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Enhanced thermoelectric performance of SnSe by controlled vacancy population

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Abstract

The thermoelectric performance of SnSe strongly depends on its low-energy electron band structure that provides high density of states in a narrow energy window due to the multi-valley valence band maximum (VBM). Angle-resolved photoemission spectroscopy measurements, in conjunction with first-principles calculations, reveal that the binding energy of the VBM of SnSe is tuned by the population of Sn vacancy, which is determined by the cooling rate during the sample growth. The VBM shift follows precisely the behavior of the thermoelectric power factor, while the effective mass is barely modified upon changing the population of Sn vacancies. These findings indicate that the low-energy electron band structure is closely correlated with the high thermoelectric performance of hole-doped SnSe, providing a viable route toward engineering the intrinsic defect-induced thermoelectric performance via the sample growth condition without an additional *ex-situ* process.

Keywords Thermoelectric, Defect engineering, Electron band structure, Vacancy, SnSe

1 Introduction

Thermoelectric materials are of great importance for renewable energy technology since they can generate electricity from the temperature gradient of wasted heat [1–4]. The quantity of interest that determines the thermoelectric performance is a dimensionless figure of merit $ZT = S^2\sigma T/\kappa$, which is influenced by the electronic and phononic properties of the thermoelectric material. Here S , σ , T , and κ are the Seebeck coefficient, electric conductivity, temperature, and thermal conductivity, respectively, while $S^2\sigma$ is called a power factor (PF). One of the ultimate goals in the research of thermoelectric materials is to achieve a high ZT value, which is generally considered a challenging task because the physical parameters comprising ZT are correlated and often act adversely on each other [1, 4]. One effective strategy to optimize both the electrical and thermal properties is defect engineering, which can simultaneously increase the charge carrier concentration and reduce the lattice thermal conductivity in the material [5–8]. The electronic contribution to the thermoelectric performance that is mainly characterized by PF , however, is directly understood by the electron

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band structure, while the phonon dispersion provides essential information on the phononic contribution.

SnSe is one of the remarkable van der Waals materials with great potential for applications [9, 10]. It has attracted recent research interest owing to its high ZT value, due to the ultrahigh PF , which is attributed to a multi-valley valence band maximum (VBM) [11–15] and band renormalization, i.e., change in the effective mass of the multi-valley VBM [11–15], in conjunction with low thermal conductivity attributed to anisotropic and anharmonic phonon dispersions [16, 17]. However, while the multi-valley VBM was recently observed through angle-resolved photoemission spectroscopy (ARPES) studies [18–21], the correlation between the electron band structure and the thermoelectric performance has not been well understood.

In this article, we report a systematic study on the evolution of the electron band structure of a SnSe single crystal with a controlled hole carrier density, investigated using ARPES measurements and first-principles density functional theory (DFT) calculations. The PF value of SnSe depends on the cooling rate during its growth process that determines the population of Sn vacancy [22]. ARPES data reveal that the slower cooling rate also leads to the overall shift of the multi-valley VBM towards the lower binding energy, indicating hole doping of SnSe. The VBM shift exactly follows the behavior of the PF value [22]. Our study reveals the role of the low-energy

electron band structure of SnSe on its thermoelectric performance that is tuned by the hole-carrier density.

2 Methods

Single crystal SnSe was synthesized with several different cooling rates as described elsewhere [22]. ARPES measurements were performed at the Beamlines 4.0.3 and 10.0.1 of the Advanced Light Source (ALS), Lawrence Berkeley National Laboratory. Single crystal SnSe was cleaved and measured in an ultra-high vacuum with a base pressure of 4×10^{-11} Torr. The measurements were done at 20 K using photon energies of 72 eV and 60 eV. Prior to the ARPES measurement, the charging effect has been examined for all the samples (see Additional file 1). The energy and momentum resolutions were set to be 18 meV and 0.01 \AA^{-1} , respectively. The electron band structure of SnSe was calculated utilizing the DFT implemented in full-potential linearized augmented plane wave package [23], within Perdew-Burke-Ernzerhof functional combined with modified Becke-Johnson potential (PBE + mBJ) [24]. The coherent potential approximation (CPA) within Korringa-Kohn-Rostoker (KKR) Green's function method implemented in SPR-KKR package [25] was adopted to understand the role of Sn vacancies on the electron band structure. For better determination of the Fermi energy, the Lloyd formula has been used [26].

