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Publication Date

1994-11-01



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UNIVERSITY OF CALIFORNIA

Materials Sciences Division

National Center for Electron Microscopy

Presented at the Materials Research Society Symposium,
Boston, MA, November 28–December 2, 1994, and to be
published in the Proceedings

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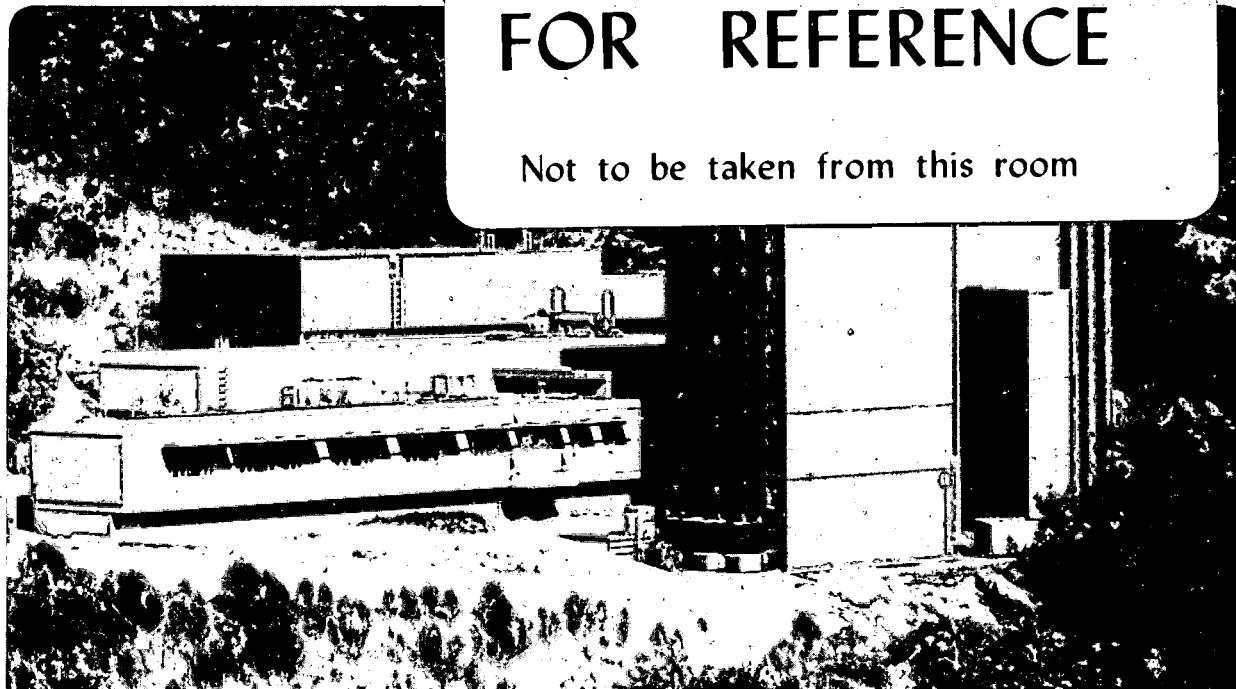
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November 1994

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**Size, Shape and Crystallinity of Luminescent Structures in
Oxidized Si Nanoclusters and H-Passivated Porous Si**

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Proc. MRS, Vol 358, (1995).

This work was supported by the Director, Office of Energy Research, Office of Basic Energy Sciences, Materials Science Division of the U.S. Department of Energy under Contract No. DE-AC03-76SF00098, financial support from the Alexander-von-Humboldt Foundation.

SIZE, SHAPE, AND CRYSTALLINITY OF LUMINESCENT STRUCTURES IN OXIDIZED Si NANOCCLUSERS AND H-PASSIVATED POROUS Si

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ABSTRACT

Near-edge and extended x-ray absorption fine structure measurements from a wide variety of H-passivated porous Si samples and oxidized Si nanocrystals, combined with electron microscopy, ir-absorption, α -recoil, and luminescence emission data, provide a consistent structural picture of the species responsible for the luminescence observed in these systems. For luminescent porous Si samples peaking in the *visible* region, i. e., $\leq 700\text{nm}$, their mass-weighted-average structures are determined here to be *particles* – not wires, whose short-range character is *crystalline* – not amorphous, and whose dimensions – typically $< 15 \text{ \AA}$ – are significantly smaller than previously reported or proposed. These results depend only on sample luminescence behavior, not on sample preparation details, and thus have general implications in describing the mechanism responsible for visible luminescence in porous silicon. New results are also presented which demonstrate that the observed luminescence is unrelated to either the photo-oxidized Si species in porous Si or the interfacial suboxide species in the Si nanocrystals.

