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pH-dependent control of feldspar dissolution rate by altered surface layers



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ABSTRACT

Relevant modeling of mass and energy fluxes involved in pedogenesis, sequestration of atmospheric CO₂ or geochemical cycling of elements partly relies on kinetic rate laws of mineral dissolution obtained in the laboratory. Deriving an accurate and unified description of mineral dissolution has therefore become a prerequisite of crucial importance. However, the impact of amorphous silica-rich surface layers on the dissolution kinetics of silicate minerals remains poorly understood, and ignored in most reactive transport codes. In the present study, the dissolution of oriented cleavage surfaces and powders of labradorite feldspar was investigated as a function of pH and time at 80 °C in batch reactors. Electron microscopy observations confirmed the formation of silica-rich surface layers on all samples. At pH = 1.5, the dissolution rate of labradorite remained constant over time. In contrast, at pH = 3, both the dissolution rates at the external layer/solution interface and the internal layer/mineral interface dramatically decreased over time. The dissolution rate at the external interface was hardly measurable after 4 weeks of reaction, and decreased by an order of magnitude at the internal interface. In another set of experiments conducted in aqueous silica-rich solutions, the stabilization of silica-rich surface layers controlled the dissolution rate of labradorite at pH = 3. The reduction of labradorite dissolution rate may result from a gradual modification of the textural properties of the amorphous surface layer at the fluid/mineral interface. The passivation of the main cleavage of labradorite feldspar was consistent with that observed on powders. Overall, our results demonstrate that the nature of the fluid/mineral interface to be considered in the rate limiting step of the process, as well as the properties of the interfacial layer (i.e. its chemical composition, structure and texture) to be taken into account for an accurate determination of the dissolution kinetics may depend on several parameters, such as pH or time. The dramatic impact of the stabilization of surface layers with increasing pH implies that the formation and the role of surface layers on dissolving feldspar minerals should be accounted for in the future.

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1. Introduction

The critical zone in the Earth's near surface sustains several processes and ecosystem services such as soil formation (Godderis et al., 2010), CO₂ uptake due to silicate weathering (Beaulieu et al., 2012) and the biogeochemical cycling of elements (Olsson et al., 1996; White et al., 2012). Understanding these processes requires to quantify elemental fluxes through the critical zone and the contribution of chemical weathering of minerals. Current approaches to estimate dissolution rates under field conditions often rely on kinetic rate laws determined under laboratory conditions. Despite extensive investigations over the last decades, rates measured in laboratory setups can still differ by up to 6 orders of magnitude compared to those observed in the field (e.g.

Zhu, 2005). The origin of this “laboratory-field discrepancy” still remains unclear (White and Brantley, 2003).

Several factors may explain this discrepancy, which can be sorted into extrinsic and intrinsic factors. Extrinsic factors involve the surrounding of the mineral, encompassing the reacting fluid composition and environmental parameters in general. Intrinsic factors are related to the mineral itself, its chemical composition and structure, and to the presence of surface layers or other interfacial properties. An increasing number of laboratory studies and field observations regarding the role of intrinsic factors challenges the classical concepts relying on the homogeneity and immutability of the reacting solid phase. For instance, mineral reactivity may exhibit spatial heterogeneities when interacting with a reactive fluid (Fischer et al., 2012; Fischer et al., 2014), and the nature of the rate-controlling phase itself may evolve or even change during the reaction (Daval et al., 2011). This latter result strengthens the conclusions drawn from field studies which suggest that mineral

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reactivity may evolve with time (Maher et al., 2004; Fantle and DePaolo, 2006; Porder et al., 2007). In particular, fluid–mineral interactions may be controlled by interfacial features, such as amorphous Si-rich surface layers (ASSL) (Wang and Giammar, 2013). ASSLs have been identified in a wide variety of natural environments and mineral substrates (Kawano and Tomita, 2001; Zhu et al., 2006; Hellmann et al., 2012), indicating their ubiquity. ASSLs were brought to the forefront of mineral reactivity by Hellmann et al. (2003), and identified as a first-order rate-controlling parameter in laboratory experiments where silicate dissolution rates dramatically drop in solutions close to amorphous silica saturation (Daux et al., 1997; Daval et al., 2011; Daval et al., 2013; Johnson et al., 2014; Maher et al., 2016).

Despite recent efforts to decipher the impact of interfacial features on dissolution rates, the dissolution mechanisms of feldspar minerals still lack consensual agreement. Indeed, early studies suggested that a diffusive control of the dissolution rate by a Si-rich surface layer located at the fluid–mineral boundary caused the parabolic cation release observed during feldspar dissolution (Correns and von Engelhardt, 1938; Wollast, 1967; Helgeson, 1971). However, the scattering of data obtained on analogous experimental setups that could not be explained solely by diffusive process (Lagache, 1976), or experimental artefacts arising from ultrafine particles within the starting material (Holdren and Berner, 1979) contradicted the surface layer hypothesis. The parabolic cation release was then explained by the preferential dissolution of higher reactivity sites, despite subsequent improvements designed to overcome the weaknesses of original protocols (Knauss and Wolery, 1986; Carroll and Knauss, 2005). Recent studies have led to the consensus that surface layers are formed at the interface between the reacted feldspar and the reacting fluid (Hellmann et al., 1990; Gout et al., 1997; Hellmann et al., 2003; Hellmann et al., 2004; Hellmann et al., 2012). Although several studies have focused on the mechanism of surface layer formation, knowledge of their impact on dissolution rates remains incomplete.

This study aims to evaluate how surface layers formed on labradorite affect the dissolution rates at 80 °C. Reacting materials consisted either of labradorite powder or of oriented surfaces of labradorite single crystals, the latter being used to evaluate whether the crystallographic orientation influences the dissolution rate and the rate-controlling mechanisms of labradorite dissolution. Aqueous fluids of hydrochloric acid (HCl) solutions of pH = 1.5, 2, 2.5 and 3, which corresponds to the highest pH value for which experiments are not impacted by secondary phases, were used to collect pH-resolved data. For some modalities, reacting fluids were saturated with respect to amorphous silica to constrain the impact of surface layers on the dissolution rates. Overall, our results emphasize that the formation of surface layers can affect the dissolution kinetics of labradorite. The pH of the reacting fluid controls the textural properties of the layers, with increasing passivation effect at higher pH.

2. Materials and methods

2.1. Sample preparation

Centimeter-sized labradorite single crystals from Madagascar were purchased from Mawingu Gems. They consist of translucent greyish cm-sized euhedral to anhedral crystals containing Fe-rich inclusions. The chemical composition of the samples ($\text{Na}_{0.45}\text{K}_{0.02}\text{Ca}_{0.52}\text{Al}_{1.49}\text{Si}_{2.49}\text{O}_8$) was determined by electron microprobe.

A first part of the samples was used to prepare oriented surfaces. Labradorite samples were oriented along (001) and (010) preferential cleavages, and cut following selected crystallographic orientations. Oriented samples were then embedded in epoxy resin and polished through a multi-step abrasive sequence with an ultimate polishing step in a colloidal silica suspension. The crystallographic orientations were verified over the whole surface using electron backscatter diffraction (EBSD) on a Tescan Vega 2 scanning electron microscope (SEM).

The initial roughness of labradorite surfaces was estimated using a Digital Instruments atomic force microscope (AFM), and a Zygo New View 7300 vertical scanning interferometer (VSI). Roughness parameters given in the present work refer to the average roughness, calculated over each entire measured area:

$$R_a = \frac{1}{n} \sum_{i=1}^n |Z_i - \bar{Z}| \quad (1)$$

and to the root mean squared roughness:

$$R_q = \sqrt{\frac{1}{n} \sum_{i=1}^n (Z_i - \bar{Z})^2}, \quad (2)$$

where i stands for the index referring to each of the n data points of the considered array, Z_i to its height and \bar{Z} to the mean height calculated over the considered array. Measurements yield typical values of $1 \text{ nm} \leq R_a \leq 18 \text{ nm}$, $1 \text{ nm} \leq R_q \leq 20 \text{ nm}$, on 5×5 to $80 \times 80 \mu\text{m}^2$ AFM images and $3 \text{ nm} \leq R_a \leq 21 \text{ nm}$, $4 \text{ nm} \leq R_q \leq 36 \text{ nm}$, on $500 \times 500 \mu\text{m}^2$ VSI images. The geometric surface area of labradorite surfaces was determined by image analysis with the ImageJ software (Abramoff et al., 2004; Schneider et al., 2012) using binocular photographs of each sample. The sample surfaces were subsequently cleaned with ethanol and air dried. Several portions of the surface were protected with $\sim 1 \text{ mm}^2$ RTV glue spots, referred to as “glue masks” hereafter (see Section 2.5). The chemical inertness of the masks with respect to the mineral surface as well as their stability were checked prior to the start of the experiments. Si release generated by a typical amount of 20 glue masks on an epoxy stud immersed for several days in 120 mL of HCl at 80 °C (in situ pH = 1.5) was negligible.