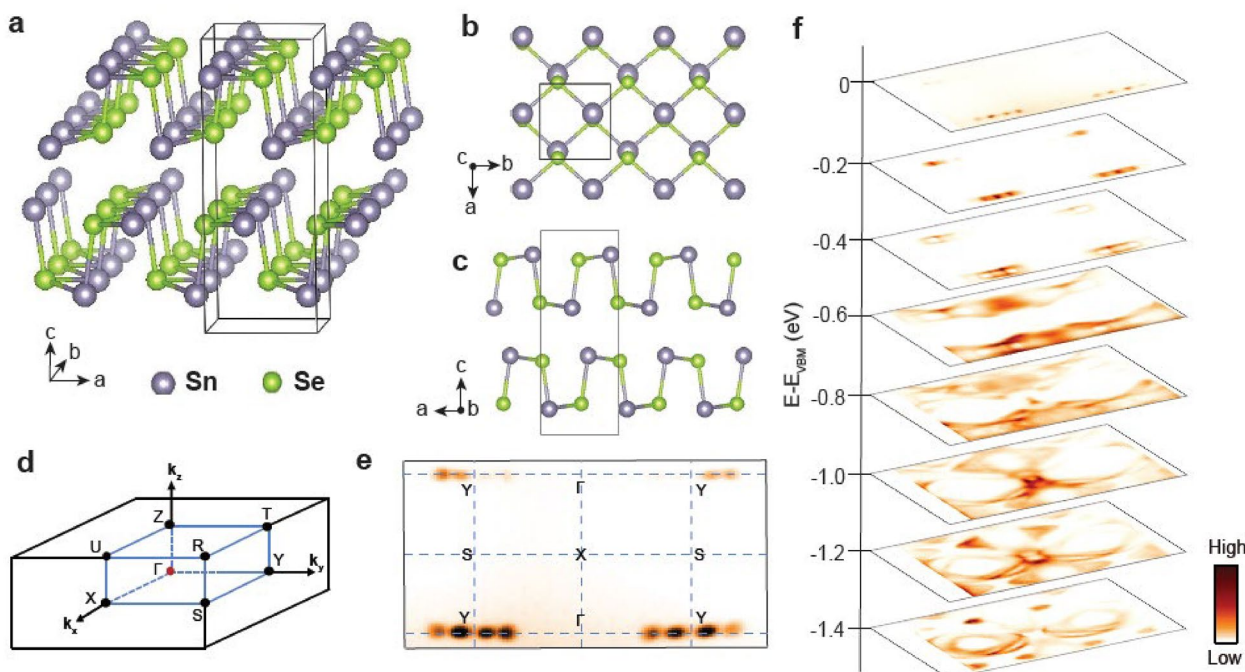


Fig. 1 The crystal structure and ARPES data of SnSe. **a–c** The crystal structure of SnSe (**a**) with a top (**b**) and side views (**c**). The black lines indicate the conventional unit cell of SnSe. **d** A Brillouin zone of the $Pnma$ phase. **e** A constant energy ARPES intensity map taken at VBM. **f** Constant energy ARPES intensity maps taken at several different $E - E_{\text{VBM}}$ from -1.4 eV to 0 eV

3 Results and discussion

3.1 Electron band structures of SnSe

The crystal structure of SnSe belongs to the space group $Pnma$ with lattice constants $a=4.15$ Å, $b=4.44$ Å, and $c=11.57$ Å, as shown in Fig. 1a–c [27]. Two layers of SnSe are stacked along the c -direction with the weak van der Waals interaction [28], constituting the SnSe unit cell depicted in Fig. 1a. Figure 1e shows a constant energy ARPES intensity map taken at the energy corresponding to VBM (E_{VBM}), since SnSe is an insulator with a band gap of 0.8–0.9 eV [11, 15, 29]. Equally-spaced four dots are observed around the Y point that expand to larger-sized circles at higher binding energy, as shown in Fig. 1f, indicating that these bands are hole bands.