INTRODUCTION

There has been intense interest in the visible room-temperature luminescence observed from anodically-grown porous silicon (por-Si) [1], from which a growing consensus has emerged to explain the luminescence with quantum-confined structures [2,3]. There remains, however, a basic lack of knowledge regarding the dimensions, shape, and crystallinity of the species actually responsible for the optical activity. This work summarizes recent x-ray absorption measurements [4] from a series of oxidized Si nanocrystals, whose shapes and sizes are known, and from a variety of anodically-grown (H-passivated) por-Si samples. The data, combined with luminescence emission measurements from each of the systems, establish new and unexpectedly smaller values for the average size of the luminescing Si structures. In addition, local crystallinity is observed and the importance of extended wire shapes is ruled out. New x-ray absorption, luminescence, and ir absorption data show that the observed luminescence in either the nanocrystals or the por-Si samples is not correlated with an oxygen-containing species. Our combined results have important implications for describing the origin of visible photoluminescence from por-Si.

EXPERIMENT

The Si *K*-edge absorption measurements were performed at the National Synchrotron Light Source using the AT&T X15B beamline [5]. A variety of por-Si samples prepared under very different conditions [6,2,7,8] were studied with TEM, x-ray- and ir-absorption, α -recoil, and luminescence excitation and emission spectroscopies. In order to minimize O contamination, which was very easily detectable, air exposure of the freshly prepared samples was limited to <10 min in all but the TEM and photo-oxidation measurements. These latter experiments involved collecting photoluminescence data as a function of laser exposure from various por-Si wafers followed by ir absorption from the differently irradiated and non-irradiated regions. Samples of the O-passivated (air-insensitive) Si nanocrystals [9] were prepared for x-ray absorption measurements by transferring them as colloidal suspensions onto graphite substrates.

RESULTS

NEXAFS from Nanocrystals

In Fig. 1(a) we show Si *K* near-edge x-ray absorption fine structure (NEXAFS) data from three different sizes of oxidized Si nanocrystals, ox-Si_x, labelled small (*s*), medium (*m*), and large (*l*). A fourth, very small (*vs*) sample was also measured but omitted in Fig. 1(a) for clarity. Comparison with data from *c*-Si (freshly HF-rinsed to remove the native oxide film) and bulk SiO₂ readily confirms that these are the two main components in the clusters [9]. A third, minor interfacial suboxide species, SiO_x is also observed and further discussed below. The *c*-Si, SiO₂, and ox-Si_x data have been normalized to common "edge jump" values, measured >100 eV above the Si *K*-edge where their x-ray absorption intensities are structureless, i. e., atomiclike. The relative concentrations of Si and SiO₂ in the nanocrystals (after density corrections) are directly proportional to their individual, easilyseparable edge jumps [10]. Now, x-ray diffraction, liquid chromatography, and TEM data [9] show that the *l*, *m*, *s*, and *vs* samples are SiO₂-coated spheres of Si whose outer