Labradorite samples were also prepared as powders. Labradorite single crystals were washed with water, dried, crushed with a hydraulic press, sieved to recover the 160–315 μm fraction and sonicated in ethanol to get rid of fine particles. The powder was then dried overnight at 35 °C. The specific surface area measured with 7-points Kr BET was $0.051 \text{ m}^2 \text{ g}^{-1}$.

2.2. Experimental protocol

2.2.1. Experimental setup

Labradorite samples, either consisting of weighed amounts of powder or polished monoliths, were introduced into 120 mL PTFE Savillex® reactors continuously stirred with magnetic bars placed over PTFE tri-pods. The reactors were filled with 100 mL of reacting solutions and incubated at 80 °C.

2.2.2. Experiments in aqueous Si-rich solutions

Fluid saturation with respect to any solid material is defined as:

$$SI = \log_{10} \left(\frac{Q(a_{\text{products}})}{K(T)} \right) \quad (3)$$

where $Q(a_{\text{products}})$ stands for the reaction quotient and $K(T)$ for the temperature (T)-dependent equilibrium constant of the dissolution reaction. The experiments were conducted in both $\text{SiO}_2(\text{aq})$ -rich and $\text{SiO}_2(\text{aq})$ -low solutions. Reacting solutions were prepared from ultra-pure water ($18.2 \text{ M}\Omega \text{ cm}^{-1}$) and high-grade HCl (37%, ACS reagent) to adjust the pH. For $\text{SiO}_2(\text{aq})$ -rich solutions, the fluid was saturated with respect to amorphous silica at 80 °C ($SI = 0$) by dissolving 1.48 g of sodium metasilicate, nonahydrate (Sigma Aldrich®, >98%) in 1 kg of solution. Total dissolution of sodium metasilicate was ensured by vigorous stirring, followed by a sonication step. pH was subsequently adjusted with HCl and verified two days after the preparation of the solution.

2.2.3. Face-specific experiments

A series of experiments at a mid-range pH (pH = 2.5) was designed to tackle the influence of anisotropy on dissolution rates. It consisted of face-specific experiments, where (i) only the desired face was exposed to the reactive fluid, and (ii) each crystal orientation was reacted separately. This procedure enabled to retrieve face-resolved fluid data and to determine the face specific dissolution rate at the boundary between the surface layer and the pristine mineral, defined hereafter as the “internal interface”. The opposite boundary of the surface layer, located at the interface with the bulk solution is referred to as the “external interface”. The dissolution rate at the external interface was determined on each orientation with the standard VSI protocol described in Sections 2.5 and 2.6. Two faces identified as the main cleavages on euhedral monoliths were selected, corresponding to (001) and (010) planes. In addition, two other orientations that were not expressed at the macro scale, namely (10 $\bar{1}$) and (110) orientations were further studied. Additionally, batch powder experiments at pH = 2 and 2.5 were conducted to evaluate the representativeness of the selected faces.

2.2.4. Sample recovery and fluid analysis

The reactors were regularly sampled for pH measurements and chemical analyses. The fluid mass loss due to sampling and fluid evaporation was estimated by regularly weighing the reactors. The Ca/Si ratios were calculated from inductively coupled plasma atomic emission spectroscopy (ICP-AES) data acquired with a Thermo ICP 6000 Series apparatus as follows:

$$\text{Ca/Si} = \frac{\Delta\text{Ca}}{\Delta\text{Si}} \quad (4)$$

where ΔCa and ΔSi stand for the amount of Ca and Si released between two sample collections.

At the end of each experiment, the fluid was recovered and the solid samples were briefly rinsed with deionized MilliQ® water and absolute ethanol (Sigma Aldrich®, ACS reagent). The samples were then dried overnight at 35 °C. Glue masks were eventually pulled off with PTFE tweezers, and potential glue left-over was removed with a single-tip swab (Puritan®) impregnated with ethanol.

2.3. Thermodynamic calculations

The in situ pH and saturation indices for labradorite, amorphous silica and potential secondary phases at 80 °C were calculated using the Chess® software (Van der Lee and De Windt, 2002) and the Chess® tdb database (Lawrence Livermore National Laboratories EQ3/6 database, 8th version). Thermodynamic constants for labradorite dissolution were calculated from those for albite (Ab) and anorthite (An), supposing an ideal solid solution between albite and anorthite end-members (see Testemale et al. (2009) for details), and equal proportions of albite and anorthite.

Activity coefficients for aqueous species were calculated with Chess® using the truncated Davies equation. Initial pH was calculated based on the initial weighed amount of HCl introduced into solution. From this starting point onwards, pH was used as an adjustable parameter to achieve charge balance.

For experiments saturated with respect to amorphous silica, the high background level for Si and Na resulting from the added sodium metasilicate powder precluded the accurate determination of Si and Na released from the mineral, which complicates the calculation of the saturation indices. For these experiments, the aqueous concentrations of Si and Na were estimated from the weighed amount of ultrapure MilliQ® water and chemical reagents in the initial solutions to which were added the estimated quantity of Si and Na released by the mineral dissolution based on Ca (when sufficiently accurate) or on Al

concentrations, assuming the stoichiometric dissolution of labradorite:

$$n_{X\text{estimated}}(t) = n_{X\text{initial}} + \left(\frac{\eta_X}{\eta_Y}\right) * n_Y(t) \quad (5)$$

where n_X stands for the estimated released amount of Si or Na due to mineral dissolution at t , $n_Y(t)$ is the total amount of calcium or aluminum released into solution at t and η_i is the stoichiometric coefficient of element i in the bulk labradorite.

2.4. Sample observation and determination of surface layer thickness

Samples were either gold- or carbon-coated, and investigated with SEM and energy dispersive X-ray (EDX) spectroscopy. For samples exhibiting layer thickness exceeding 1 μm , transects of reacted samples were cut perpendicularly to the reacted surface with a diamond saw across the masked area when possible and embedded in epoxy resin. This cross section was then polished, carbon-coated and the layer thickness was measured with SEM. For samples with layers thinner than 1 μm , ultrathin electron transparent cross sections were milled by focused ion beam (FIB) through the reacted surface, which was previously re-coated with a thick carbon layer to prevent Pt and Ga ion beam damages to the sample (Lee et al., 2007). FIB thin sections were prepared following conventional procedures (Saldi et al., 2015), and analyzed by transmission electron microscopy (TEM) using a JEOL 2100F microscope operating at 200 kV, equipped with a field emission gun and an energy dispersive X-ray analysis system from JEOL.

2.5. Determination of surface retreat

Sample topography was measured by VSI on each polished mineral sample prior to reaction. A portion of the surface was then masked with a RTV glue spot to avoid fluid-mineral contact at this particular location. The masks were removed at the end of the experiments, and the sample topography was measured with VSI. The height difference between the reacted and the masked areas created by labradorite dissolution is referred to as surface retreat.

For samples which exhibited large surface retreats, the dissolution profiles were corrected from the starting topography using a three-step routine developed in Matlab® environment. Briefly, linear profiles were recorded across the masked area at the exact same location of the sample surface prior to and after reaction (Fig. 1A). Profiles were then superimposed on a common x scale by linear interpolation:

$$z(x) = z(x_1) + \frac{(z(x_2) - z(x_1))}{(x_2 - x_1)} * (x - x_1) \quad (6)$$

where z is the height at a given abscissa, x is the projected distance from the common origin of both profiles to any recorded pixel in the post-reaction profile, and x_1 and x_2 correspond to the projected distances along the pre-reaction profile of the closest recorded pixels preceding and following x respectively. The trend due to global tilt of the sample was then subtracted for both profiles by subtracting a linear fit performed on a common portion of surface area, i.e. belonging to the masked zone (see Fig. 1B), unaffected by the reaction. The obtained corrected profiles (Fig. 1C) were eventually subtracted one from another to obtain surface retreat along the x coordinate (Fig. 1D). Final retreat value corresponds to the difference between the mean height of the masked area and the mean height of the unmasked area, while uncertainty integrates height variation on both portions.