Figure 2 shows the low-energy electron band structure of SnSe. Figure 2a is the electron band structure measured by ARPES along the Y- Γ -Y direction in the Brillouin zone of the SnSe (Fig. 1d, e). Around the Y point, four humps and three dips are observed close to E_{VBM} . Two of the humps in the first Brillouin zone are denoted by α and β . To understand the observed ARPES spectra, the electron band structure of SnSe was calculated using the PBE+mBJ method, as shown in Fig. 2b. The observed ARPES spectra are in agreement with the calculated electron band structure, showing the characteristic four humps around the Y point. The calculated energy gap is 0.86 eV, which is consistent with previous results [11, 15, 29], indicating that the PBE+mBJ method well describes the electronic correlations in SnSe. The difference between the measured and calculated electron band structure is the lack of photoelectron intensity at 1.0–1.5 eV below E_{VBM} around the Γ point and the bandwidth of the low-energy electron band structure. While the former is attributed to the matrix element effect, the decrease of the bandwidth by 17% in the calculated band structure (compared at the bottom of the band at half the unit cell, i.e., $\frac{1}{2}\Gamma\text{Y}$) might originate from the slight difference in the structural parameters [18, 19].

Along the Γ -X direction, another hole band denoted by γ is observed with its top at 0.2 eV below E_{VMB} as shown in Fig. 2c. Along the Y-S direction, the top of the hole band is observed at 0.1 eV below E_{VBM} , indicating that the energy difference between the hump and dip observed close to E_{VMB} is 0.1 eV as shown in Fig. 2c. The observed ARPES spectra are consistent with previously reported results taken at $k_z=16.8(\pi/c)$ [18], corresponding to a photon energy of 72 eV that has been also used in our experiments.

3.2 Engineering thermoelectric performance

In order to study the effect of the cooling rate during SnSe single crystal growth on the electron band structure, Fig. 3a, b show ARPES spectra measured along

the Y- Γ -Y direction for two representative SnSe crystals grown with different cooling rates. The dashed line denotes the E_{VBM} of the data shown in Fig. 2a, corresponding to a cooling rate of 0.5 K/h. At 1 K/h, E_{VBM} remains almost the same as that of 0.5 K/h. However, with increasing cooling rate to 5 K/h, the overall multi-valley band shifts toward higher binding energy, indicating that the cooling rate tunes the charge carrier density of SnSe. Since the cooling rate during the growth process determines the population of the Sn vacancy, as shown in recent transport and STM study [22], the electron band structures have been calculated for different Sn vacancy levels using the KKR-CPA method (Fig. 3c, d). Here, x denotes the population of Sn vacancies compared to the number of Sn atoms in a perfect SnSe crystal, implying that the stoichiometry of the sample is Sn_{1-x}Se . With increasing x , the overall multi-valley VBM shifts toward lower binding energy. The non-dispersive fuzzy states

