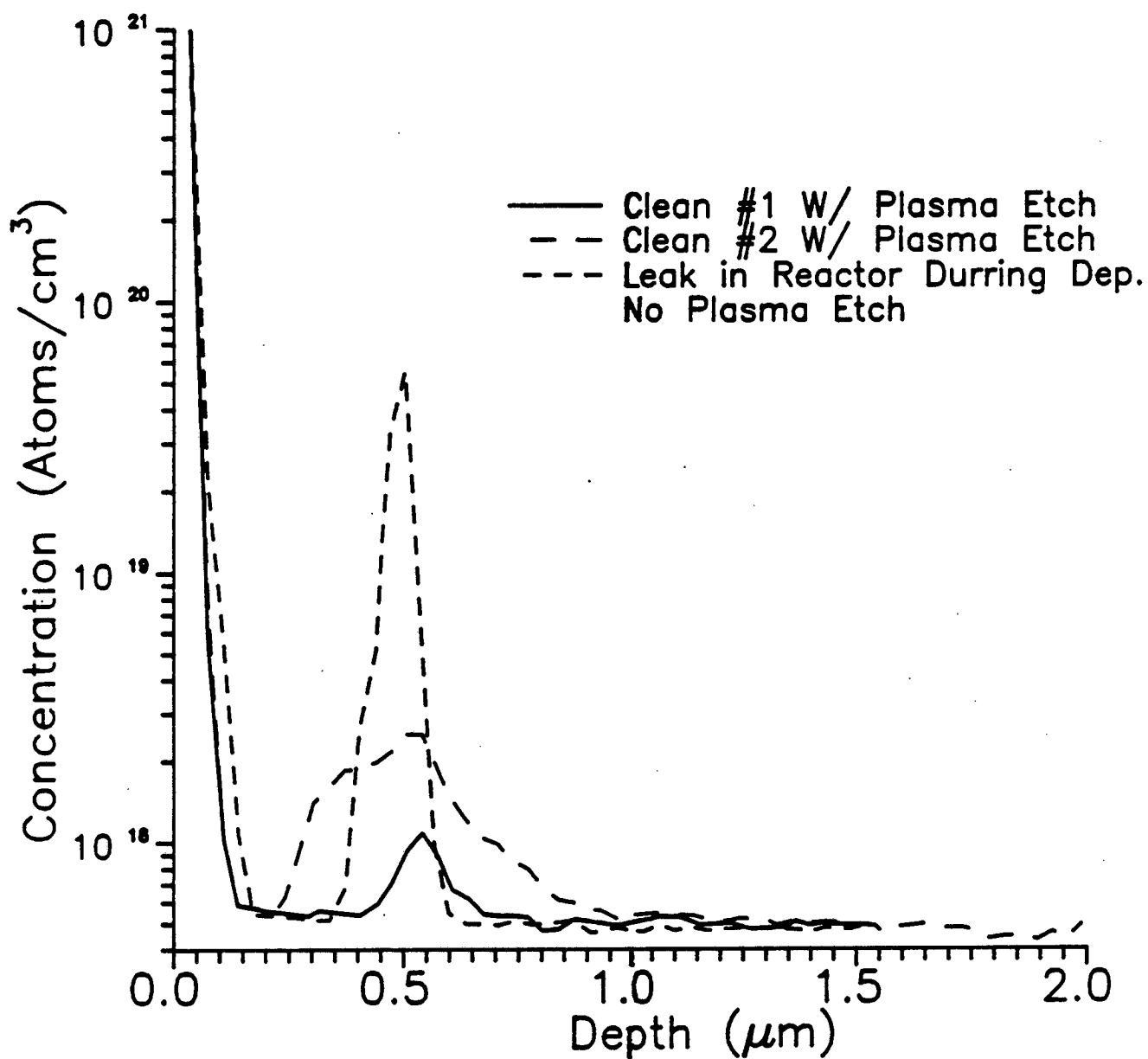


Fig. 9. SIMS of LPVPE Epi Films:  
Oxygen Concentration

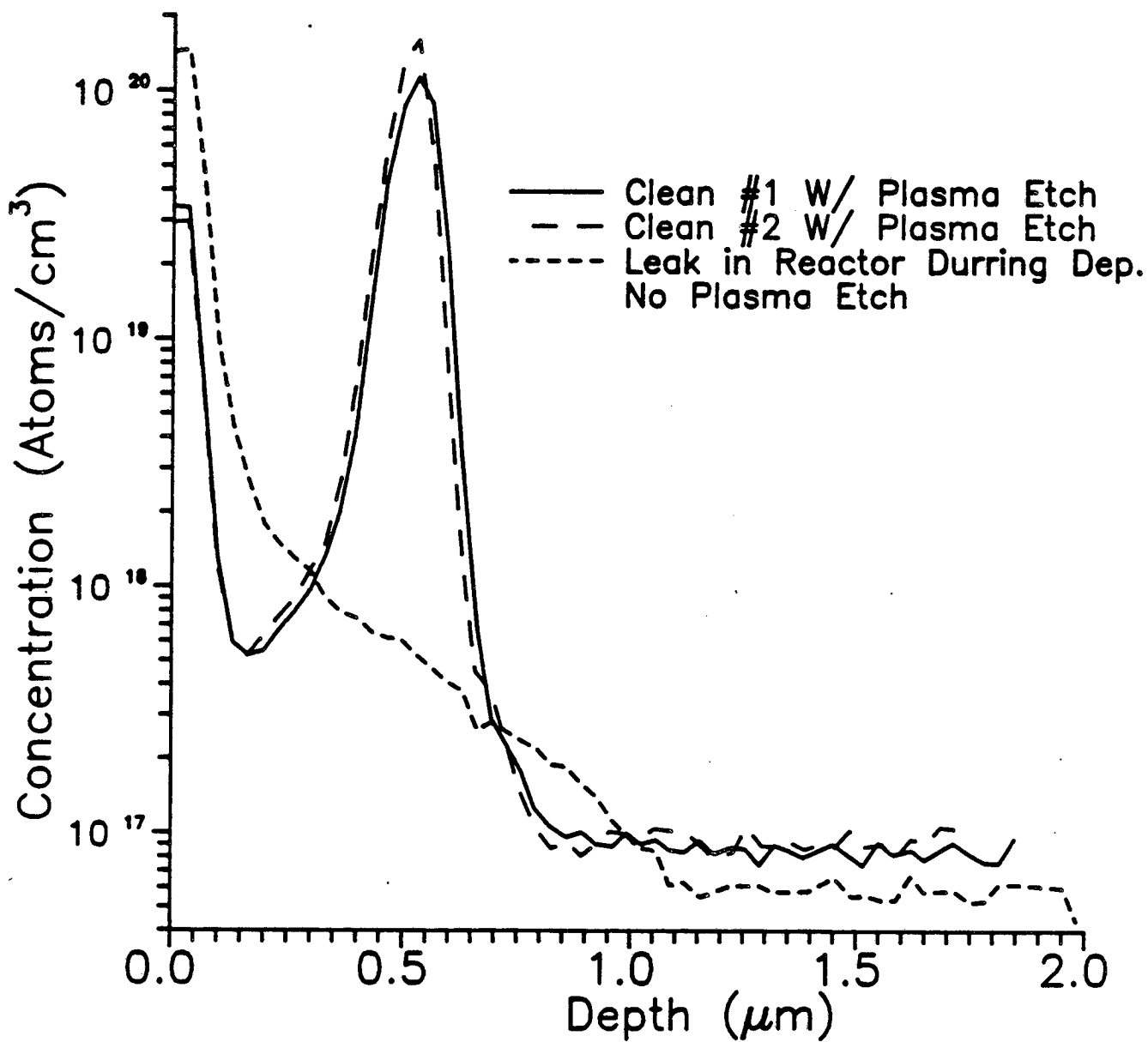


Fig. 10. SIMS of LPVPE Epi Films:  
Carbon Concentration

### **3.1.3. Preliminary detector test results**

- **Responsivity**
- **Dark current**

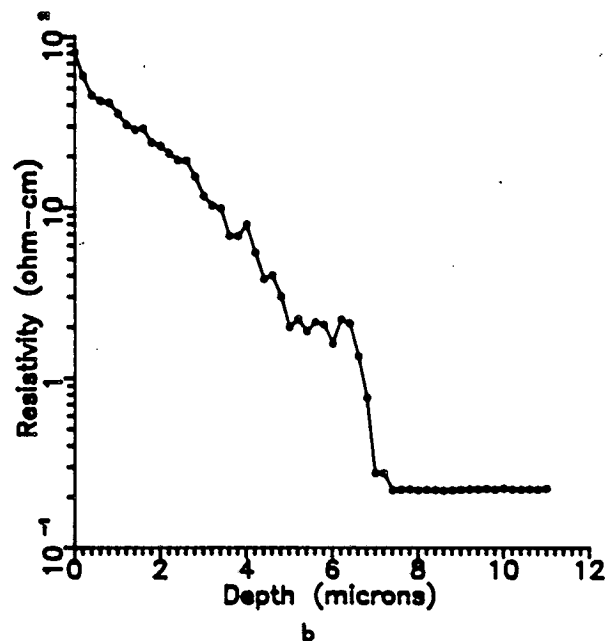
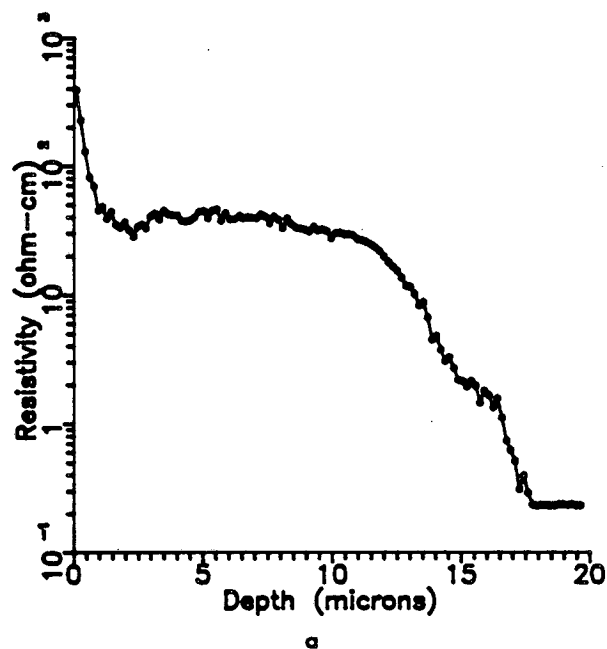


Fig. 11. Spreading resistance as a function of depth from the epilayer surface for: (a) an area of epilayer close to the leading edge of the wafer in II-16 where the growth rate was  $\sim 0.06 \mu\text{min}^{-1}$ , and (b) an area of epilayer farthest from the leading edge of the same wafer where the growth rate was  $\sim 0.02 \mu\text{min}^{-1}$ . The slight rise in resistivity at the very surface is due to the native oxide.