As shown on Fig. 2A and B, small surface retreats (<100 nm) were locally measured in the direct vicinity of the boundary between the masked and the unmasked portions (typically 50 μm each side). The mean surface retreat was estimated from the peak-to-peak distance measured between the reference and the retreated portions of the surface on the altitude distribution histogram (Fig. 2C). The uncertainty on

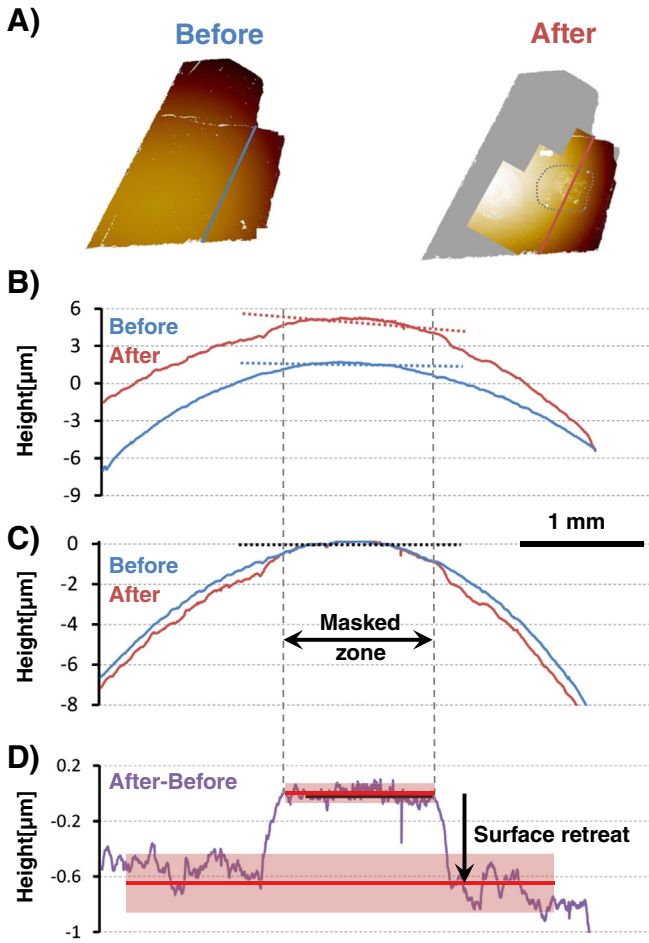


Fig. 1. Determination of large surface retreats (>0.1 μm). Surface topography was acquired by VSI for each sample prior to and after dissolution. Linear height profiles were measured across the masked area indicated by a grey dashed contour at the exact same location on both maps (A). Linear regression (dashed lines) for each profile was retrieved from a portion of the masked zone that remained preserved from dissolution (B). The parameters of these regressions were used for correcting their respective tilts (C), and corrected profiles were subtracted to estimate the surface retreat (D). The red rectangles indicate the ranges of topography variations due to surface roughness.

surface retreat was estimated from their full width at half maximum. As shown on Fig. 2D, relevance of the selected altitudes on the histogram were checked for being consistent with retreats from linear profiles on the same zones to avoid artefacts resulting from residual scratches (Fig. 2D).

2.6. Estimation of the dissolution rate

Absolute dissolution rate at the external interface was estimated from the surface retreat following Arvidson et al. (2004):

$$r = \frac{\Delta h}{\Delta t * V_m} \tag{7}$$

where *r* stands for the absolute dissolution rate in mol·m⁻²·s⁻¹, Δ*h* is the surface retreat, Δ*t* is the alteration duration and *V_m* is the molar volume of labradorite, in m³·mol⁻¹.

Rate at the internal interface was estimated on the basis of the release of elements from the mineral into the solution, and calculated as follows:

$$R(X) = \frac{\Delta X}{\Delta t * S * \eta_X} \tag{8}$$

where Δ*X* and Δ*t* stand for the amount of element *X* released to the solution by the dissolution process and the elapsed time between two consecutive aqueous samples respectively, *S* is the surface area of the mineral, and η_{*X*} is the stoichiometric coefficient of element *X* in the bulk labradorite. To avoid distortion due to the incorporation of element *X* into secondary products, Al or Ca were preferentially and independently used as elemental tracers in most experiments to estimate the dissolution rates.

The surface term *S* in Eq. (8) can either refer to the initial geometric surface area for macroscopic crystal samples or to the initial BET surface area for powders. This difference in the surface estimation precludes a direct comparison of the absolute rates between experiments run on powders or cleaved surfaces.

2.7. Propagation of uncertainties

Error bars for data based on measurements of cation release into solution were propagated from ICP-AES measurements. Elemental ratio uncertainties were estimated as:

$$\sigma_{X/Y} = \frac{\Delta X}{\Delta Y} * \sqrt{\frac{\sigma_{X(t)}^2 + \sigma_{X(t-1)}^2}{\Delta X^2} + \frac{\sigma_{Y(t)}^2 + \sigma_{Y(t-1)}^2}{\Delta Y^2}} \tag{9}$$

where Δ*X* and Δ*Y* stand for the amount of calcium and silicon released between two sample collections, while σ_{*X(t)*} and σ_{*Y(t)*} account for analytical errors on *X* concentration (*X*) and *Y* concentration (*Y*) at *t* respectively. Uncertainties on batch dissolution rates were estimated as:

$$\sigma_{R(t)} = \frac{1}{S * \Delta t * \nu_X} * \sqrt{\sigma_{X(t)}^2 + \sigma_{X(t-1)}^2} \tag{10}$$

where σ_{*X(t)*} stands for the error on element *X* concentrations and ν_{*Ca*} for the stoichiometric coefficient of calcium within labradorite.

Uncertainties on rate (*R*) ratios were estimated as:

$$\sigma_{R_1/R_2} = \frac{R_1}{R_2} * \sqrt{\frac{\sigma_{R_1}^2}{R_1^2} + \frac{\sigma_{R_2}^2}{R_2^2}} \tag{11}$$

where σ_{*R*} stands for the error estimated on batch rates (see above).

3. Results

A summary of the different modalities tested in the present work can be found in Table 1. Information retrieved from fluid analyses can be found in Table A.1, while data from solid sample observations (VSI, SEM or TEM) are gathered in Table 2.

3.1. Formation of surface layers

Surface layers at the fluid-feldspar interface were formed in all samples. At pH = 1.5, labradorite dissolution is characterized by an incongruent release of cations into the solution (Fig. 3A). The relative lack of Si released in the fluid phase depicted by Ca/Si values above stoichiometric ratio (Fig. 3A) are due to the formation of a distinct Si-rich layer (Fig. 3B), whose thickness ranges from 7.1 to 26.3 μm (Table 2). SEM-EDX and electron diffraction by TEM analyses indicated that the surface layer mainly consists of pure amorphous silica (Fig. 3B). The formation of about 50 μm-thick ASSLS was also observed with SEM on labradorite crystals reacted at pH = 1.5 in silica-rich solutions (Fig. 3C). As shown in Fig. 4, TEM investigations of FIB thin sections revealed thin layers (thickness < 100 nm, see Table 2) at the surface of labradorite crystals reacted at pH = 2.5 and 3. In these experiments, no incongruent cation release was detected as the layers presumably got thinner. At these higher pH values, the composition of surface layers could hardly be