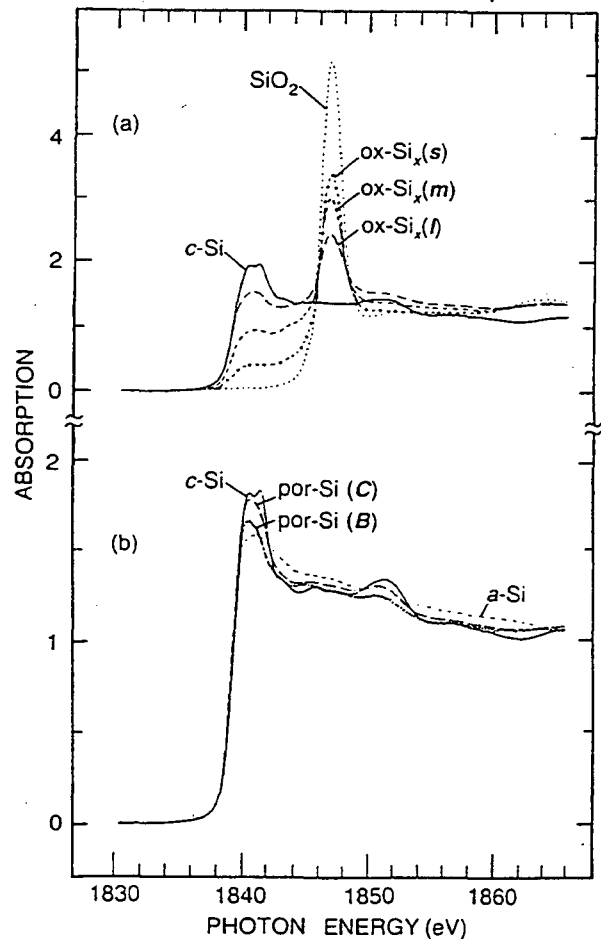


Fig. 1. (a) Si *K*-edge NEXAFS data from *c*-Si, SiO₂, and oxidized Si nanocrystals of different size, labelled small (*s*), medium (*m*), and large (*l*). (b) NEXAFS data from *c*-Si, *a*-Si, and two differently prepared por-Si samples. All data in (a) and (b) are normalized to unity edge jump.

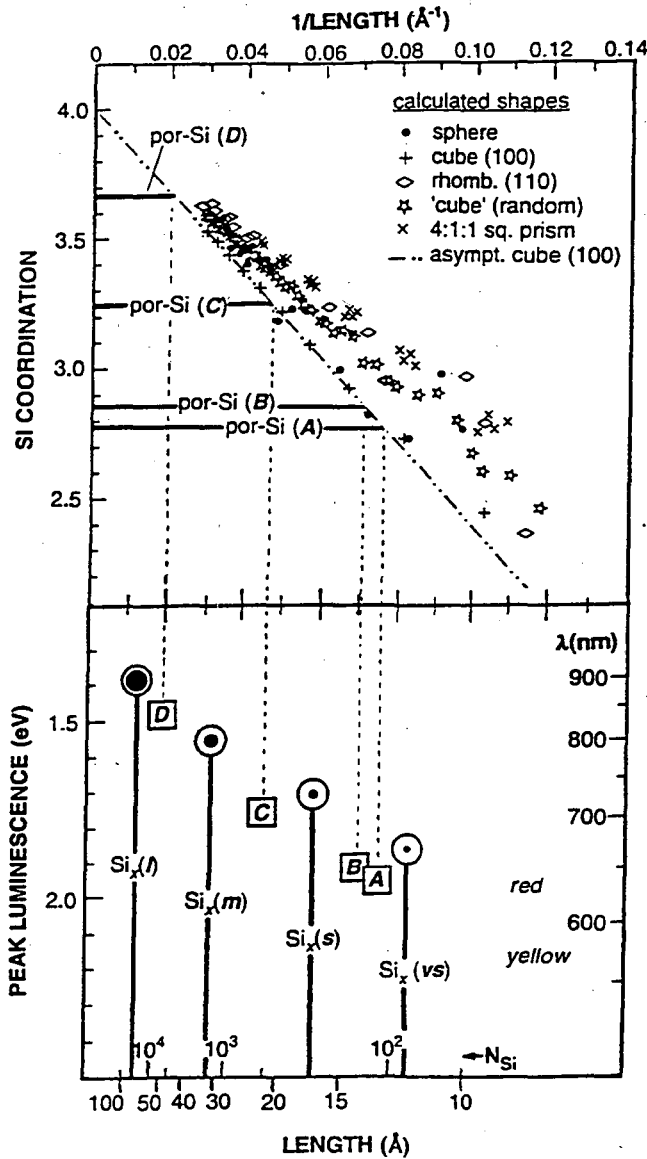
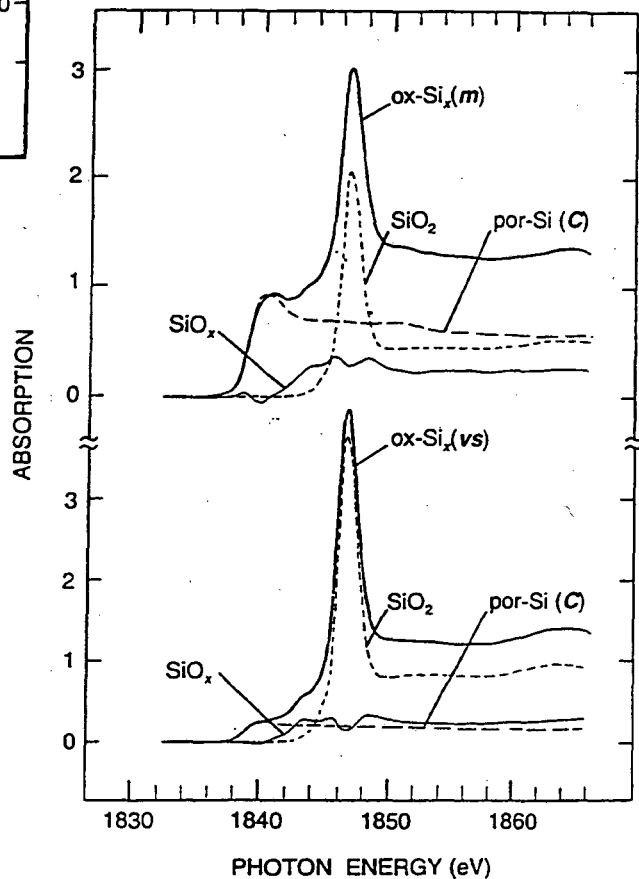