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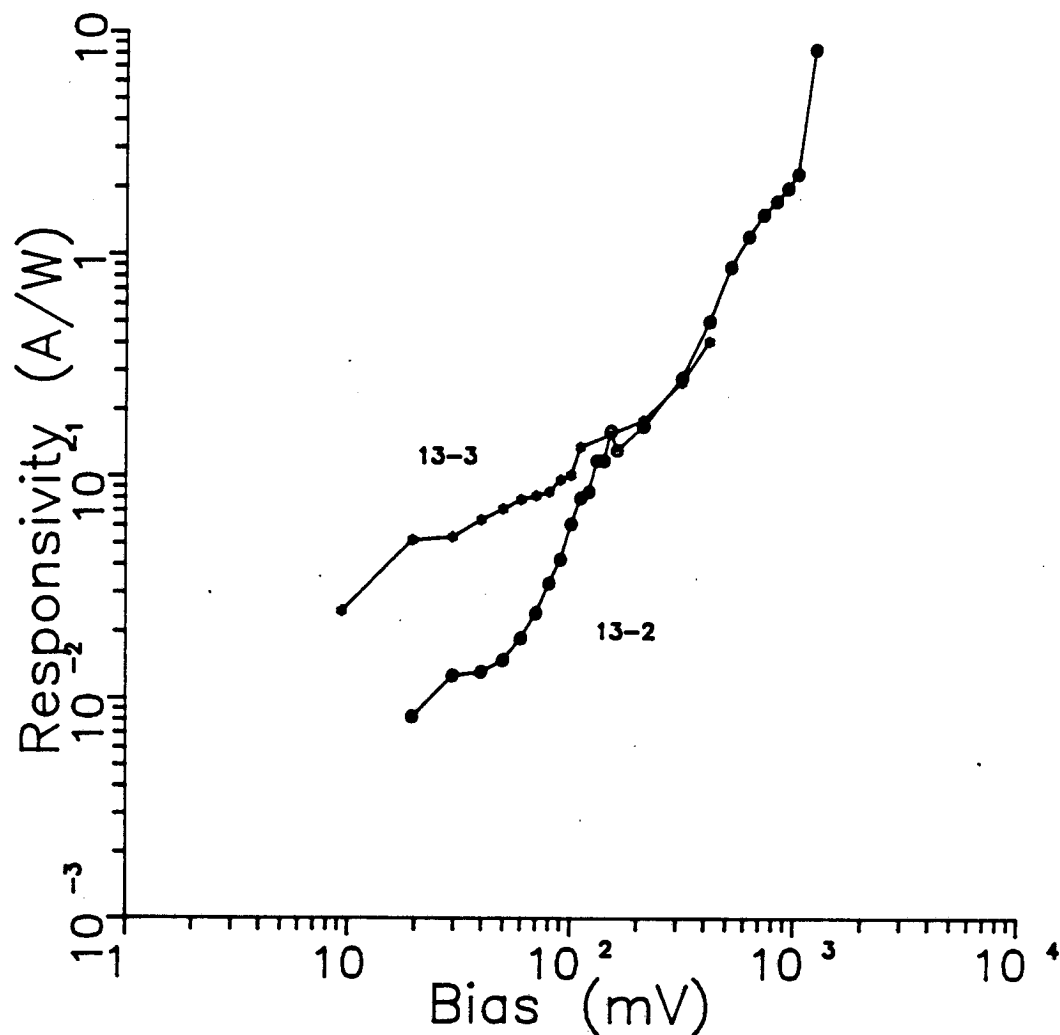


Fig. 12. Responsivity as a function of bias for detectors 13-2 and 13-3 at 2.3 K under reverse bias. The substrate material is moderately doped ( $5 \times 10^{15} \text{ cm}^{-3}$ ). Such material exhibits hopping conduction but does not have extended wavelength spectral response. Tests were performed with a narrow band filter at  $\lambda = 98.9 \mu\text{m}$ .

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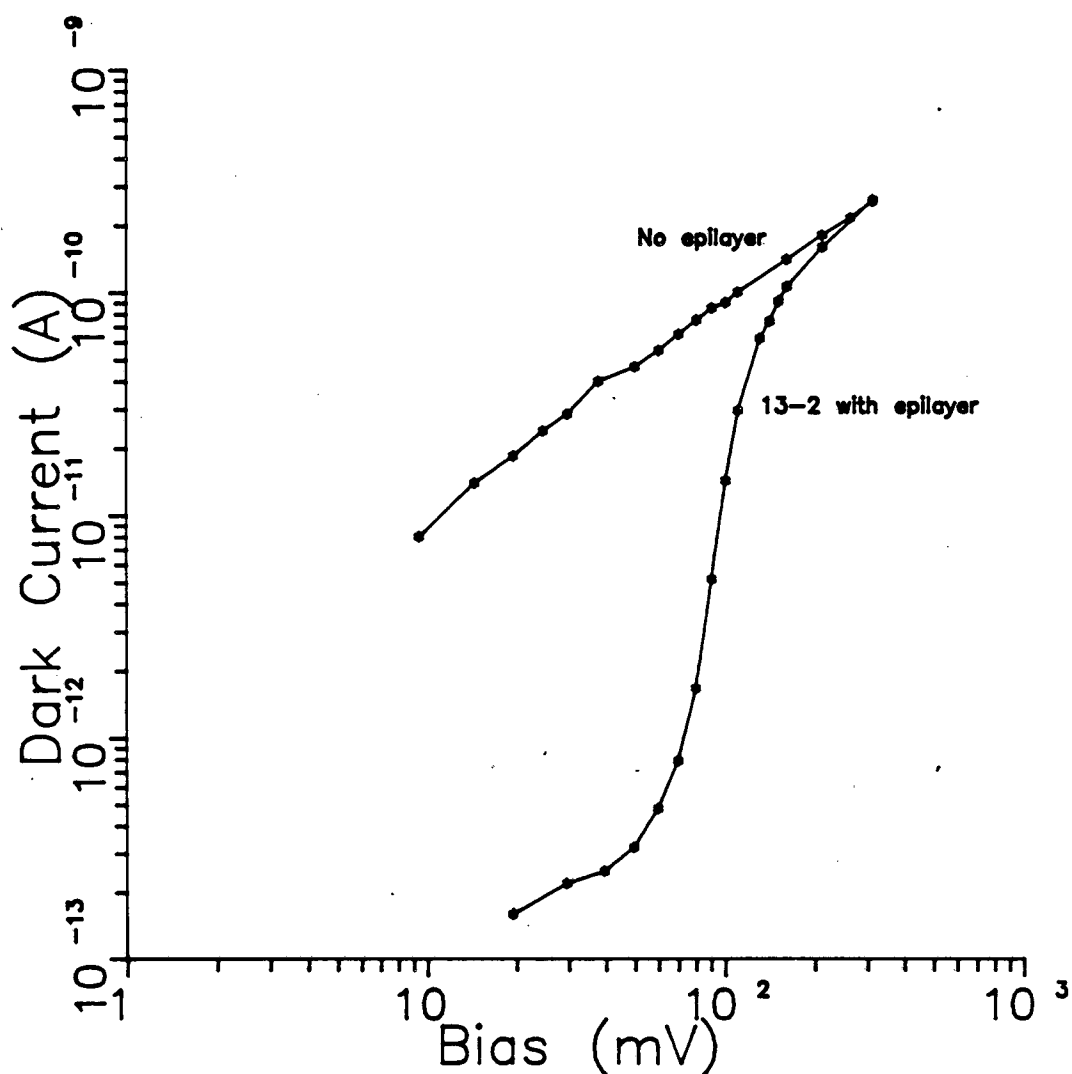