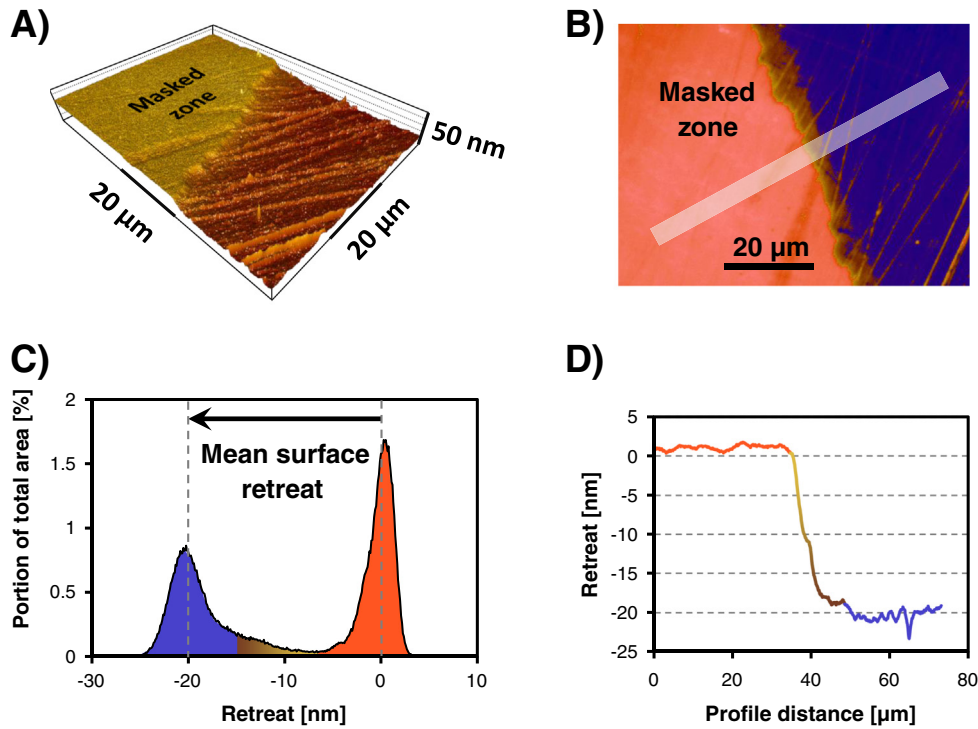


Fig. 2. Determination of small surface retreats (<100 nm). VSI images were acquired in the direct vicinity of the boundary between the masked and the unmasked portions of the surface (A). The mean surface retreats were measured based on the most representative heights of each portion (B) from the histogram plots (C). The profile (D) corresponds to the mean heights across the grey rectangle shown in B.

distinguished from the bulk mineral, aside from a modest depletion in alkali and alkaline earth cations.

3.2. pH-dependence of dissolution rates at the internal and external interfaces

Our results show that the temporal evolution of labradorite dissolution rate is pH-dependent. At pH = 1.5, Ca concentration follows a linear trend with time, which reflects constant mineral dissolution rate (Fig. 5). In contrast, cation release at pH = 3 was best fitted with a power law, underscoring that reaction rate decreases over time at the internal interface.

Like the internal interface, the reaction rate of the external interface was also pH-dependent. The surface retreats measured by VSI at the external interface evolved linearly at pH = 1.5, indicating a constant reaction rate (Fig. 6). In comparison, retreat values measured by VSI at pH = 3 were very low (about 500 nm after 28 days of reaction, i.e. two orders

of magnitude smaller than at pH = 1.5 for the same duration) and suggested a reduction in the reaction rate.

3.3. Impact of the anisotropy (experiments run at pH = 2.5)

Overall, the crystallographic orientation did not measurably affect the reaction rates at the external interface when the reaction was run in aqueous silica-low solutions (Fig. 7). The dissolution rate was reduced only in the (001) crystallographic orientation, similar to the powder experiment (Fig. 8A). In contrast, we found $R(t)/R_i > 1$ in experiments with the orientations (110), (10 $\bar{1}$) and (010), which underscores an apparent modest increase of the reaction rate. In the experiments conducted in silica-rich solutions, the (001), (010) and (110) orientations exhibited a decrease of dissolution rates similar to that observed for the powdered labradorite experiment in silica-low fluid (Fig. 8B). However, the dissolution rate of the (10 $\bar{1}$) orientation remained approximately constant.

Table 1

Summary of experiments conducted with oriented single crystal surfaces. For some experiments, the reacting fluid was saturated with respect to amorphous silica (SiO₂ (am), see text). Numbers within brackets correspond to the Miller indices of the polished face. Two additional experiments were conducted on labradorite powders in silica-low fluids at pH = 2.5 (80-2.5-0-P) and 2 (80-2-0-P).

	pH = 1.5 ; SiO ₂ (am)	pH = 1.5	pH = 2.5 ; SiO ₂ (am)	pH = 2.5	pH = 3 ; SiO ₂ (am)	pH = 3
(001)	80-1.5-SiO ₂ -1 80-1.5-SiO ₂ -2	80-1.5-0-1 80-1.5-0-2 80-1.5-0-3 80-1.5-0-4	80-2.5-SiO ₂ -001 ^a	80-2.5-0-00 ^a 80-2.5-0-001-2 ^a		80-3-0-1 80-3-0-2
(010)		80-1.5-0-1 80-1.5-0-2 80-1.5-0-4	80-2.5-SiO ₂ -010 ^a	80-2.5-0-010 ^a	80-3-SiO ₂ -1	80-3-0-1 80-3-0-2
(10 $\bar{1}$)			80-2.5-SiO ₂ -10 $\bar{1}$ ^a	80-2.5-0-10 $\bar{1}$ ^a		
(110)			80-2.5-SiO ₂ -110 ^a	80-2.5-0-110 ^a		

^a Face-specific experiments, see text in Section 2.2.3 for further details.

Table 2

VSI, SEM and TEM measurements on dissolved labradorite samples. Global surface retreats and associated uncertainties (σ) are reported in the table. Values for layer thickness and associated uncertainties (σ) were estimated from SEM and TEM measurements.

Sample	Orientation [Miller indices]	pH	Time [Days]	Retreat [μm]	σ (Retreat) [μm]	Layer thickness [μm]	σ (Layer thickness) [μm]	Rate (ext) [$\text{mol}/\text{m}^2/\text{s}$]	σ (Rate) [$\text{mol}/\text{m}^2/\text{s}$]
<i>[SiO₂](t = 0) = 0 M</i>									
80-1.5-0-1-001	001	1.5	28.0	47.9	10.0	26.3	6.1	2.0E-07	4.2E-08
80-1.5-0-1-010	010	1.5	28.0	31.1	6.0	18.5	13.8	1.3E-07	2.5E-08
80-1.5-0-2-001	001	1.5	4.0	9.5	3.0	7.1	3.8	2.8E-07	8.9E-08
80-1.5-0-2-010	010	1.5	4.0	5.0	2.5	14.3	2.5	1.5E-07	7.4E-08
80-1.5-0-3-001	001	1.5	1.0	3.4	0.3	ND	ND	4.0E-07	3.6E-08
80-1.5-0-4-001	001	1.5	14.0	20.0	5.0	ND	ND	1.7E-07	4.2E-08
80-1.5-0-4-010	010	1.5	14.0	11.9	4.5	ND	ND	1.0E-07	3.8E-08
80-2.5-0-001	001	2.5	5.0	0.583	0.091	ND	ND	1.4E-08	2.1E-09
80-2.5-0-001-2	001	2.5	12.0	ND	ND	0.014	0.002	ND	ND
80-2.5-0-010	010	2.5	6.0	0.572	0.058	ND	ND	1.1E-08	1.1E-09
80-2.5-0-110	110	2.5	6.0	1.026	0.212	ND	ND	2.0E-08	4.2E-09
80-2.5-0-10 $\bar{1}$	10 $\bar{1}$	2.5	5.0	0.820	0.181	ND	ND	1.9E-08	4.3E-09
80-3-0-1-001	001	3	28.0	0.594	0.078	0.027	0.001	2.5E-09	3.3E-10
80-3-0-1-010	010	3	28.0	0.466	0.100	ND	ND	2.0E-09	4.2E-10
80-3-0-2-001	001	3	64.0	0.366	0.100	0.026	0.005	6.8E-10	1.9E-10
80-3-0-2-010	010	3	64.0	0.213	0.040	ND	ND	4.0E-10	7.4E-11
<i>[SiO₂](t = 0) = 5.2E-3 M</i>									
80-1.5-SiO ₂ -2-001	001	1.5	60.0	0.638	0.150	47.740	11.290	1.3E-09	3.0E-10
80-2.5-SiO ₂ -001	001	2.5	20.7	0.022	0.008	ND	ND	1.3E-10	4.3E-11
80-2.5-SiO ₂ -001-2	001	2.5	20.0	ND	ND	0.010	0.003	ND	ND
80-2.5-SiO ₂ -010	010	2.5	18.0	0.021	0.012	ND	ND	1.4E-10	7.9E-11
80-2.5-SiO ₂ -110	110	2.5	18.0	0.014	0.005	ND	ND	9.4E-11	3.6E-11
80-2.5-SiO ₂ -10 $\bar{1}$	10 $\bar{1}$	2.5	18.0	0.005	0.005	ND	ND	3.3E-11	3.3E-11
80-3-SiO ₂ -1-010	010	3	28.0	0.010	0.004	0.057	0.006	4.2E-11	1.7E-11

4. Discussion

4.1. Origin of the reduction of dissolution rates

The possible mechanism explaining the significant pH-dependent reduction of labradorite dissolution rates observed at the *internal* and *external* interfaces are discussed below. We considered potential experimental artefacts, as well as the contribution of extrinsic (relative to the reaction conditions) and intrinsic factors (relative to the mineral itself). The dataset suggests that the temporal reduction of labradorite dissolution rate is caused by the formation of an amorphous layer with pH-dependent transport properties.