Fig. 3. Si *K*-edge NEXAFS data from two oxidized Si nanocrystals, labelled medium (*m*), and very small (*vs*). Both spectra, normalized to unity edge jump (note different scales), are shown to be composed of three components: an SiO₂ outer shell, a Si inner shell (here represented by a por-Si(*C*) sample from Fig. 1(b)), and an interfacial suboxide species SiO_{*x*}. Despite very different nanocrystal sizes, the thicknesses of the oxide and suboxide shells are essentially identical.

Fig. 2. *Top*: Correlation between average Si coordination for particles of different shapes versus their inverse characteristic length (diameter for sphere, side for cube). Average includes bulk and surface atoms, which have <4 first neighbors, explaining trend of lower values with decreasing particle size. Experimental Si coordinations from EXAFS data of four differently prepared por-Si samples labeled A, B, C, and D are indicated. *Bottom*: Correlation between average Si particle size and measured peak luminescence energy. Also indicated are total number of Si atoms, N_{Si}, contained in a cubic particle of corresponding size. Different sized oxidized Si nanocrystals labeled *vs*, *s*, *m*, and *l* are shown as circles, different por-Si structures from top figure are shown as squares.



diameters are 96 ± 19 , 45 ± 7 , 33 ± 6 , and 26 ± 6 Å, respectively. From these values, the spherical shapes, and the relative concentrations of Si and SiO₂ measured here, we determine the mean inner diameters of the corresponding Si cores to be 78, 31, 17, and 11 Å. These nanocrystal sizes are plotted as circles in the bottom half of Fig. 2 versus their measured peak luminescence energies. The high quantum efficiencies of these particles (>25% at low temperatures [9]), the essentially constant oxide thicknesses found here (5.5 - 7.5 Å) for very different diameters, and the well defined trend seen in the bottom of Fig. 2, are all fully consistent with the Si_x cores being the optically active species in these samples. Also supporting this interpretation are recent high resolution luminescence data from these nanocrystals [10], which reveal vibronic structures directly related to Si.

Despite this strong evidence for the Si cores being the active chromophores, there have been suggestions that it is the interfacial suboxide species which is the light-emitting species. Variations in SiO_x thickness or stoichiometry with nanocrystal size are believed to account for the observed variations in luminescence peak energy. Figure 3 argues against this suggestion, showing that the suboxide component, isolated by subtracting the outer core of SiO₂ and the inner core of Si (represented here by a por-Si sample [11], see below), is experimentally indistinguishable in the *m* and *vs* nanocrystals, both in edge position, i. e., chemical composition, and relative edge jump, i. e., thickness (measured to be 1.5 ± 0.5 Å). Identical results are also obtained for the *l* and *s* nanocrystals.

NEXAFS and ir Absorption from Por-Si

Normalized NEXAFS data from por-Si appear in Fig. 1(b), along with comparison data from *c*-Si and HF-rinsed *a*-Si. For clarity, we show only two of the four differently prepared samples studied here, labelled *C* [2] and *B* [7]. The qualitatively closer resemblance of the por-Si data to that of *c*-Si rather than *a*-Si, and the lack of (Si-O)-related absorption at ~1842 - 1848 eV, already noted elsewhere [7], are apparent.

As in the case of the Si nanocrystals, there have been suggestions that an oxygen-containing Si species is responsible for the observed visible luminescence in por-Si. Even freshly prepared por-Si that has been minimally exposed to air contains oxygen in the form of physisorbed water trapped in the pores. This oxygen goes undetected in the Si *K*-edge NEXAFS data because the samples from which the x-ray absorption measurements are made are not exposed to exciting radiation, i. e., the oxygen remains in the form of water and does not react with the Si. By contrast, the photoluminescence measurements necessarily involve exciting radiation, which dissociates the water and leads to formation of Si-O species. That this process of photo-induced oxidation occurs is not the issue; rather, it is whether the so-generated Si-O species is related to the observed photoluminescence.