Fig. 13. Dark current as a function of detector bias for detector 13-2 with an epilayer and for the same "detector" without an epilayer at 2.3 K under reverse bias. Below a bias of  $\sim 100$  mV, the blocking layer effectively reduces hopping conduction in this moderately doped material ( $N_A - N_D = 5 \times 10^{15} \text{ cm}^{-3}$ ).

XBL 884-1160

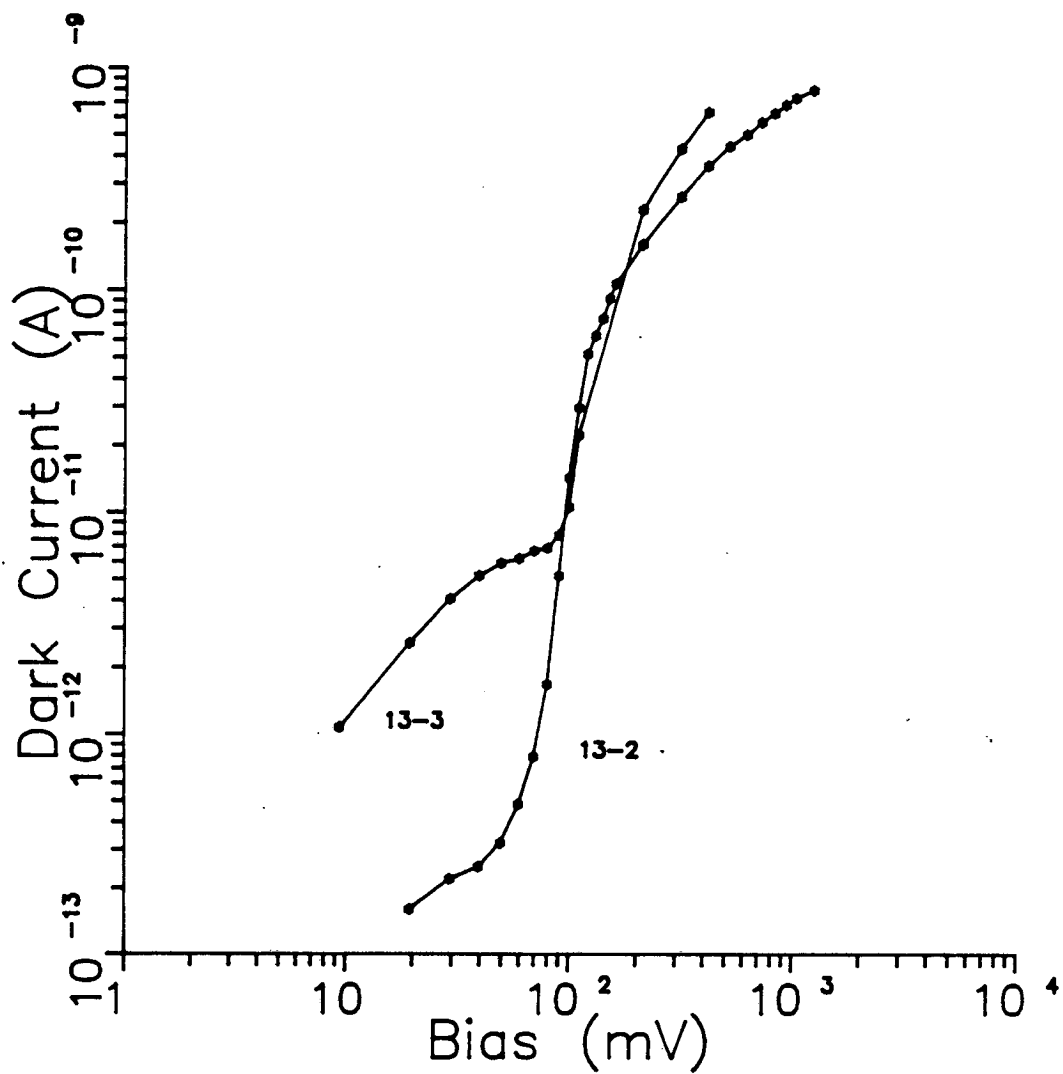


Fig. 14. Dark current as a function of bias for detectors 13-2 and 13-3 at 2.3 K under reverse bias.