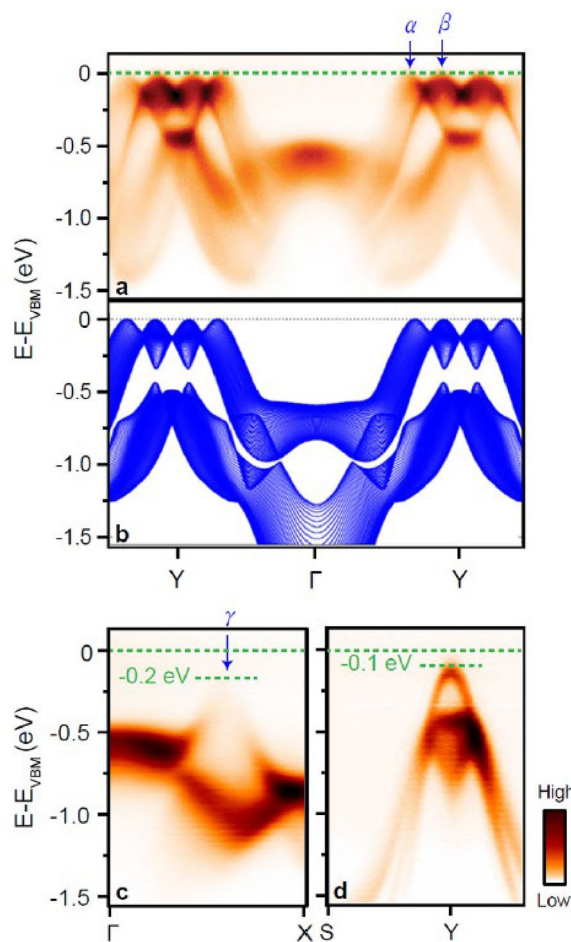


Fig. 2 ARPES spectra and theoretical calculations. **a** ARPES spectra of a SnSe single crystal with a crystal growth cooling rate of 0.5 K/h taken along the Y- Γ -Y direction. **b** The calculated electron band structure of SnSe along the Y- Γ -Y direction. **c**, **d** ARPES spectra taken along the Γ -X (**c**) and S-Y (**d**) directions