We have investigated this point using transmission ir-absorption measurements to monitor the dependence of photo-oxidation as a function of radiation exposure. The top of Fig. 4 shows transmission data from two non-irradiated, differently prepared por-Si samples, *B* and *C*. The bottom traces are *difference* spectra from those samples *before and after* exposure to 4 min of 390-nm radiation from a high pressure Xe lamp at ~ 2 mW/cm², conditions identical to those used in our photoluminescence measurements. In both samples there is enhanced absorption around 1100 cm⁻¹, representative of Si-O-Si stretching frequencies, and a corresponding loss of absorption around 2100 cm⁻¹, representative of Si-H stretching frequencies. Expanded views of these regions in the por-Si(*C*) sample are shown in Fig. 5 for different exposure times. A clear monotonic trend with exposure is observed for the Si-O and Si-H species (this trend saturates after ~ 100 sec).

Significantly, however, the luminescence peak intensities and energies remain *constant* with radiation exposure. We therefore conclude that there is no correlation between the photo-oxidized species and the observed photoluminescence.

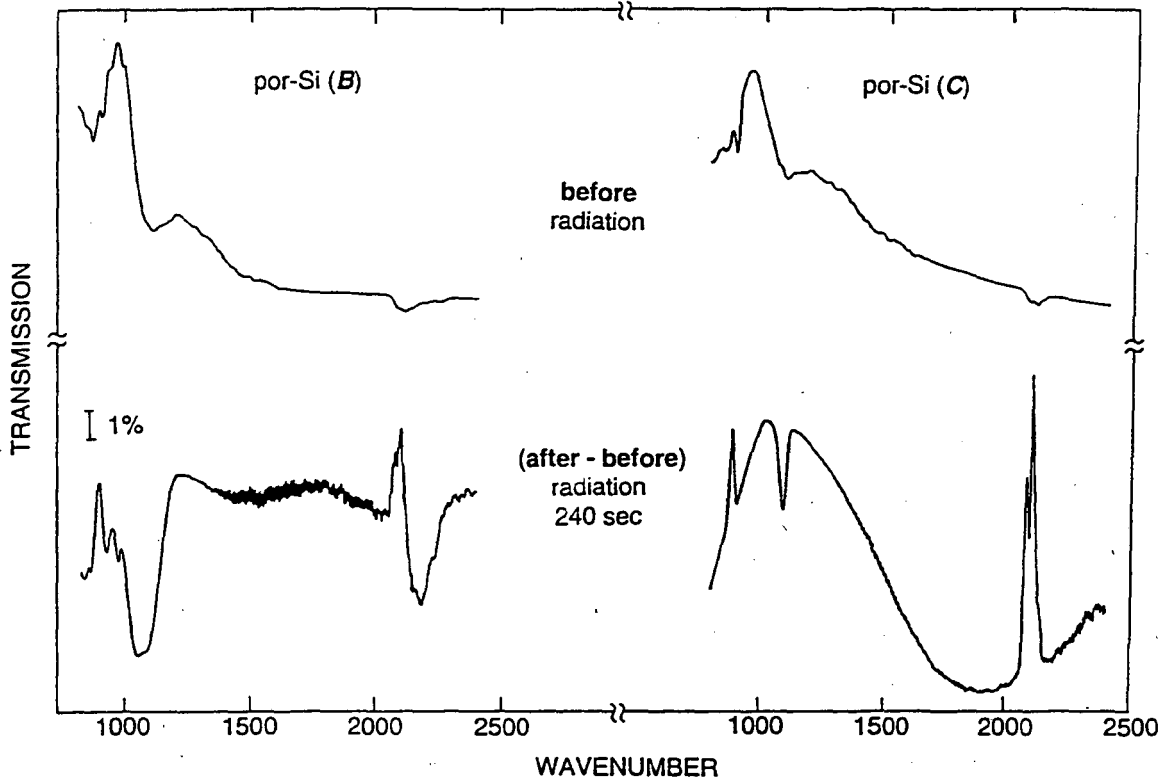


Fig. 4. Infra-red absorption data from two por-Si samples. Photo-induced oxidation due to radiation is evident in difference spectra (bottom traces) by appearance of Si-O species at $\sim 1100 \text{ cm}^{-1}$ with a corresponding decrease of Si-H species at $\sim 2100 \text{ cm}^{-1}$.