4.1.1. Experimental artefacts

Parabolic release of cations such as those obtained at pH = 3 (Fig. 5) may stem from the precipitation of secondary phases or the development of diffusion gradients within the reactor due to insufficient stirring (Verney-Carron et al., 2010).

The formation of secondary phases can lead to a gradual uptake of released cations with concomitant change of cation concentrations similar to those displayed in Fig. 5. As reported in Table A.1, however, all experiments initiated in pH < 3 solutions were undersaturated ($SI < 0$) with respect to the mineral phases implemented in the EQ3/6 database, except for high pressure and/or temperature polymorphs of silica that were unlikely to precipitate under our experimental conditions. Experiments conducted at pH = 3 reached saturation with respect to a series of secondary Al-bearing phases. However, no significant deviation from the theoretical Ca/Al and Ca/Si ratios for labradorite was detected (experiments 80-3-0-1, 80-3-0-2 and 80-3-SiO₂-1, Fig. 9). In addition, SEM observations confirmed the absence of secondary phases in all experiments. Despite being thermodynamically favored, precipitation of gibbsite, boehmite, kaolinite, pyrophyllite and beidellite would probably require higher temperature or pressure to overcome the activation barrier of nucleation over the reaction time of this study (see e.g. Eberl and Hower, 1976; Adschiri et al., 1992; Grauby et al., 1993; Bird et al., 1994).

The development of compositional gradients within the reactor can also result in parabolic profiles when the reaction is transport-limited.

In the present study, the most rapid dissolution rates among the tested modalities were observed for the experiments at pH = 1.5. These experiments yielded linear cation concentration profiles, indicating that the solution was sufficiently stirred. Hence, other experiments conducted at higher pH (i.e. with slower dissolution rates) were not affected by transport-limitation.

Overall, experimental artefacts, such as the formation of secondary minerals or the development of compositional gradients in the reactor, were very unlikely to explain the reduction of the labradorite dissolution rates.

4.1.2. Extrinsic factors

The influence of temperature, fluid saturation and pH on mineral reaction rates is underscored in the following equation (Lasaga, 1984; Lasaga, 1995; Lasaga and Berner, 1998):

$$r(T, \text{pH}) = A * \exp\left(\frac{-E_a}{RT}\right) * 10^{-n * \text{pH}} * f(\Delta G_r) \quad (12)$$

where A is the Arrhenius pre-exponential factor, E_a is the apparent activation energy of the reaction, T is the absolute temperature, R is the gas constant, n is the reaction order with respect to proton and $f(\Delta G_r)$ accounts for the effect on the rate of the Gibbs free energy of the reaction. As temperature was kept at 80 °C for all experiments, it can be discarded as potential explanatory extrinsic factor. The effect of ΔG_r variations and pH drift during the dissolution were estimated for modalities presenting significant changes of their reaction conditions. The theoretical expected rate, accounting for ΔG_r and pH changes, was calculated based on (i) the initial experimental rate, (ii) E_a and n sourced from Palandri and Kharaka (2004) and (iii) the $f(\Delta G_r)$ function, which was taken from the relation determined on labradorite dissolution by Taylor et al. (2000). This relation showed that labradorite dissolution rate measured at 25 °C is ΔG_r -independent as long as $\Delta G_r < -42$ kJ/mol. As ΔG_r remained systematically below -42 kJ/mol for all experiments (Table A.1), it did not significantly contribute to the temporal decline of the dissolution rate. Detailed simulations of the evolution of cation concentration (Eq. (12)) and of the corresponding reaction rate for experiments

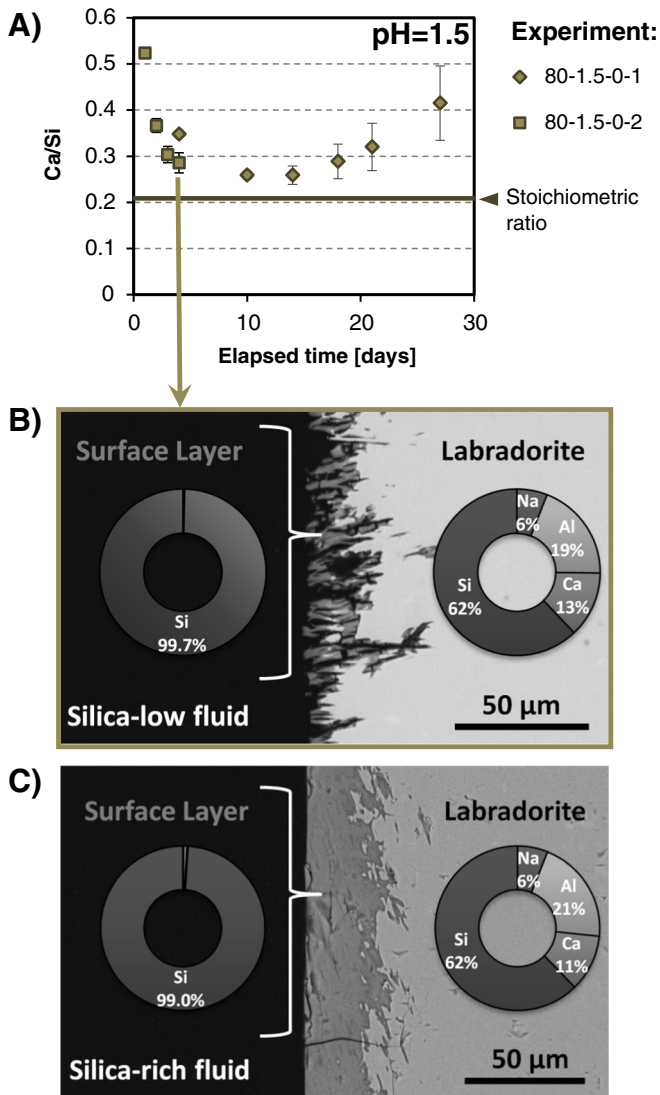


Fig. 3. Formation of ASSLs in experiments conducted in pH = 1.5 HCl solutions at 80 °C. The error bars indicate the analytical uncertainty of the $\Delta\text{Ca}/\Delta\text{Si}$ ratio (A) as calculated via error propagation. Bold horizontal line indicates Ca/Si stoichiometric ratio in Labradorite. SEM micrograph acquired on sample cross sections of labradorite samples reacted in aqueous silica-low solutions at pH = 1.5 for 4 days (B), or reacted in solutions saturated with respect to amorphous silica for 60 days (C).

with the highest pH variations confirmed that the impact of pH and ΔG_r on the reaction rate can be considered as negligible (e.g. Fig. 10).

Finally, the Al concentration may also control the dissolution rate of feldspar minerals (Oelkers et al., 1994), although this “Al-effect” has

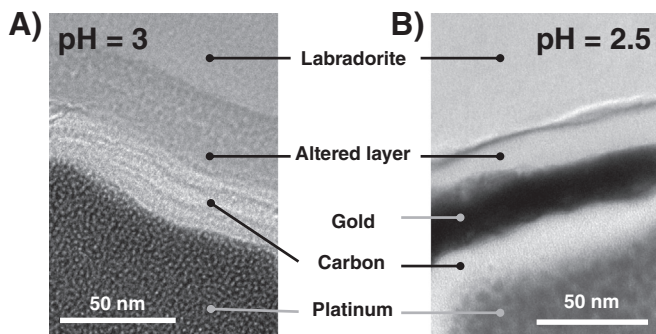


Fig. 4. TEM micrographs at the outer boundary of labradorite samples reacted at 80 °C in aqueous silica-low solutions at pH = 3 for 28 days and pH = 2.5 for 12 days.