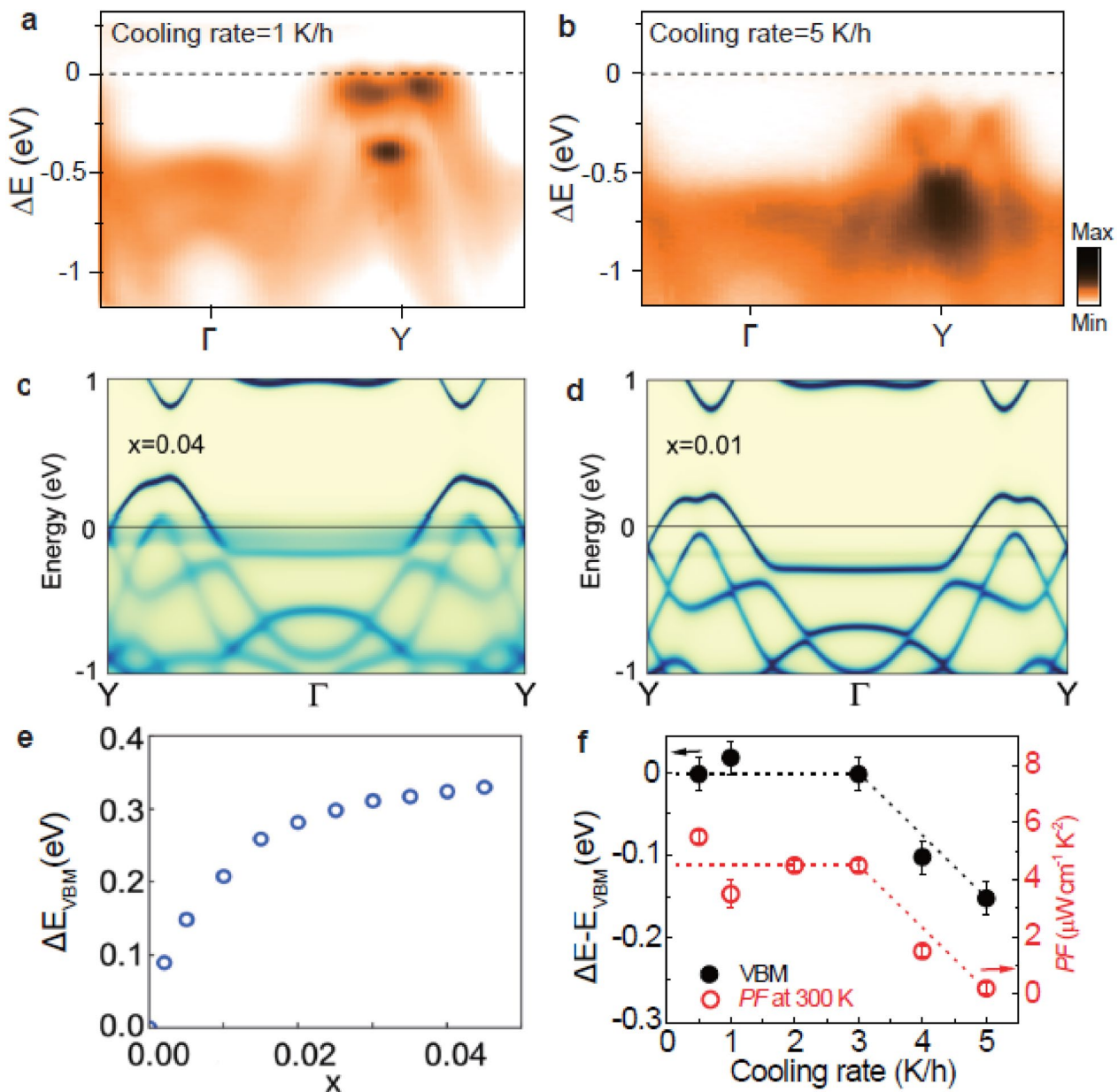


Fig. 3 Relation between PF and the electron band structure. **a, b** ARPES intensity maps taken along the Γ -Y direction of SnSe with a crystal growth cooling rate of 1 K/h (**a**) and 5 K/h (**b**). ΔE is the energy relative to the VBM of the 0.5 K/h sample. **c, d** A calculated electron band structure along the Γ -Y- Γ direction using the KKR-CPA method for $Sn_{1-x}Se$ with $x=0.04$ (**c**) and $x=0.01$ (**d**). **e** Doping dependence of the VBM obtained by calculations. **f** Crystal growth cooling rate dependence of $\Delta E - E_{VBM}$ (closed circles) and PF (open circles) extracted from the ARPES data and previous transport results [22], respectively

that become obvious at $x=0.04$ originate from impurity scattering. The shift of E_{VBM} as a function of doping ratio x , estimated from the shift of the VBM spectral weight in KKR calculation, is summarized in Fig. 3e. With increasing x , ΔE_{VBM} gradually increases, indicating that Sn vacancies lead to hole doping of SnSe. Based on the comparison, the shift of E_{VBM} with the increasing cooling rate shown in Fig. 3a, b indicates the decreasing population of

Sn vacancies. As a result, a lower cooling rate generates higher hole doping of SnSe.

Figure 3f shows the cooling rate dependence of E_{VBM} (black filled circles), compared to that of PF [22] (red empty circles). E_{VBM} does not show a notable change up to a cooling rate of 3K/h. When the cooling rate further increases, E_{VBM} decreases, indicating that SnSe gradually has less hole doping due to Sn vacancies. Surprisingly,

ΔE_{VBM} exactly follows the cooling rate dependence of PF taken at 300K [22]. This finding suggests that the high PF , observed in hole-doped SnSe [11–15], is closely correlated with the charge carrier density (hole concentrations), i.e., the population of Sn vacancies, which is determined by the cooling rate during the single crystal growth of SnSe.

An alternative origin of the high PF value observed in hole-doped SnSe [11–15] is the light effective mass of the multiple hole bands, i.e., α , β , and γ bands, which tends to strongly influence the electric properties of a material. Indeed, parabolic fitting to the top of the α , β , and γ bands shown in Fig. 2 corresponding to a cooling rate of 0.5 K/h confirms the light effective mass of $m_{\Gamma X}^{\alpha}=0.19\pm 0.02 m_e$ and $m_{\Gamma Y}^{\alpha}=0.11\pm 0.01 m_e$ for the α band, $m_{\Gamma X}^{\beta}=0.15\pm 0.01 m_e$ and $m_{\Gamma Y}^{\beta}=0.21\pm 0.02 m_e$ for the β band, and $m_{\Gamma X}^{\gamma}=0.08\pm 0.02 m_e$ and $m_{\Gamma Y}^{\gamma}=0.09\pm 0.02 m_e$ for the γ band, when m_e , $m_{\Gamma Y}^*$, and $m_{\Gamma X}^*$ are free electron mass, when the effective mass of each band estimated parallel and perpendicular to the Γ -Y direction, respectively. The effective masses obtained from ARPES and band structure calculations are consistent with the previous results [18, 19]. At a faster cooling rate of 5 K/h, the effective masses of the α and β bands along the Γ -Y direction are estimated to be $m_{\Gamma Y}^{\alpha}=0.13\pm 0.02 m_e$ and $m_{\Gamma Y}^{\beta}=0.20\pm 0.03 m_e$, which remain almost the same within the fitting error. As a result, the effective mass does not affect PF significantly.