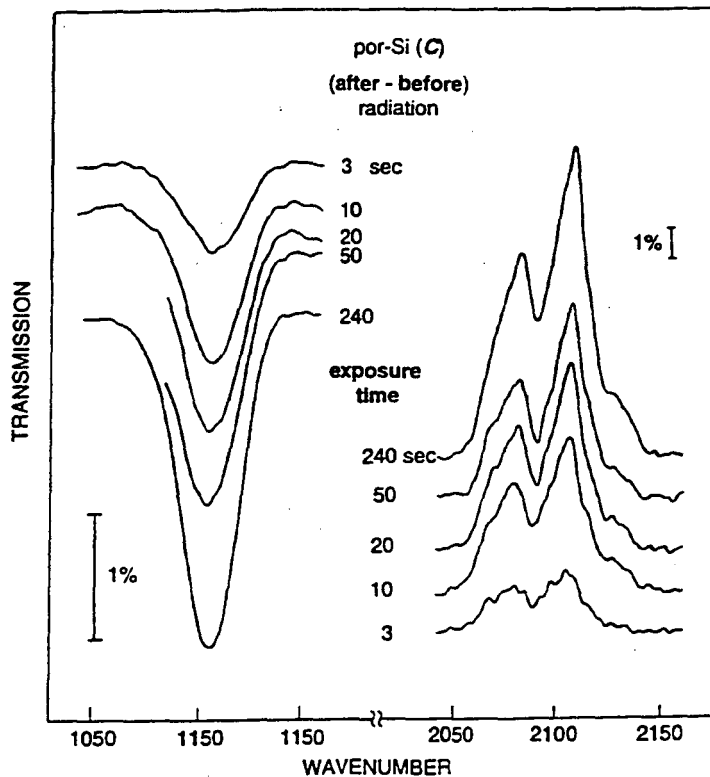


Fig. 5. Infra-red absorption difference spectra from a por-Si sample showing dependence of photo-oxidized species formation and loss of Si-H species as a function of radiation exposure. These processes are unrelated to the measured luminescence intensity and peak energy (not shown), which are independent of exposure.

EXAFS from Por-Si

While the NEXAFS data provide chemical information, i. e., electronic structure, the corresponding extended x-ray absorption fine structure (EXAFS) data readily provide local coordination and distance information, i. e., geometric structure [12]. In Fig. 6(a), we show the raw EXAFS data for the same samples displayed in Fig. 1(b). Locally ordered structure in the por-Si samples is seen from the second- and third-neighbor shells appearing as peaks at ~ 3.4 and 4.1 Å in the Fourier-transformed (FT) data [Fig. 6(b)]; such structure in *a*-Si is essentially absent, of course, due to static disorder. Relative to bulk *c*-Si, the apparently similar reduction of first-neighbor FT peak intensities at ~ 2.0 Å for the por-Si and *a*-Si samples is better understood by filtering and back-transforming the first-neighbor shells [Fig. 6(c)]. The peak intensity of the amplitude function $A(k)$, which envelopes the oscillatory EXAFS $\chi(k)$, is shifted for *a*-Si as a result of the Debye-Waller-like static-disorder term [12]. Such exponential damping of high- k scattering is unimportant for $k < 4$ Å⁻¹, so in that region $A(k)$ for the *a*-Si and *c*-Si samples is the same. This reflects their identical Si coordinations of 4. By contrast, $A(k)$ for the por-Si samples is *unshifted* relative to that for *c*-Si and is *smaller* at *all* values of k . This is an obvious indication that the average number of first-neighbor Si atoms, n_{Si} , in por-Si is < 4 .