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## 3.2. Ion Implanted BIB Detectors

- **Concept:**
  - In case pure and structurally perfect epitaxial layers are hard to produce, we can resort to implantation of dopants into an ultra-pure crystal.
- **Low energy B<sup>+</sup>-implantation tests:**
  - three B<sup>+</sup> energies: 150 keV, 95 keV, 50 keV form a 0.4  $\mu\text{m}$  thick layer with  $N_A = 3.5 \times 10^{16} \text{ cm}^{-3}$ .
  - annealing at 400°C for one hour in argon.
  - extended wavelength response.
  - responsivity = 0.5 A/W, dark current  $< 10^{-14} \text{ A}$ , at bias = 100 mV and  $T = 2.0 \text{ K}$ . NEP  $\approx 4 \times 10^{-16} \text{ W} / \sqrt{\text{Hz}}$ .

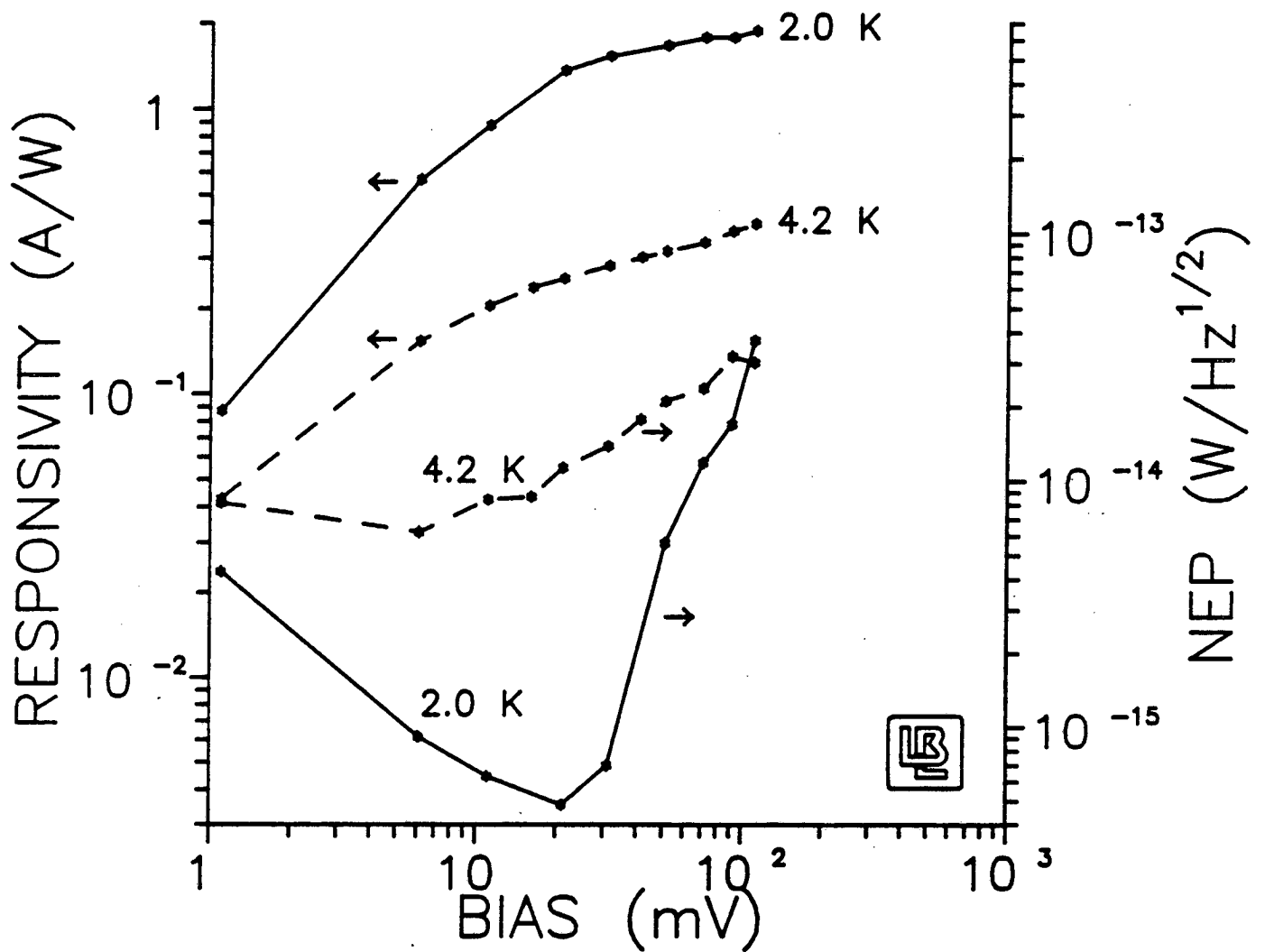


Fig. 15. Responsivity of a Ge BIB detector, low energy B<sup>+</sup>-implant type. Active layer depth = 0.6  $\mu\text{m}$ ,  $[B] = 1 \times 10^{16} \text{ cm}^{-3}$ ,  $\lambda_{\text{filter}} = 98.9 \mu\text{m}$ ,  $f_{\text{chopper}} = 23 \text{ Hz}$