4 Conclusions

In summary, the electron band structure of a SnSe single crystal has been investigated using ARPES. The different cooling rate during the growth process leads to the overall multi-valley band shift of the electron band structure of SnSe, indicating that the cooling rate tunes the population of Sn vacancies and hence the charge carrier density, which is supported by first-principles calculations. Surprisingly, the shift of the electron band structure exactly follows the cooling rate dependence of the thermoelectric power factor of SnSe. These findings suggest a simple but efficient way to fabricate an intrinsic defect-induced high-efficiency thermoelectric phase of SnSe and provide a viable route toward the engineering of thermoelectric performance via the sample growth condition without an additional *ex-situ* process.

Abbreviations

VBM	Valence band maximum
PF	Power factor
ARPES	Angle-resolved photoemission spectroscopy
DFT	Density functional theory
PBE + mBJ	Perdew-Burke-Ernzerhof functional combined with modified Becke-Johnson potential
CPA	Coherent potential approximation
KKR	Korringa-Kohn-Rostoker

Supplementary Information

The online version contains supplementary material available at <https://doi.org/10.1186/s40580-023-00381-7>.

Additional file 1: Figure S1. Comparison of core-level spectra for SnSe with or without the photoemission charging effect. **Figure S2.** ARPES E-k dispersions along the Γ -Y direction for SnSe single crystals with different growth cooling rates.

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Author contributions

J-EL, HR, CH, and SKM proposed and designed the research. VQN and SC performed single-crystal growth. J-EL, JH, HR, and CH carried out the ARPES measurements and analyzed the ARPES data with assistance from JDD and SKM; KK carried out the density functional calculations and provided theoretical support with assistance from BIM; J-EL, JH, KK, and HR wrote the manuscript and revised it with assistance from CH and SKM. All authors contributed to the scientific planning and discussions.

Declarations

Competing interests

The authors declare that they have no competing interests.

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References

- S.-D. Chen, Y. He, A. Zong, Y. Zhang, M. Hashimoto, B.-B. Zhang, S.-H. Yao, Y.-B. Chen, J. Zhou, Y.-F. Chen, S.-K. Mo, Z. Hussain, D. Lu, Z.-X. Shen, *Phys. Rev. B* **96**, 081109 (2017)
- J.P. Heremans, M.S. Dresselhaus, L.E. Bell, D.T. Morelli, *Nat. Nanotechnol.* **8**, 471 (2013)
- G.J. Snyder, E.S. Toberer, *Nat. Mater.* **7**, 105 (2008)
- J. Xin, Y. Tang, Y. Liu, X. Zhao, H. Pan, T. Zhu, *NPJ Quantum Mater.* **3**, 9 (2018)
- Y. Zheng, T.J. Slade, L. Hu, X.Y. Tan, Y. Luo, Z.-Z. Luo, J. Xu, Q. Yan, M.G. Kanatzidis, *Chem. Soc. Rev.* **50**, 9022–9054 (2021)
- L. Hu, T. Zhu, X. Liu, X. Zhao, *Adv. Funct. Mater.* **24**, 5211–5218 (2014)
- H. Jang, J.H. Park, H.S. Lee, B. Ryu, S.-D. Park, H.-A. Ju, S.-H. Yang, Y.-M. Kim, W.H. Nam, H. Wang, J. Male, G.J. Snyder, M. Kim, Y.S. Jung, M.-W. Oh, *Adv. Sci.* **8**, 2100895 (2021)
- W. He, D. Wang, H. Wu, Y. Xiao, Y. Zhang, D. He, Y. Feng, Y.-J. Hao, J.-F. Dong, R. Chetty, L. Hao, D. Chen, J. Qin, Q. Yang, X. Li, J.-M. Song, Y. Zhu, W. Xu, C. Niu, X. Li, G. Wang, C. Liu, M. Ohta, S.J. Pennycook, J. He, J.-F. Li, L.-D. Zhao, *Science* **365**, 1418–1424 (2019)
- P. Ranjan, S. Gaur, H. Yadav, A.B. Urgunde, V. Singh, A. Patel, K. Vishwakarma, D. Kalirawana, R. Gupta, P. Kumar, *Nano Converg.* **9**, 26 (2022)
- B.C. Wyatt, S.K. Nemani, B. Anasori, *Nano Converg.* **8**, 16 (2021)
- L.-D. Zhao, G. Tan, S. Hao, J. He, Y. Pei, H. Chi, H. Wang, S. Gong, H. Xu, V.P. Dravid, C. Uher, G.J. Snyder, C. Wolverton, M.G. Kanatzidis, *Science* **351**, 141 (2016)