The reason for the lower average Si coordination number and the apparent lack of other first neighbors in the FT data is that the surface Si atoms are also coordinated to H (the EXAFS backscattering amplitude of H is negligible). The substantially smaller values of n_{Si} obtained by fitting [12] the filtered por-Si data indicate large H/Si ratios in these samples. For example, $n_{\text{Si}} = 2.85 \pm 0.1$ in sample *B*, meaning H/Si = 1.15 ± 0.1 , or an average H content of 115%. This should be compared with typical H concentrations of $\sim 10\%$ in device-quality *a*-Si:H [13]. Forward α -recoil measurements [14], which determine H/Si directly, give results for the corresponding

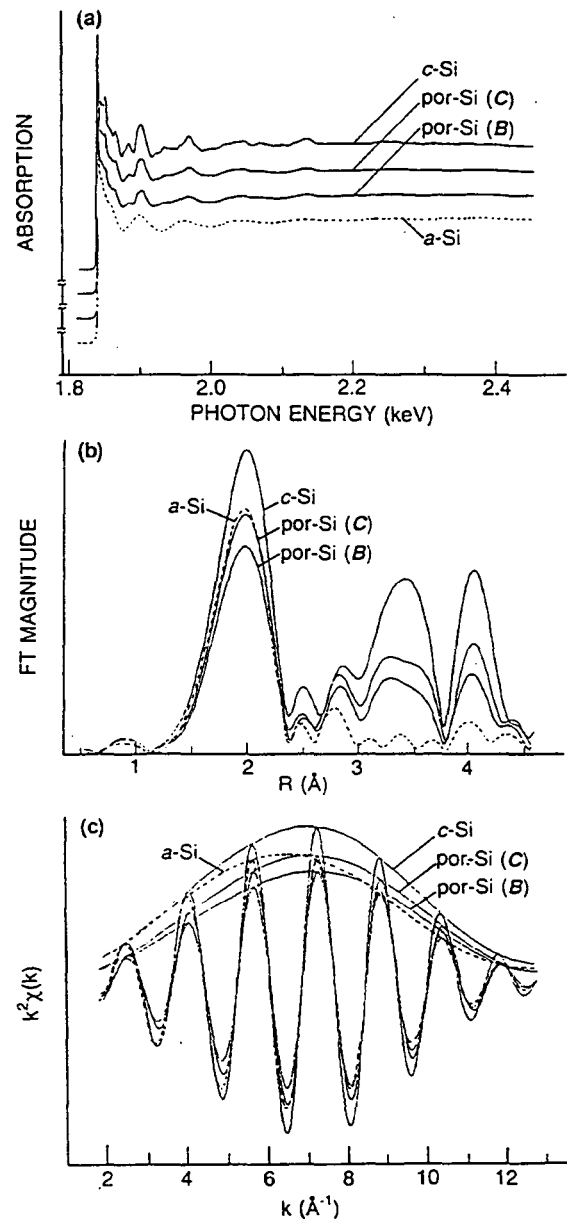


Fig. 6. (a) Raw Si *K*-edge EXAFS data from *c*-Si, *a*-Si, and two different por-Si samples. (b) Fourier transforms of edge-truncated, k^2 -multiplied, background-subtracted data from samples in (a). 1st-, 2nd-, and 3rd-shell peaks at ~ 2.0 , 3.4 , and 4.1 Å are uncorrected for phase shift. Artifactual peaks at ~ 2.5 and 2.9 Å are due to truncation. (c) Back-transformed, filtered 1st-shell data from (b). Reduced EXAFS amplitudes from por-Si samples indicate lower Si coordination as a result of surface coordination to H. Measured average Si coordinations for different por-Si samples are plotted in Fig. 2.

por-Si samples that are completely consistent with those from EXAFS. Furthermore, these integrated H concentrations are confirmed by transmission ir-absorption measurements [15] (the ir results also show that H from bulk Si or from possible OH contamination is unimportant in the α -recoil data).

The large measured H/Si ratios imply large surface/volume ratios for the representative Si structures, the average sizes of which are determined in the top half of Fig. 2. We first plot calculated Si coordinations for different shapes as a function of inverse characteristic dimension, or length (i.e., diameter for a sphere, side for a cube). The dot-dashed line, drawn from the asymptotic limit of $n_{\text{Si}} = 4$ for infinitely large Si dimensions, is extended for (100)-faceted cubic particles of decreasing size. The 4:1:1 square-prism shapes are included to represent prolate particles. All particle surfaces (terminated with H) are ideal. To this calculated plot we then indicate the Si coordinations measured from the EXAFS amplitudes in the four different por-Si samples, labelled *A*, *B*, *C*, and *D* (all n_{Si} values are ± 0.1). Finally, the average Si particle sizes are obtained from the intersection of mean experimental Si coordinations with the asymptotic dot-dashed line. These sizes are plotted as squares in the bottom half of Fig. 2 against the peak luminescence energies measured in the corresponding samples. A correlation is obvious.