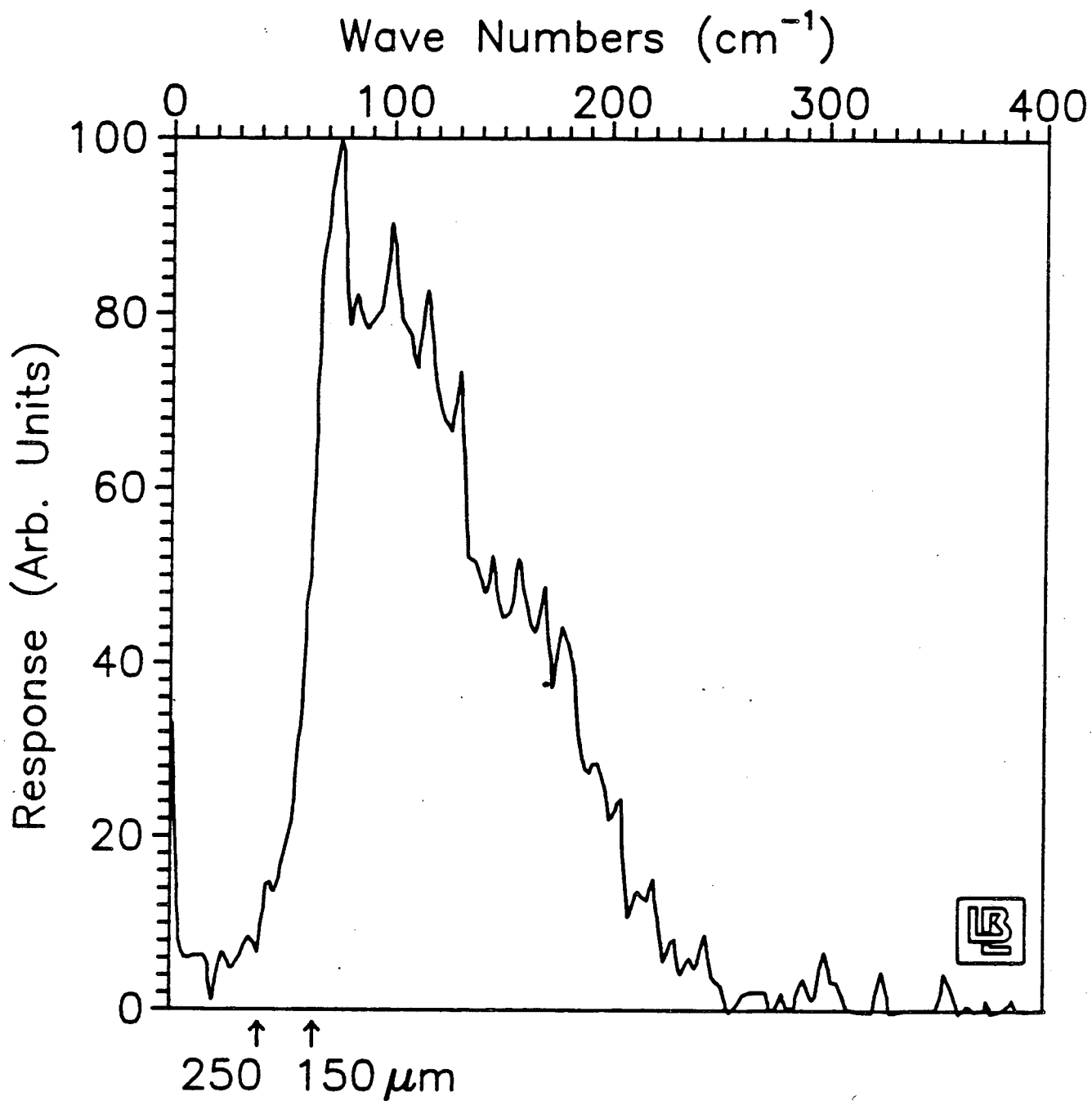


Fig. 16. Spectral response of Ge BIB detector, low energy B<sup>+</sup>-implantation type.

- **High energy B<sup>+</sup>-implantation tests:**
  - **14 implant energies up to 4 MeV doubly charged boron ions lead to a 5  $\mu\text{m}$  thick layer with  $N_A = 1 \times 10^{16} \text{ cm}^{-3}$ .**
  - **Variable temperature Hall effect and resistivity measurements indicate full activation of shallow acceptor dopant B. No deep levels are detectable after annealing. Below 15 K, hopping conduction becomes dominant.**
  - **Infrared transmission measurements and device tests are in progress.**