12. L.-D. Zhao, C. Chang, G. Tan, M.G. Kanatzidis, *Energy Environ. Sci.* **9**, 3044–3060 (2016)
13. K. Peng, X. Lu, H. Zhan, S. Hui, X. Tang, G. Wang, J. Dai, C. Uher, G. Wang, X. Zhou, *Energy Environ. Sci.* **9**, 454–460 (2016)
14. G. Shi, E. Kioupakis, *J. Appl. Phys.* **117**, 065103 (2015)
15. R. Guo, X. Wang, Y. Kuang, B. Huang, *Phys. Rev. B* **92**, 115202 (2015)
16. L.-D. Zhao, S.-H. Lo, Y. Zhang, H. Sun, G. Tan, C. Uher, C. Wolverton, V.P. Dravid, M.G. Kanatzidis, *Nature* **508**, 373 (2014)
17. C.W. Li, J. Hong, A.F. May, D. Bansal, S. Chi, T. Hong, G. Ehlers, O. Delaire, *Nat. Phys.* **11**, 1063 (2015)
18. Q. Lu, M. Wu, D. Wu, C. Chang, Y.-P. Guo, C.-S. Zhou, W. Li, X.-M. Ma, G. Wang, L.-D. Zhao, L. Huang, C. Liu, J. He, *Phys. Rev. Lett.* **119**, 116401 (2017)
19. I. Pletikosić, F. von Rohr, P. Pervan, P.K. Das, I. Vobornik, R.J. Cava, T. Valla, *Phys. Rev. Lett.* **120**, 156403 (2018)
20. C.W. Wang, Y.Y. Xia, Z. Tian, J. Jiang, B.H. Li, S.T. Cui, H.F. Yang, A.J. Liang, X.Y. Zhan, G.H. Hong, S. Liu, C. Chen, M.X. Wang, L.X. Yang, Z. Liu, Q.X. Mi, G. Li, J.M. Xue, Z.K. Liu, Y.L. Chen, *Phys. Rev. B* **96**, 165118 (2017)
21. Z. Wang, C. Fan, Z. Shen, C. Hua, Q. Hu, F. Sheng, Y. Lu, H. Fang, Z. Qiu, J. Lu, Z. Liu, W. Liu, Y. Huang, Z.-A. Xu, D.W. Shen, Y. Zheng, *Nat. Commun.* **9**, 47 (2018)
22. V.Q. Nguyen, T.L. Trinh, C. Chang, L.-D. Zhao, T.H. Nguyen, V.T. Duong, A.T. Duong, J.H. Park, S. Park, J. Kim, S. Cho, *NPG Asia Mater.* **14**, 42 (2022)
23. Blaha, P., Schwarz, K., Madsen, G. K. H., Kvasnicka, D., and Luitz, J. *WIEN2k, an augmented plane wave plus local orbitals program for calculating crystal properties* (Technical University of Wien, 2001).
24. J.P. Perdew, K. Burke, M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865–3868 (1996)
25. Ebert, H. *Fully relativistic band structure calculations for magnetic solids—formalism and application*. (Springer Berlin Heidelberg, 2000).
26. R. Zeller, *J. Phys.: Condens. Matter* **16**, 6453–6468 (2004)
27. S.-U. Kim, A.-T. Duong, S. Cho, S.H. Rhim, J. Kim, *Surf. Sci.* **651**, 5–9 (2016)
28. M. Taniguchi, R.L. Johnson, J. Ghijsen, M. Cardona, *Phys. Rev. B* **42**, 3634–3643 (1990)
29. M. Parenteau, C. Carlone, *Phys. Rev. B* **41**, 5227–5234 (1990)

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