DISCUSSION

Before discussing the trend at the bottom of Fig. 2, we point out that the particle sizes represented by the squares are actually upper limits. The quoted particle sizes are obtained from the (100)-cubic asymptote, so any other assumed shape(s) would clearly lead to smaller sizes. Furthermore, TEM micrographs of the por-Si samples show them to contain *c*-Si, often in the form of columns, whose dimensions are ≥ 3 nm and whose amounts vary between 10 - 30% depending on the type of sample studied, e. g., *B* versus *C*. The remainder of the samples consists of Si structures too small to characterize with TEM. Since EXAFS measurements average over *all* the Si structures, including the contributions from *c*-Si, the reduced values quoted for n_{Si} – and thus the mass-weighted average particle sizes – would be even smaller had these larger *c*-Si contributions not been included. It is significant to note that subtracting the *c*-Si contributions from the por-Si data still leaves evidence for local crystallinity, i. e., the 2nd- and 3rd-neighbor shells in the FT data are not removed.

The bottom of Fig. 2 displays the important and surprising result that the relationship between peak luminescence energy and average particle size for the por-Si samples is nearly identical to that for the Si nanocrystals. Indeed, excluding the unetched *c*-Si contributions to our quoted n_{Si} values make the two trends virtually indistinguishable. The implication is clear: since the luminescence spectrum from a given Si particle is directly correlated with its size, the luminescence spectrum from a given por-Si sample is also associated with a characteristic Si dimension. Thus, for example, 720-nm peak luminescence from Si is representative of a structure whose average size is ~ 20 Å, *regardless of how the sample was prepared*.

The correlation in Fig. 2 also provides information about the shape of the Si species responsible for the luminescence in the *visible* region < 700 nm (> 1.75 eV). The columnar, extended-wire-like structures observed with TEM in por-Si samples *A* and *B* are all > 20 Å, i. e., too large to account for our EXAFS results. If extended wire shapes were responsible for the luminescence in these samples, they would have to be of extremely small

dimensions, $<10 \text{ \AA}$. This implies, therefore, that the optically active structures are generally not extended wires, but particles. We infer that these particles are predominantly (100)-faceted based on surface sensitive reflection ir-absorption measurements [16] from the por-Si *A* and *B* samples, which indicate H-Si frequencies representative of SiH_2 species [(110) or (111) surfaces would exhibit mainly SiH frequencies]. More general conclusions cannot be extended to samples luminescing in the near- and far-infrared region, i. e., $>700 \text{ nm}$, because the distinction between wires and particles [see Fig. 2] is less significant.

The average sizes of the Si structures responsible for the *visible* luminescence in por-Si are small, well below easy detection with TEM or diffraction. For example, Fig. 2 indicates that 2-eV luminescence is associated with particles of dimensions typically less than 13 \AA (recall that the values shown in squares are upper limits). This is considerably smaller than any size previously associated with such luminescence. A 13-\AA cube contains a total number of Si atoms $N_{\text{Si}} \approx 110$ (≈ 60 for a sphere [17], see Fig. 2) and a calculated band gap of $\sim 3.8 \text{ eV}$ [18], whereas a 26-\AA cube has corresponding numbers of ~ 900 and 2.2 eV . Factors of two in size are therefore very important.

Basic questions about visible photoluminescence from por-Si still remain. Among these is a reliable calculation of the size of the coulombic interaction between electrons and holes localized on Si particle. A complete description of the visible-luminescence mechanism in por-Si will ultimately need to address this and other issues, along with the fact that the optically active Si species in this material has dimensions as small as those found here.

ACKNOWLEDGEMENTS

S. Schuppler is grateful to the Alexander-von-Humboldt Foundation for financial support. The EXAFS experiments were performed at the NSLS, Brookhaven National Laboratory, which is supported by the DOE, Division of Materials Sciences and Division of Chemical Sciences. The TEM measurements, performed at the National Center for Electron Microscopy, LBL, are supported by the Director, Office of Basic Energy Sciences, DOE, under contract number DEAC03-76SF00098.

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16. Surface sensitivity of <1 μ m was obtained using a grazing internal incidence angle in a Ge plate positioned next to the por-Si samples.
17. For a Si cube of side L , $N_{\text{Si}} = 8L^3/a_0^3$; for a sphere of diameter L , $N_{\text{Si}}^{\text{sph}} = (\pi/6)N_{\text{Si}}^{\text{cube}}$.
18. See, e. g., C. Delerue, G. Allan, and M. Lannoo, *Phys. Rev. B* **48**, 11024 (1993), and references therein.

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