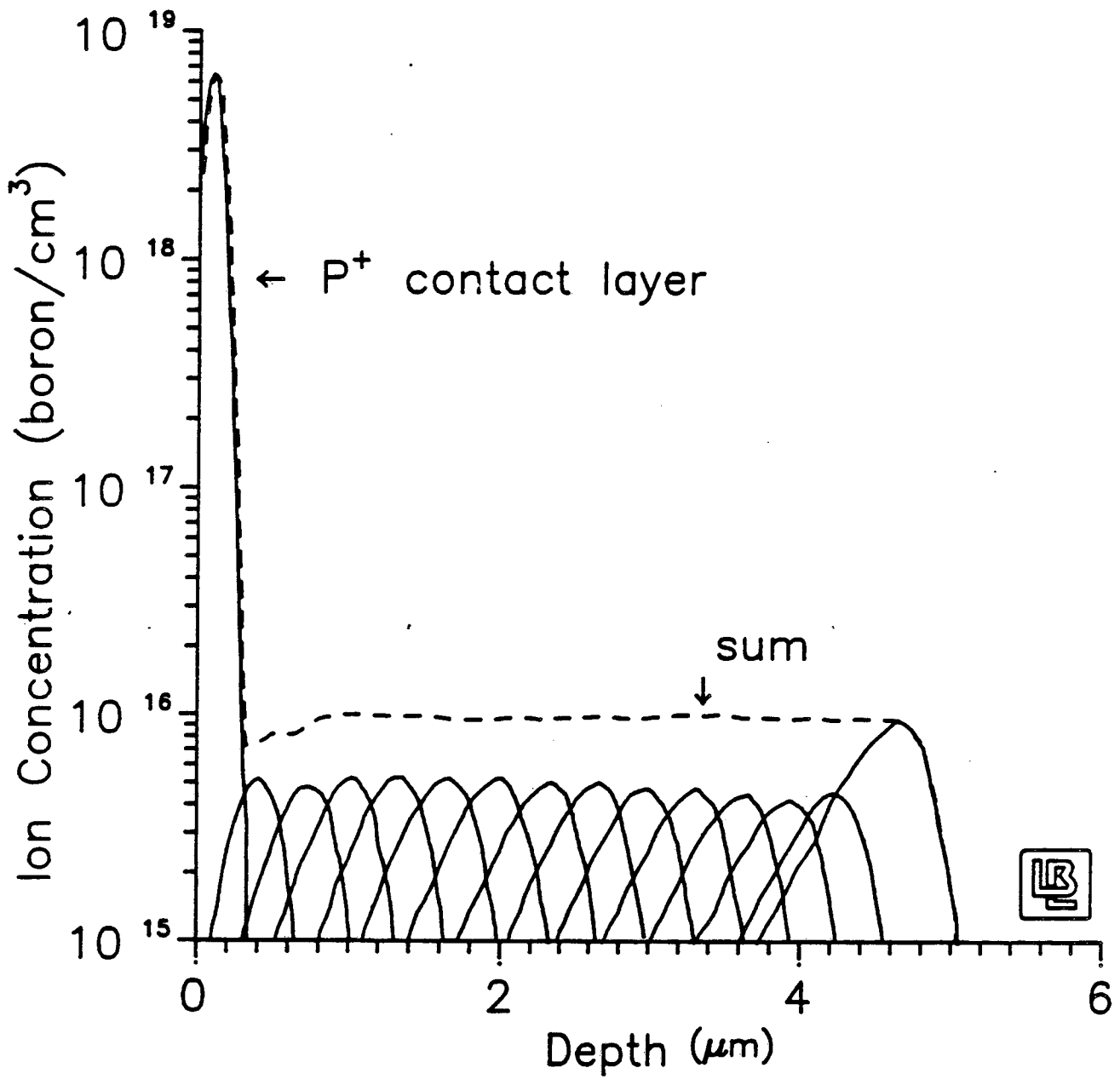


Fig. 17. Ge BIB, high energy ion implant profile



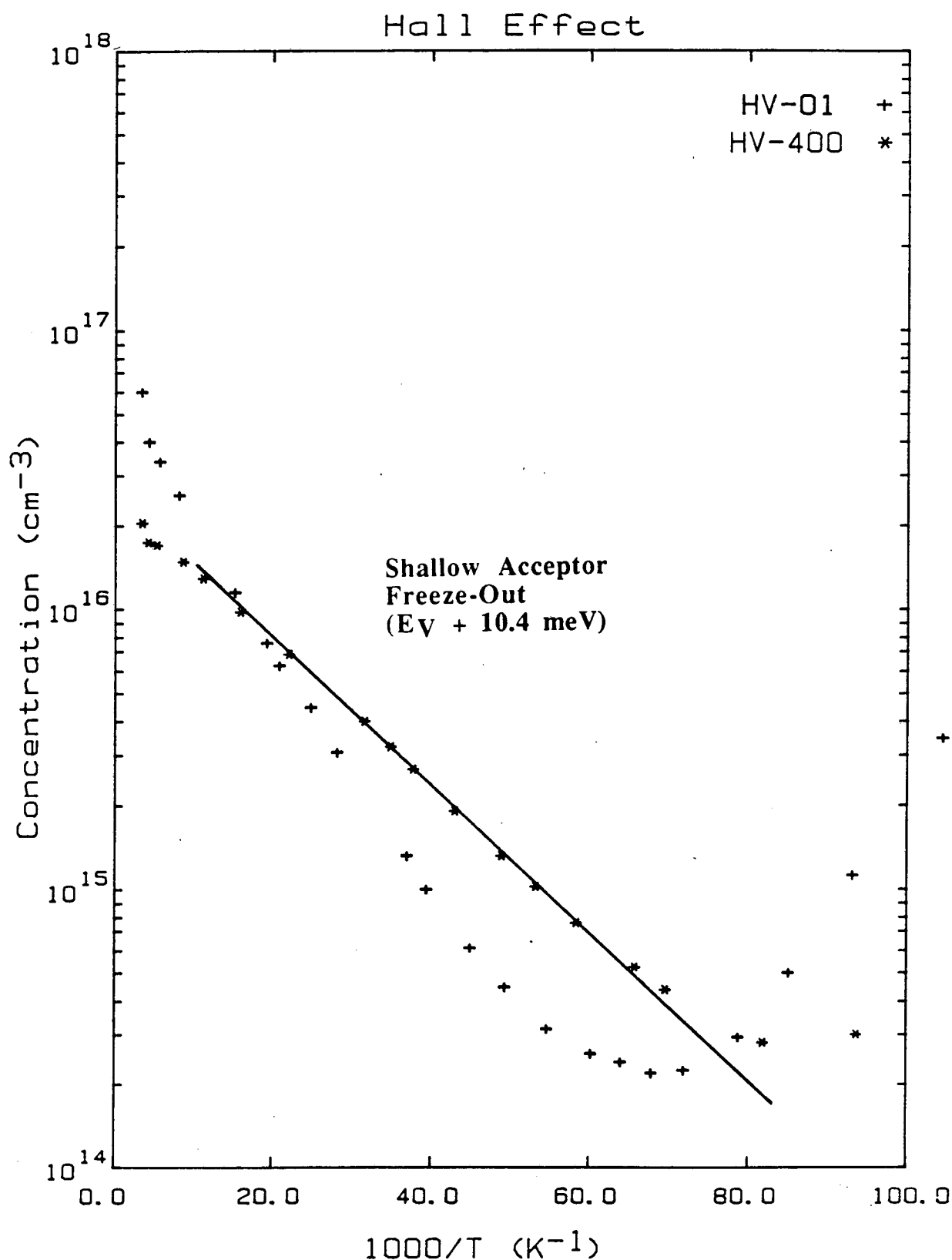


Fig. 18. Free carrier freeze-out of high energy  $B^+$ -implanted layer. Before annealing (+), the slope of the freeze-out curve is steeper than after annealing (\*). The latter slope corresponds to  $\sim 10.4 \text{ meV}$ , the binding energy of shallow boron acceptors. Below 15 K, hopping conduction becomes dominant.

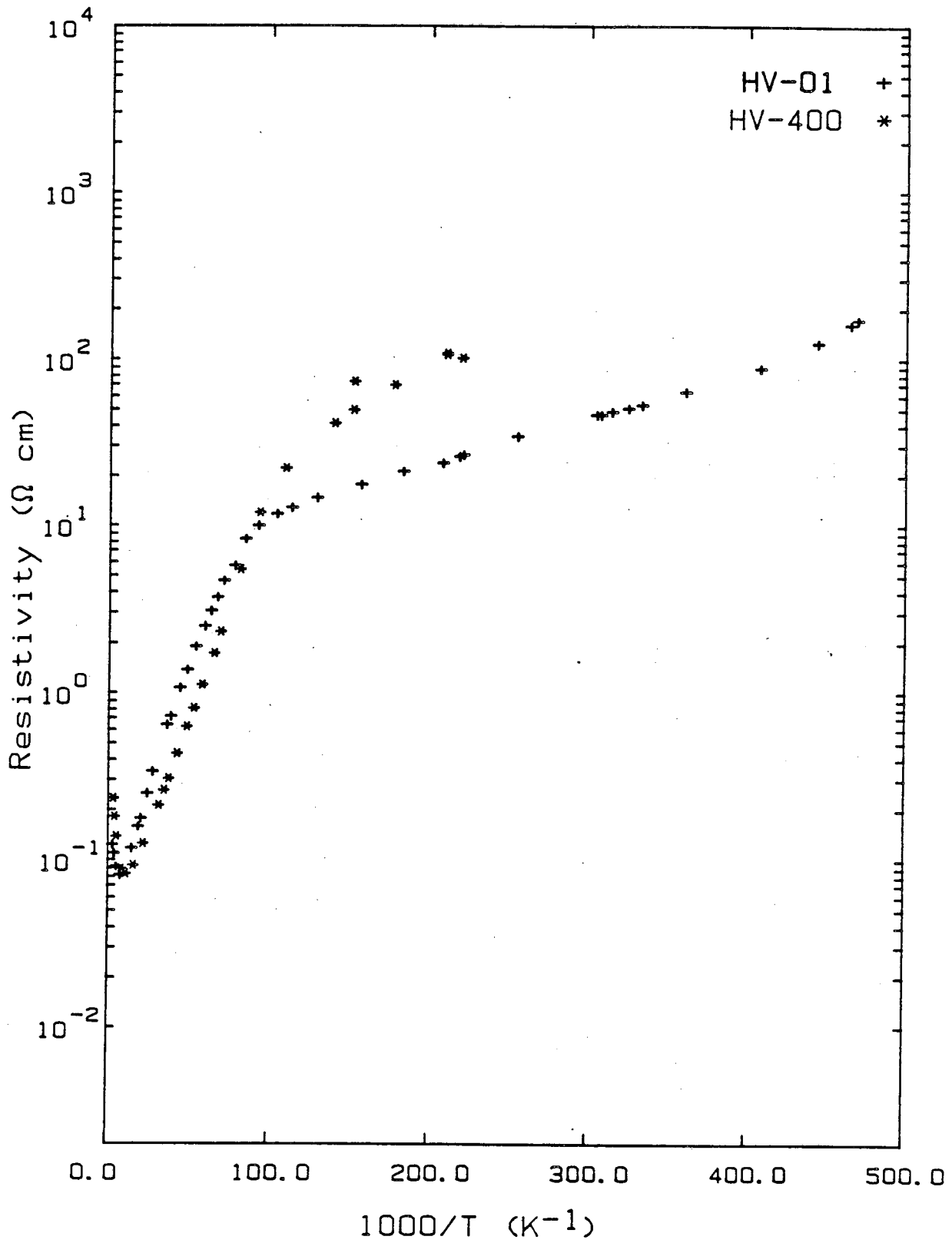


Fig. 19. Resistivity as a function of inverse temperature of the high energy  $B^+$ -implant layer before (+) and after (\*) annealing. Hopping conduction becomes dominant below 15 K.

## **4. CONCLUSIONS**

- **A LPVPE technique for the low temperature growth of epitaxial Ge layers has been developed.**
- **Hall effect and resistivity measurements indicate that the epi layers are lightly p-type due to residual copper contamination.**
- **First generation Ge BIB detectors made with moderately doped substrates ( $5 \times 10^{15} \text{ cm}^{-3}$ ) exhibit effective blocking of the hopping current.**
- **First generation Ge BIB detectors exhibit responsivities around 1 A/W.**
- **Second generation devices using low pressure VPE are being processed.**
- **Ion implanted active layers are tested.**
- **It is currently not known what temperatures will be required to reduce the dark current down to levels which are acceptable for SIRTf applications.**

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