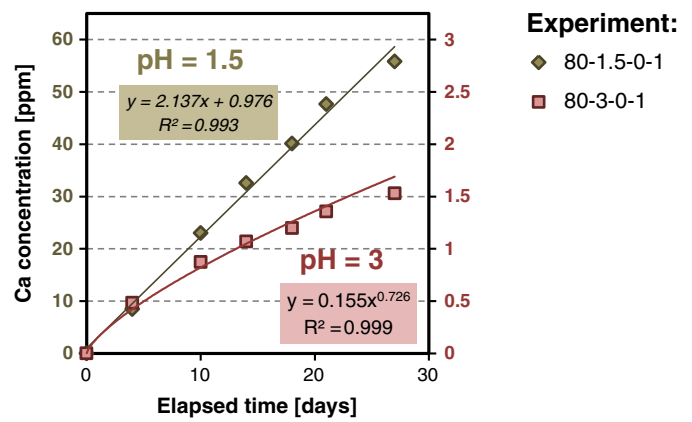


Fig. 5. Cation release from labradorite dissolution at 80 °C in aqueous silica-low solutions at pH = 1.5 (green diamonds) and pH = 3 (red squares). Coefficients of determination for each fit are displayed. ICP-AES errors of concentrations are about the size of symbols used. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

been questioned in several studies (e.g. Burch et al., 1993; Lasaga, 1995; Lüttge, 2006; Hellmann et al., 2010). In our case, Al was likely not a primary parameter explaining the reduction of labradorite dissolution rates, as dissolution rates remained constant in experiments with the greatest Al concentrations (pH = 1.5). In addition, we investigated the correlation between the reaction rate and the parameter $\frac{a_{\text{H}^+}^{3n}}{a_{\text{Al}^{3+}}}$ (Fig. 11), where n refers to a stoichiometric coefficient equal to the number of Al ions in the potential precursor sites involved in the rate limiting step of the dissolution reaction (Oelkers et al., 1994; Schott and Oelkers, 1995). The lack of correlation between $\frac{a_{\text{H}^+}^{3n}}{a_{\text{Al}^{3+}}}$ and the labradorite dissolution rate ($R^2 = 0.23$, Fig. 11), confirmed that the Al-effect was not a primary rate-controlling factor. To sum up, extrinsic parameters evaluated in this study did not appear to contribute significantly to the reduction of reaction rates observed at pH > 1.5.

4.1.3. Intrinsic factors

The origin of parabolic cation release has been previously debated and attributed (i) to the formation of surface layers, supposed to affect the transport properties of aqueous species at the fluid–mineral interface (Correns and von Engelhardt, 1938; Wollast, 1967; Helgeson, 1971; Luce et al., 1972; Busenberg and Clemency, 1976), and (ii) to the preferential dissolution of high energy sites or ultrafine particles during the early stages of the dissolution as a result of intrinsic structural defects or sample preparation (Lagache et al., 1961; Lagache, 1965; Holdren and Berner, 1979; Schott et al., 1981).

A transitional effect due to the dissolution of ultrafine particles is unlikely in our case as we used a well-standardized elimination protocol that was previously proven efficient (Schott et al., 1981). In addition, reduction of labradorite dissolution rate was also observed on the polished surfaces, which are unlikely to be covered with ultrafine particles.

The contribution of high surface energy sites was evaluated based on the extent of reaction from which significant reduction of dissolution rate started. In our case, this value depended on the reaction conditions (Fig. 12). In contrast, reaction rates controlled by the consumption of high surface energy sites are expected to be lower after comparable extents of reaction, whatever the pH conditions.

As none of the aforementioned processes could reasonably explain the reduction of labradorite dissolution rate, surface layers likely represent the only explanatory factor. Supporting this argument are the observations that (i) thin surface layers are formed at pH > 1.5, with a characteristic thickness of surface passivation (Daval et al., 2011; Daval et al., 2013; Sissmann et al., 2013; Saldi et al., 2015; Maher et al.,

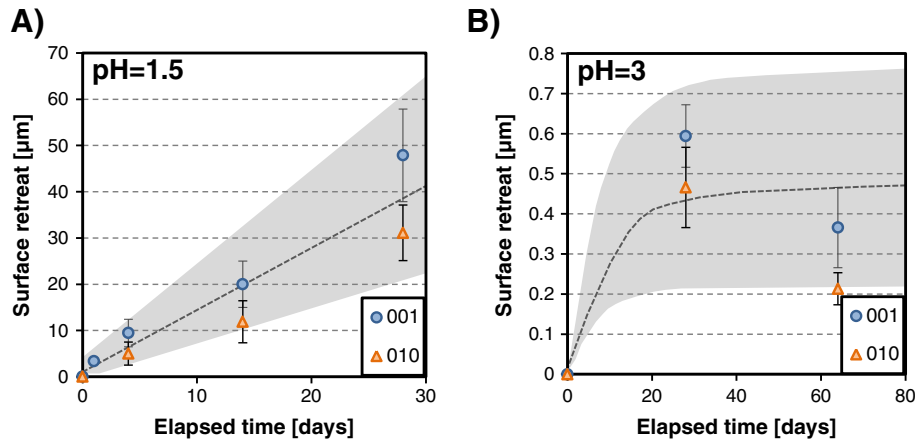


Fig. 6. Surface retreat as a function of reaction time for (001) and (010) orientations for pH = 1.5 (A) and pH = 3 (B) experiments conducted in silica-low solutions, indicative of the reactivity of the external interface. The large error bars are due to the significant increase of surface roughness.

2016); (ii) the aqueous silica level exerts a first order control on the labradorite dissolution rate (Fig. 13), as expected when mineral surfaces are covered with passivating ASSL (e.g. Daux et al., 1997; Grambow and Muller, 2001; Daval et al., 2011; Daval et al., 2013; Johnson et al., 2014).

4.2. Influence of control parameters and mechanisms involved

4.2.1. Internal interface

As illustrated on Fig. 5 and Fig. 12, higher pH values resulted in larger relative decrease of labradorite reaction rate during the reaction. This may be explained by the pH-sensitivity of the constituent phase of the surface layer. The phase consisted here in amorphous silica at pH = 1.5, and an Al/Si-rich or Si-rich phase at higher pH, in agreement with studies of thin surface layers developed on altered feldspars (e.g. Nugent et al., 1998; Kawano and Tomita, 2001; Zhu et al., 2006; Hellmann et al., 2012).

As reported for ASSLs formed on silicate glasses (e.g. Verney-Carron et al., 2010), we propose that the properties of the surface layers formed on minerals are pH-dependent, resulting in different transport properties. The degraded transport properties may then influence the mineral dissolution, with non-passivating characteristics at pH = 1.5 and gradual transport control with increasing pH. The progressive increase of passivating properties through time is reflected by a release profile of cations that did not strictly follow a parabolic law (proportional to $t^{1/2}$).

This could be explained by the progressive evolution of the layer’s properties through time due to continuous fluid-layer contact (Putnis et al., 2005; Putnis, 2009) similar to ageing of silica sols (Melero-Garcia et al., 2009), or the ageing of amorphous precursor to secondary phases, such as pseudoboehmite or pregibbsite gels (McHardy and Thomson, 1971; Bottero et al., 1980; Bird et al., 1994; Tsuchida, 2000).

The pH-dependent transport properties of surface layers may be explained by the competition between the intrinsic kinetics of labradorite dissolution and surface layer densification (see Daval et al., 2009b). Schematically, if labradorite dissolution is faster than layer densification, then the dissolution may proceed unhindered. On the other hand, if layer densification is faster than labradorite dissolution, it will ultimately result in passivation of the mineral surface. Arguably, the initial dissolution rate of labradorite dissolution is pH-dependent, and increases with decreasing pH (compare, for instance, initial rate reported for pH = 1.5 and pH = 3 on Fig. 12). Alternatively, oligomerization of aqueous silica, which represents the first steps towards SiO₂ (am) formation, is also pH-dependent (e.g. Icopini et al., 2005). The oligomerization rate increases with increasing pH. If the oligomerization of SiO₂(aq) is considered as a relevant proxy for silica layer densification, then it is obvious that the densification rate is faster when pH is less acidic. As a consequence, both labradorite dissolution rate and layer densification rate evolve in such a way that passivation is expected in less acidic environments, in fairly good agreement with our results. This reasoning would explain why ASSLs do not passivate the labradorite surface at pH = 1.5, in spite of their thickness.

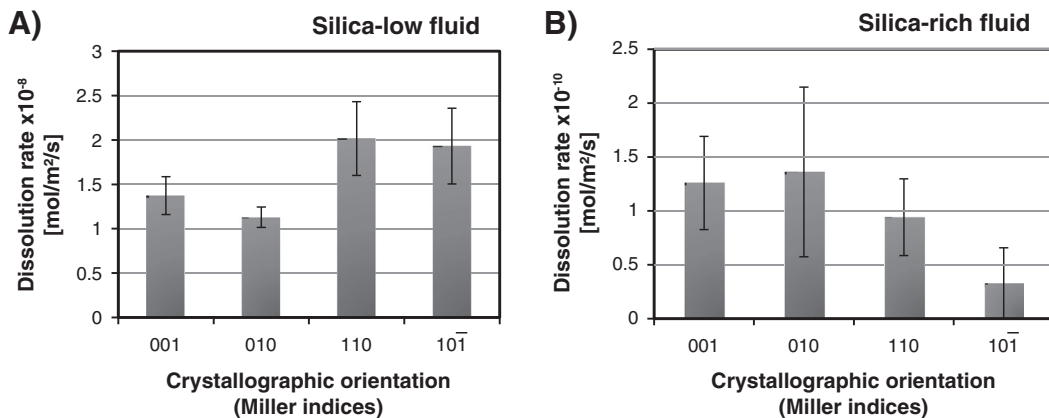


Fig. 7. Dissolution rate at the external interface in mol/m²/s for various orientations tested at pH = 2.5 in both silica-low (A) and silica-rich (B) solutions. The dissolution rates were calculated based on VSI measurement, and the error bars are derived from uncertainty on retreat measurements.

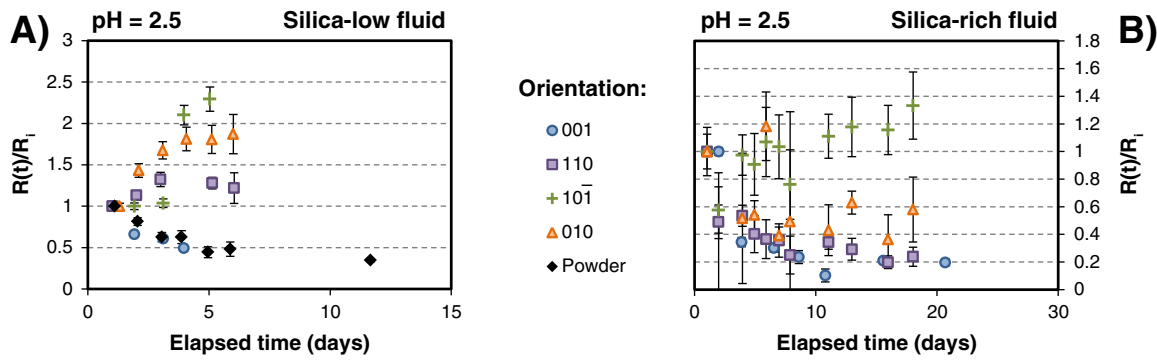


Fig. 8. Temporal variation of the ratios of instantaneous rates ($R(t)$) at the internal interface to the initial rate (R_i) for experiments at $\text{pH} = 2.5$. Cation release was recorded independently for each specific orientation. A ratio value $R(t)/R_i < 1$ accounts for a reduction of the reaction rate with time, while $R(t)/R_i > 1$, stands for an increase of the reaction rate. Error bars indicate error on ratios due to uncertainties on ICP-AES measurements and were calculated following the methodology presented in Section 2.7.

To further understand the mechanisms affecting the bulk labradorite dissolution, we tested the effect of anisotropy on the passivating ability of the surface layers in dissolution experiments with specific crystal orientations (Fig. 8). The most striking feature is that of the relative decrease of dissolution rate over time of the main labradorite cleavage (001), corresponds to that observed in the powder experiments at $\text{pH} = 2.5$ in unsaturated conditions (Fig. 8A). This shows that the passivation behavior of the statistically most abundant orientation within a mineral powder, reasonably assumed here to be the preferential cleavage of labradorite (001), is representative of passivation behavior of the powder. Moreover, the effect of surface layers is face-specific. The behavior of (110) orientation, which was not passivated in pure HCl solution ($\text{pH} = 2.5$), was close to that of powder and (001) experiments when saturated with respect to amorphous silica. Orientation (010) displays the same evolution, while (10 $\bar{1}$) was not passivated under either condition. On the whole, (001) orientation seems to be the only passivated orientation in an undersaturated fluid, while (10 $\bar{1}$) was the only non-passivated orientation among those tested in a saturated fluid. Consistent with the mechanism leading to passivation proposed above, the fact that the (001) face is among the slowest dissolving faces of feldspars (Zhang and Lüttge, 2009; Pollet-Villard et al., 2016) may explain why it was the sole face that was systematically passivated at $\text{pH} = 2.5$.

To sum up, the temporal evolution of the dissolution rate of the internal interface is likely controlled by the spontaneous evolution of the textural properties of the surface layer, such as its densification,

which is pH-dependent. The evolution of the external interface dissolution rate further suggests an intrinsic change of the structural properties of the surface layer itself.

4.2.2. External interface

The dissolution rate of the external interface is globally isotropic (Fig. 7), which indirectly supports that the structure and chemical composition of the surface layers are similar regardless of the considered face. Moreover, rates reported at $\text{pH} = 2.5$ are about two orders of magnitude slower in saturated fluids (ranging from $3.3 \cdot 10^{-11}$ to $1.4 \cdot 10^{-10} \text{ mol} \cdot \text{m}^{-2} \cdot \text{s}^{-1}$) than those for non-saturated fluids under the same conditions (from $1.1 \cdot 10^{-8}$ to $2.0 \cdot 10^{-8} \text{ mol} \cdot \text{m}^{-2} \cdot \text{s}^{-1}$, Table 2). Such a dramatic sensitivity of the dissolution rate of the external interface on the aqueous concentration of silica indicates that the surface layer is primarily made of silica. An additional striking observation is that the reactivity of the external interface gradually decreased as a function of time at $\text{pH} = 3$, whereas it remained constant at $\text{pH} = 1.5$ (Fig. 6). Since the dissolution of the external interface is chemically-controlled (as opposed to the dissolution of the internal interface which has been shown to be controlled by the transport properties of the layer), the most likely explanation for the intrinsic decrease of the surface layer reactivity is that it either chemically or structurally evolved with time. In particular, a similar chemically-controlled dependence of the surface layer dissolution rate on its structure was previously shown by the decline of the release of Si from wollastonite during the progressive polymerization of the ASSL (Schott et al., 2012).

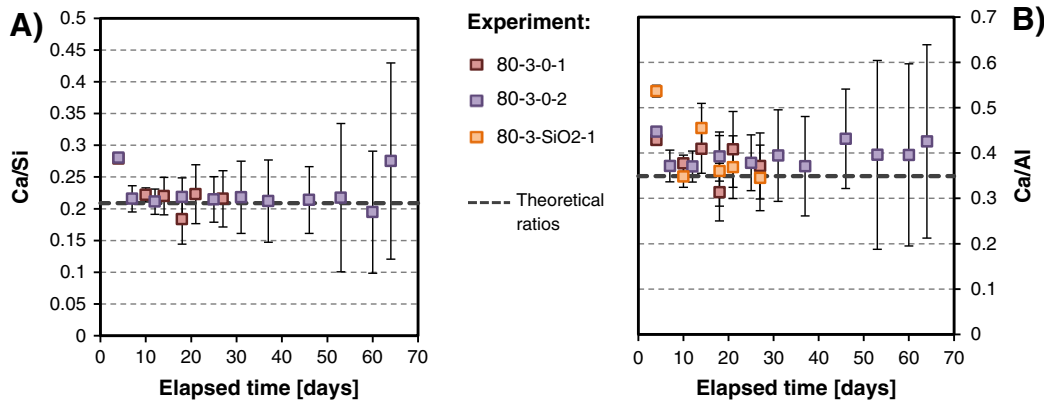


Fig. 9. Ca/Si (A) and Ca/Al (B) elemental ratios for $\text{pH} = 3$ experiments conducted in aqueous silica-low (red and purple squares) or in silica-rich solutions (orange squares). Error bars indicate error on elemental ratios derived from uncertainties on ICP-AES measurements. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

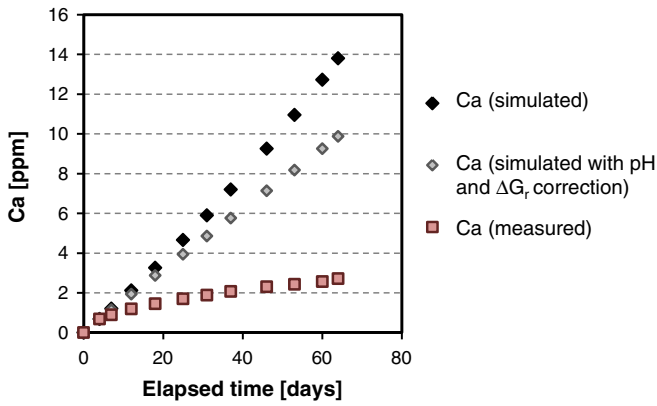


Fig. 10. Elemental release for 3-80-0-2 experiment conducted at pH = 3 in a silica-low solution (red squares) compared with the theoretical values (black diamonds) and taking into account measured pH and ΔG_c drifts (grey diamonds). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

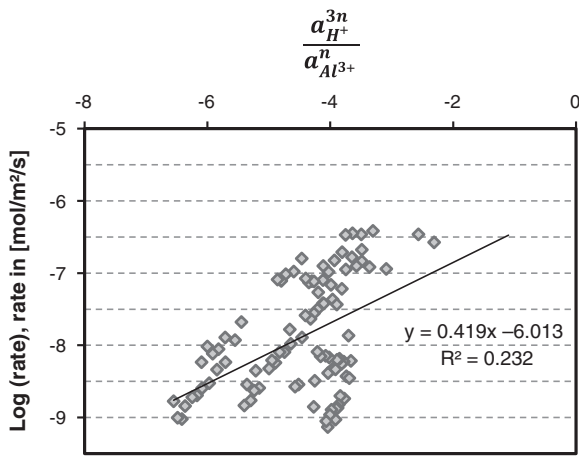


Fig. 11. Log(rate) versus Al effect parameter (Oelkers et al., 1994).

Altogether, our results point out that the dissolution of labradorite at pH > 1.5 is characterized by an evolution of the reactivity of the external and internal interfaces. This evolution may be attributed to concomitant changes in the structural and textural properties of the surface layers, that are pH-dependent.

4.3. Implication of interfacial control of dissolution rates of altered surface layer

To the best of our knowledge, the interpretations proposed above are not at odds with previously published data. In particular, several studies (Knauss and Wolery, 1986; Carroll and Knauss, 2005) reported cation release profiles that were not strictly linear, despite tremendous efforts involved in the solid phase preparation.

Under field conditions, surface layers have been observed to form under various weathering conditions, either on laboratory-grade (Nugent et al., 1998) or on environmental feldspar samples (Kawano and Tomita, 2001; Zhu et al., 2006; Hellmann et al., 2012). The passivating mechanism revealed in the present study may therefore be relevant to field conditions. A systematic characterization of the textural and compositional properties of feldspar mineral surface layers (e.g. the diffusivity (Gin et al., 2015), density (Rebiscoul et al., 2003; Rebiscoul et al., 2004) and texture (Schott et al., 2012)) could therefore be considered at the field scale. However, results from such studies may be complicated by the thickness of the layers, which in most cases prevent obtaining quantitative data (Zhu et al., 2006).

Because passivation is an increasing function of pH, we anticipate that the process associated with the reduction of the dissolution rates, such as the evolution of textural or structural properties of interfacial layers, will be particularly relevant at pH typically found in continental soils. The effect of lowering temperature from 80 °C to ambient temperature, however, remains unclear. The respective contribution of labradorite dissolution and layer densification to the dissolution reduction at lower temperature is a function of the activation energy of these processes, which is currently unknown. However, previous studies suggest that for minerals such as olivine or wollastonite, passivation is enhanced at lower temperature (Daval et al., 2009a). Consequently, a much slower maturation of the surface layer is expected at field-relevant temperatures.

5. Conclusion

The present study shows that the reactivity of labradorite feldspar at pH > 1.5, rather than being constant, is controlled by the ageing of the outermost part of the mineral, at the interface with the reacting fluid. The dissolution patterns of labradorite feldspar in acidic fluid at 80 °C enabled us to relate the evolution of the reaction rate to the formation of a surface layer. The solution pH appeared to control the textural and structural properties at this interface, ultimately governing the rate of matter and energy exchanges. At pH = 1.5, surface layers are permissive enough to enable direct fluid-mineral interaction. In

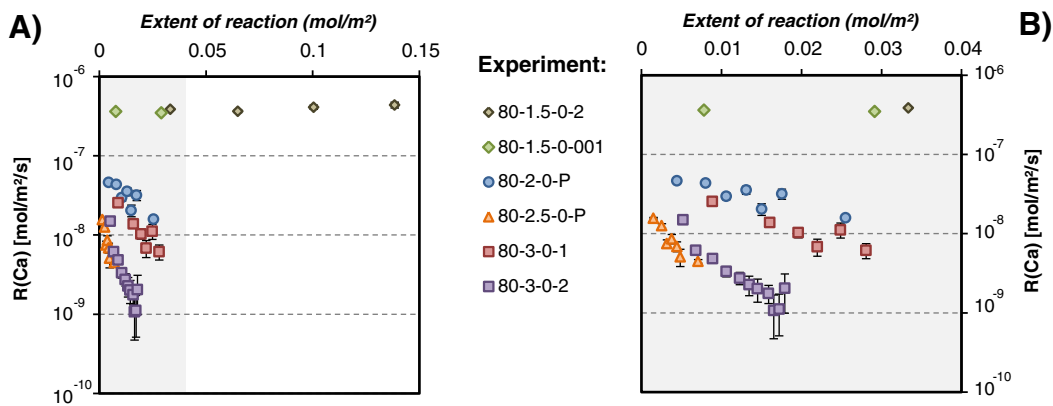


Fig. 12. Reaction rate (mol/m²/s) based on Ca release into solution as a function of the extent of reaction (mol/m²) at various pH conditions. Diamonds correspond to experiments conducted at pH = 1.5 on polished surfaces, circles to powder experiments at pH = 2, triangles to powder experiments at pH = 2.5 and squares to experiments on polished surfaces at pH = 3. (B) is a close-up view of (A). Reported errorbars account for ICP-AES uncertainties as calculated via error propagation.

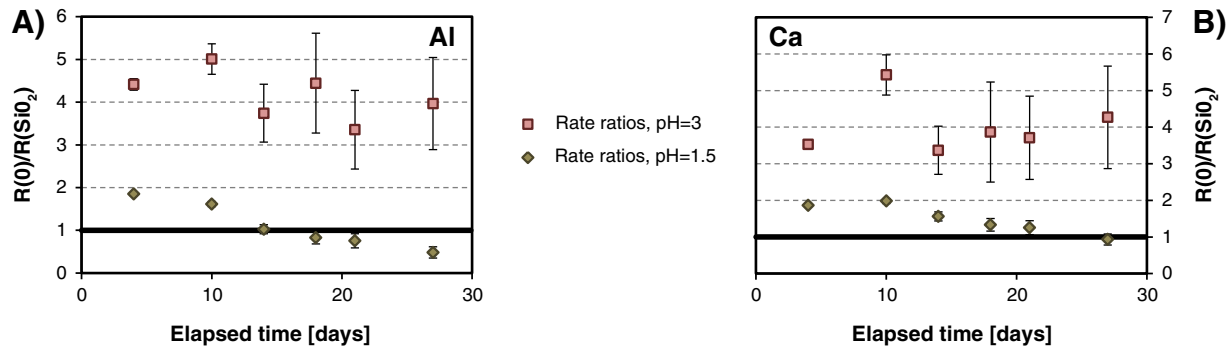


Fig. 13. Rate ratios for experiments conducted at 80 °C in HCl solutions at pH = 1.5 (green diamonds) and pH = 3 (red squares). A value >1 accounts for a higher rate recorded in experiments conducted in silica-low solutions. Results are based on Al (A) and Ca (B) data. Error bars indicate error on ratios due to uncertainties on ICP-AES measurements. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

contrast, layers formed at higher pH values exhibited increasing passivating properties over time due to putative changes in their textural and structural properties. The decrease of the reaction rate also depends on crystallographic orientation, as revealed by a series of face-specific experiments. In this way, the behavior of the principal cleavage of labradorite, namely the (001) crystallographic orientation, was shown to be representative of the bulk behavior of associated mineral powder.

Principal outcomes of this study are: (i) the current quasi univocal understanding of mineral reactivity in terms of dissolution rate constants should be revised in favor of more specific scenarios that would account for interfacial properties, (ii) further efforts should be directed towards a better understanding of the functioning of the fluid/mineral interface on different crystal faces, including the most exposed one (001), and (iii) the early to transitional stages of mineral dissolution should be considered, as they may contain critical information to understand mineral reactivity in field-relevant conditions.

Supplementary data to this article can be found online at <http://dx.doi.org/10.1016/j.chemgeo.2016.08.035>.

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